

Aug 2013 ←

Ceder Group, Oct 2013

54%

← Aug 2015

46%

Apr 2017 →

~~"The two coast MIT PhD"~~

Thermodynamics and kinetics of Mg intercalation for multivalent cathode applications

Sai Gautam Gopalakrishnan

Thesis advisor: Prof. Gerbrand Ceder

Defense: April 25, 2017



Types of modern batteries

Primary:
“Use once”



Secondary:
“Rechargeable”



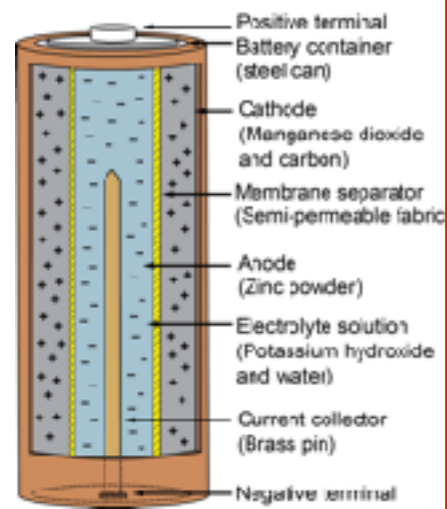
Solid-liquid-solid

Intercalation-liquid-Intercalation

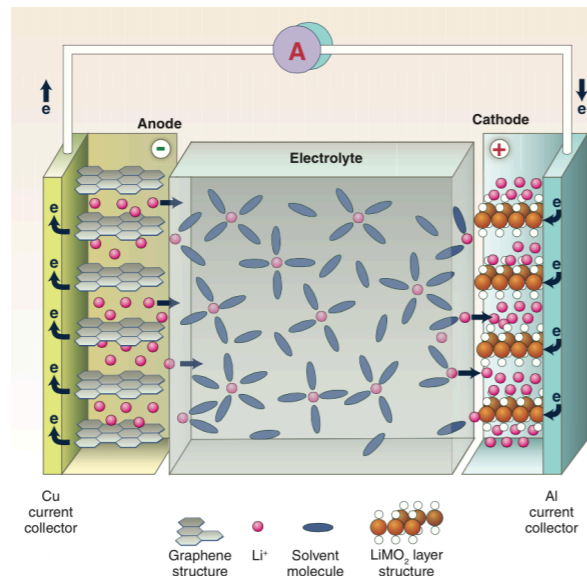
Liquid-Solid-Liquid

Solid-solid-solid

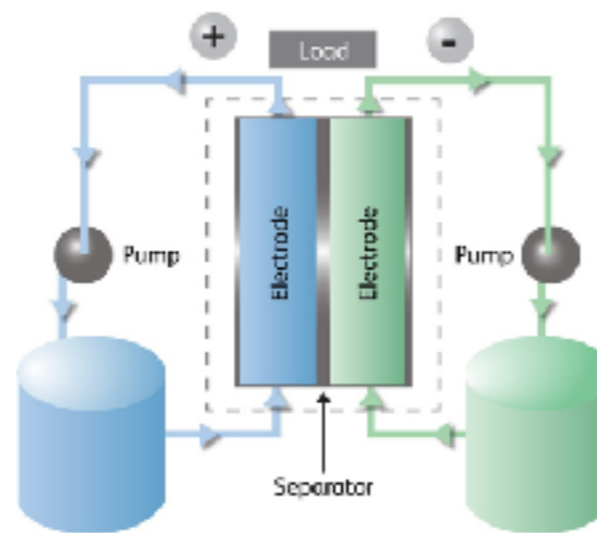
Liquid-liquid-liquid



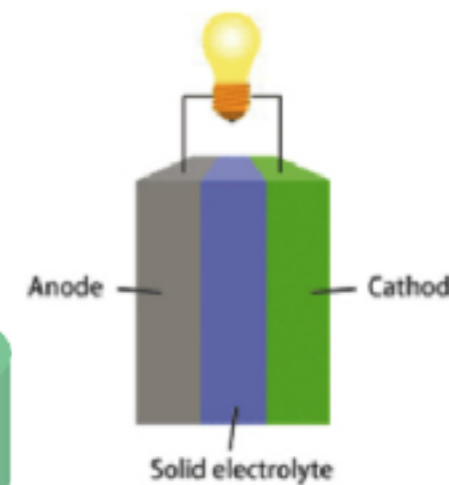
Alkaline



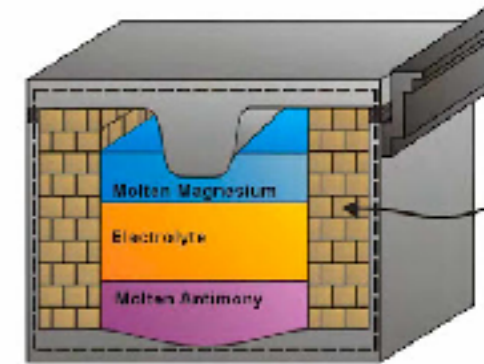
Li-ion



Flow

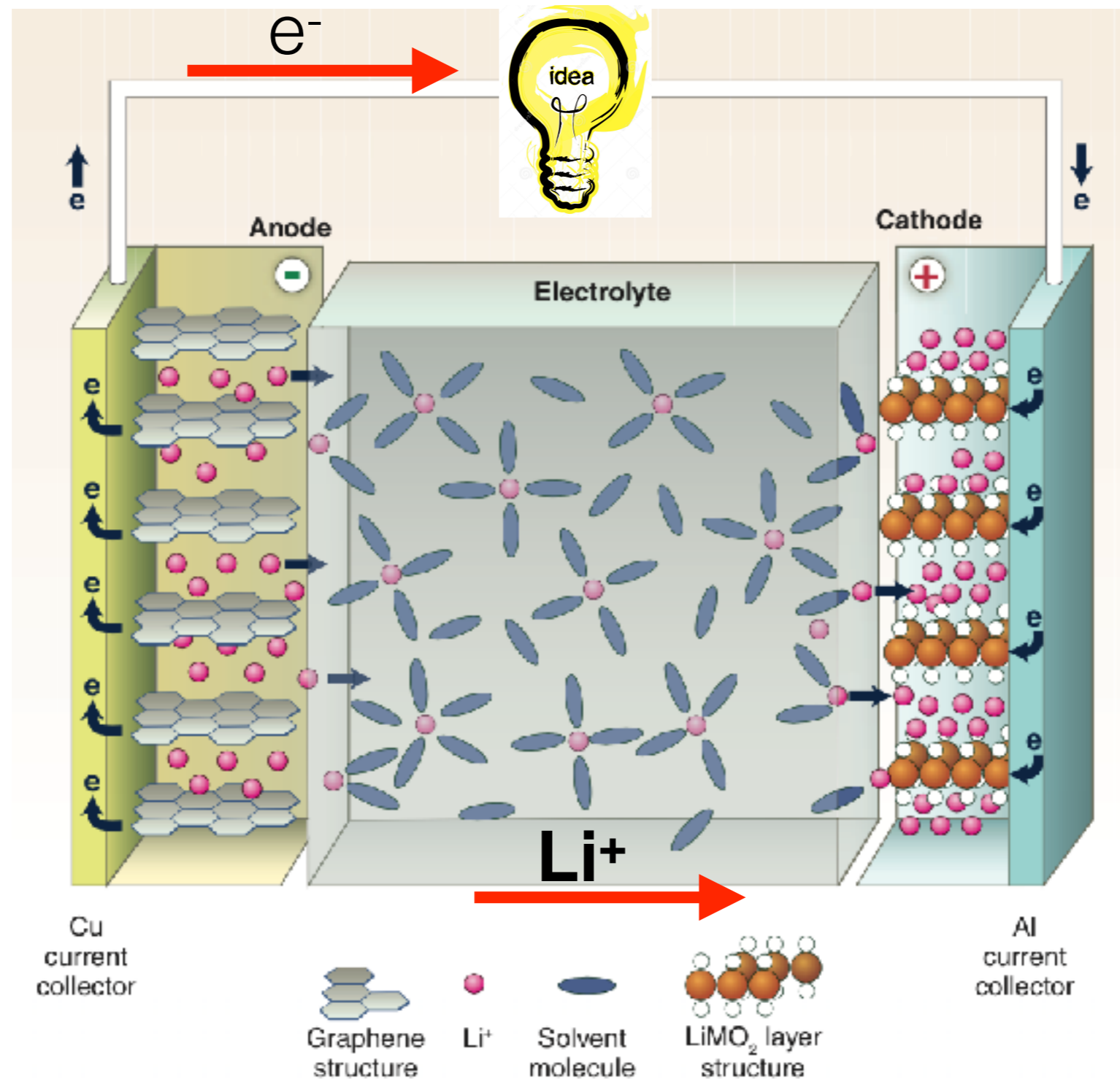


All-solid-state

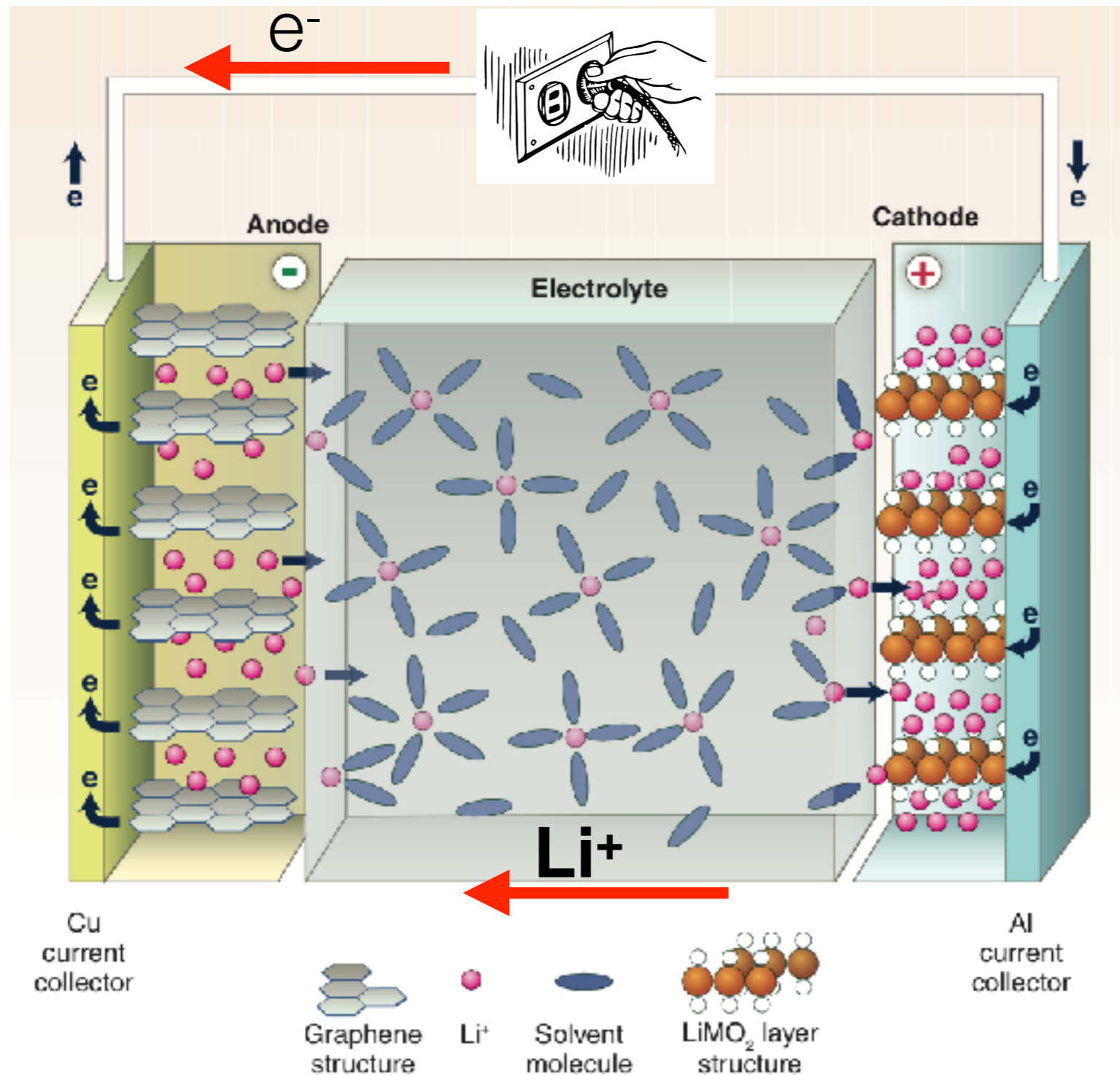


Liquid metal

Batteries 101: How do modern Li-ion batteries work?



Batteries 101: How do modern Li-ion batteries work?



Batteries 101:

How do modern Li-ion batteries work?

Voltage (V) : Potential to do work, $\propto (-\nabla\mu_{\text{Li}})$

Capacity (mAh/g) : Amount of charge stored (per electrode)

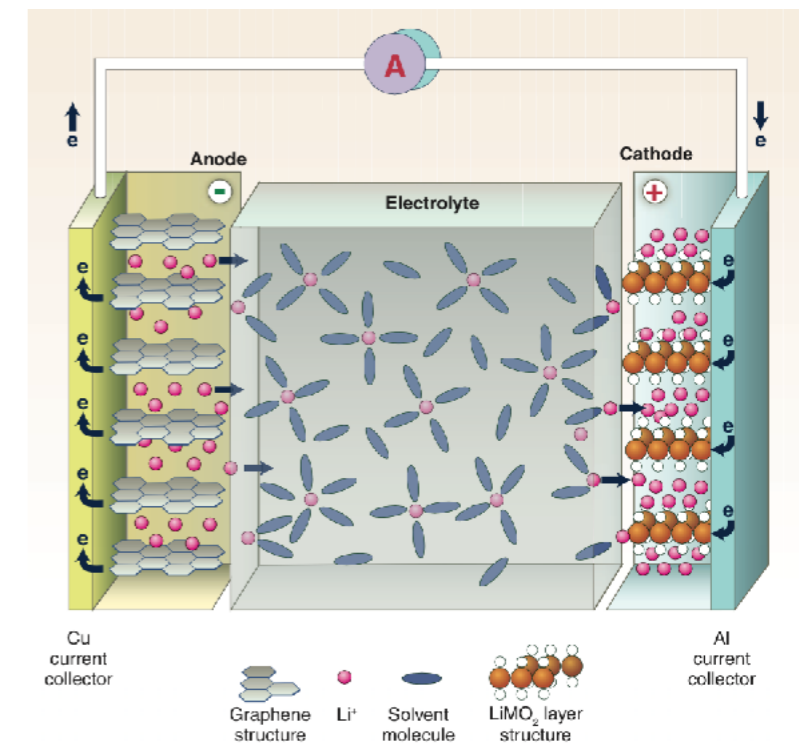
Energy stored (Wh) : Voltage x Capacity x Mass of cell

Gravimetric Energy density
(Wh/kg) : Energy stored per weight of cell

Volumetric Energy density
(Wh/l) : Energy stored per volume of cell

Rate: How fast can you (dis)charge?
Depends on ionic “mobility”

Electrochemical cycle : Charge + Discharge



Why Multi-valent (MV) batteries?

- Next generation of electric devices will benefit from higher energy density storage systems
 - ◆ Multi-valent == More electrons (Mg^{2+} , Ca^{2+} , Al^{3+} , etc.)
 - ◆ Li-ion technology approaching fundamental limits
 - ◆ Smaller batteries useful for portable electronics
 - ◆ Lighter batteries favorable for electric vehicles

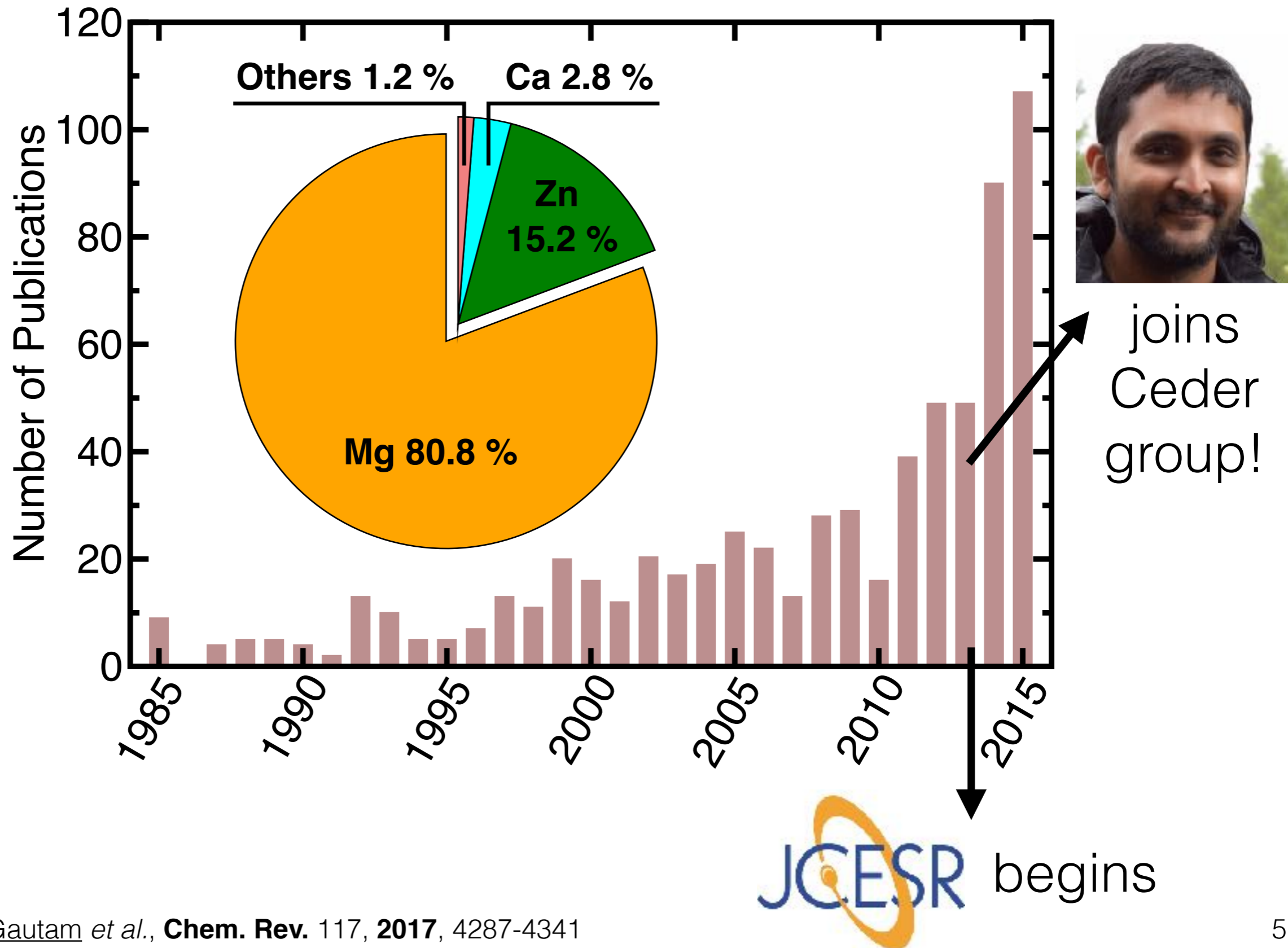


- Why Mg?

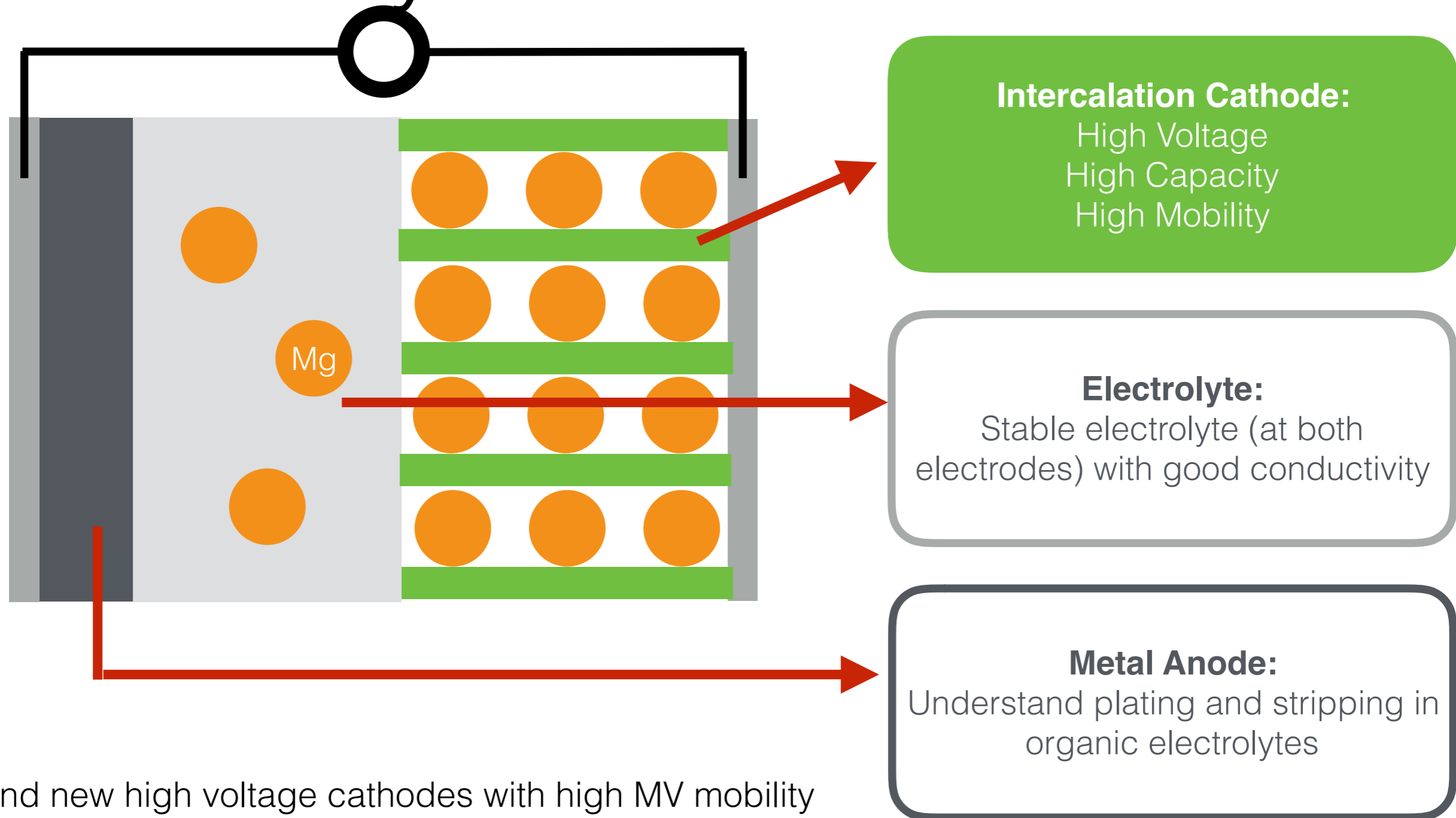
- ◆ Superior volumetric capacity for Mg metal as anode ($\sim 3833 \text{ mAh/cm}^3$) vs. Li metal (~ 2046) or Li in graphite (~ 800)
- ◆ Mg is safer than Li
- ◆ Mg is “geopolitically” cheaper than Li



Increasing focus on MV technology



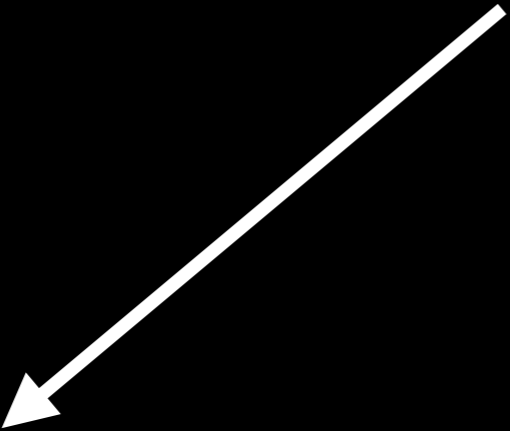
Challenges of using a MV chemistry



1 Find new high voltage cathodes with high MV mobility

2 Understand MV intercalation mechanisms in existing cathodes

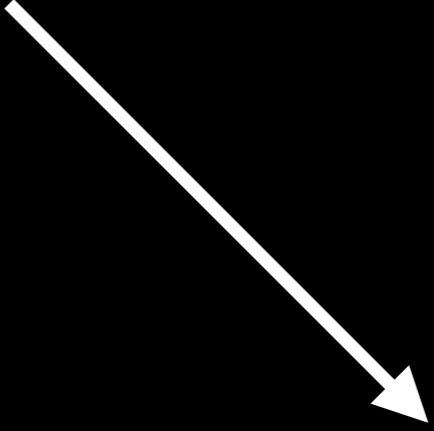
**MV
cathode
design
challenge**



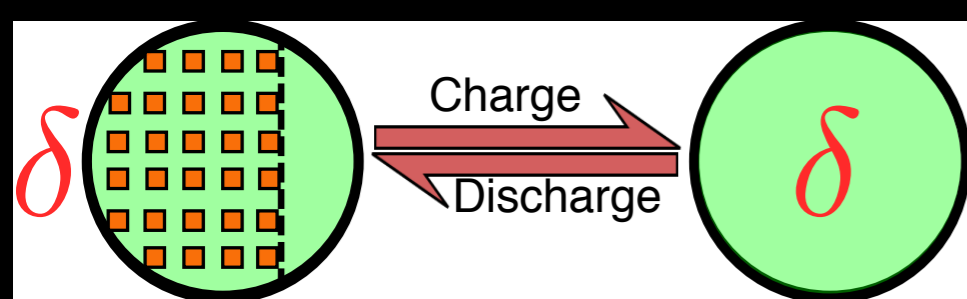
**High Voltage
(> 2.5 V)**



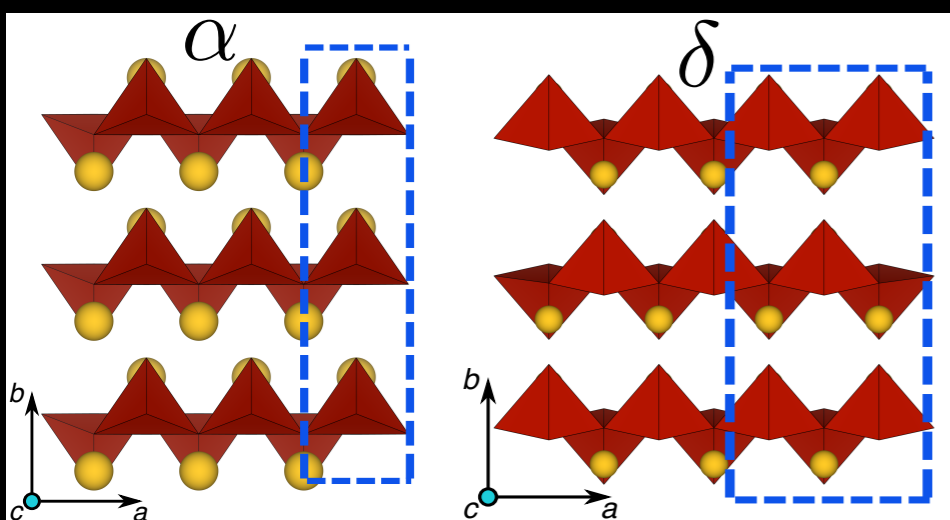
**High Capacity
(> 200 mAh/g)**



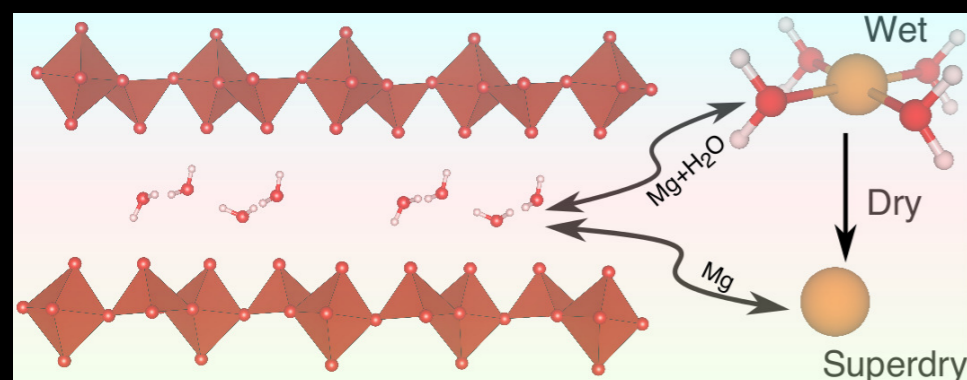
**High Mobility
?**



“The intercalation phase diagram of Mg in V_2O_5 from First-principles”,
 G.S. Gautam et al., **Chem. Mater.** 27,
 2015, 3733-3742

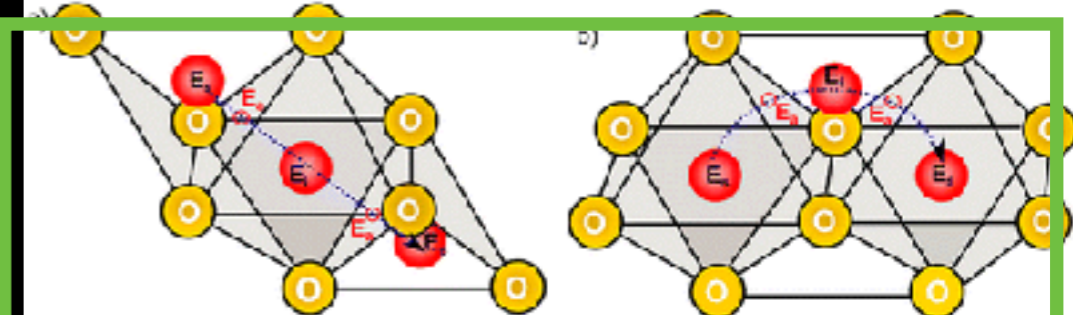


“First-principles evaluation of multi-valent cation insertion into orthorhombic V_2O_5 ”,
 G.S. Gautam et al., **Chem. Commun.** 51,
 2015, 13619-13622

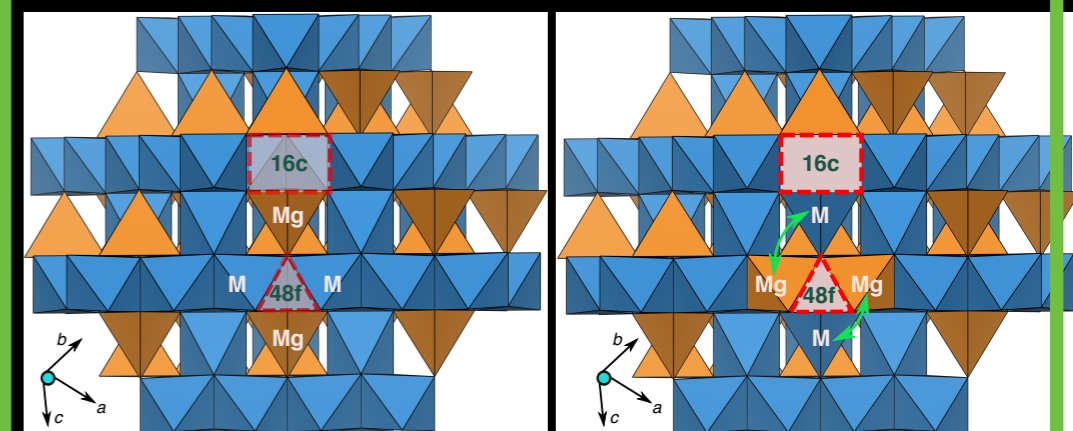


“Role of structural H_2O in intercalation electrodes: the case of Mg in nano crystalline Xerogel- V_2O_5 ”,
 G.S. Gautam et al., **Nano Lett.** 16, 2016,
 2426-2431

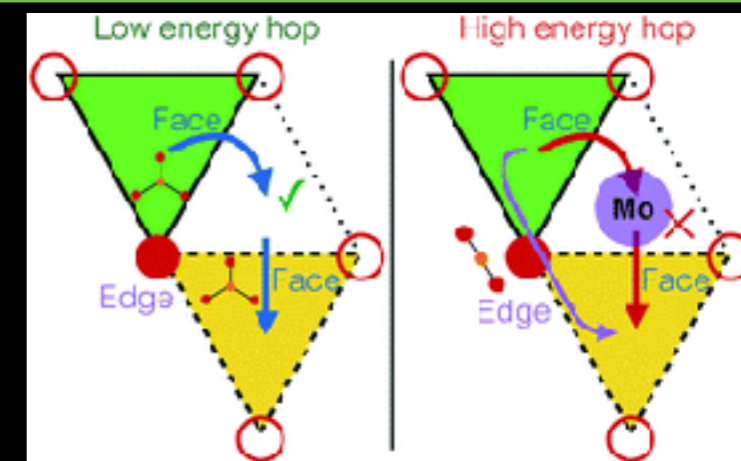
MV
 cathode
 design
 challenge



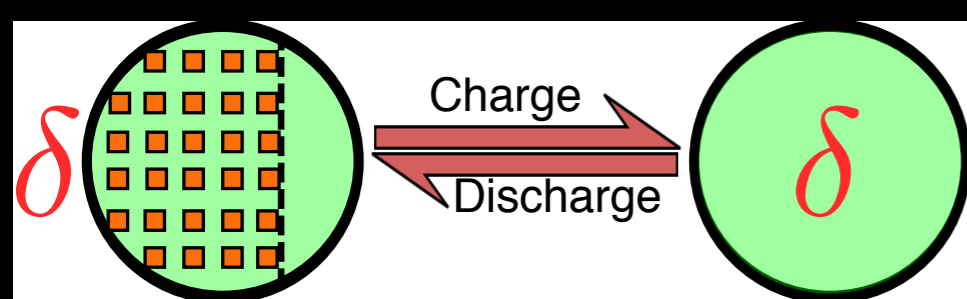
“Materials Design Rules for Multivalent Ion Mobility in Intercalation Structures”,
 Z. Rong, R. Malik, P.Canepa, G.S. Gautam et al., **Chem. Mater.** 27, 2015, 6016-6021



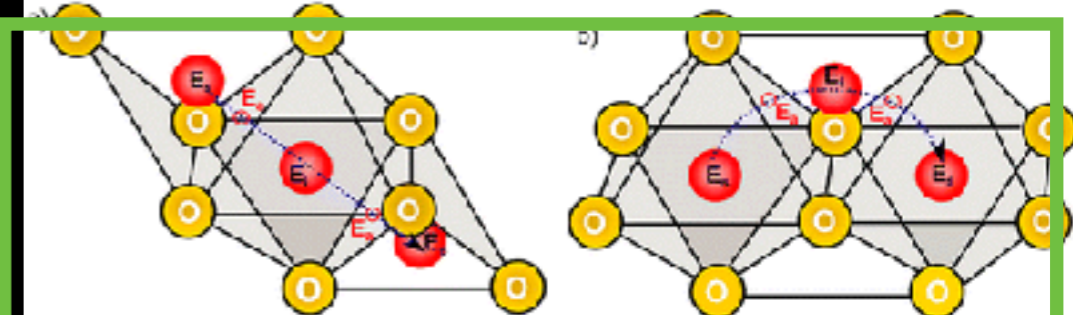
“Influence of inversion on Mg mobility and electrochemistry in spinels”,
 G.S. Gautam et al., **under review**



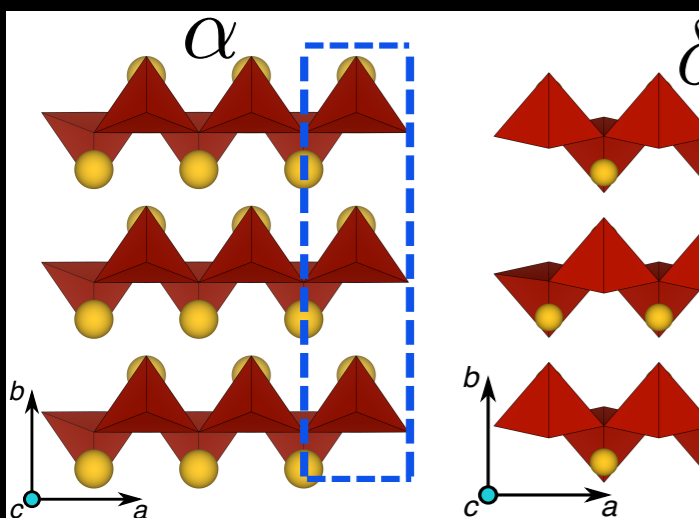
“Impact of intermediate sites on bulk diffusion barriers: Mg intercalation in $Mg_2Mo_3O_8$ ”,
 G.S. Gautam et al., **J. Mater. Chem. A** 4,
 2016, 17643-17648



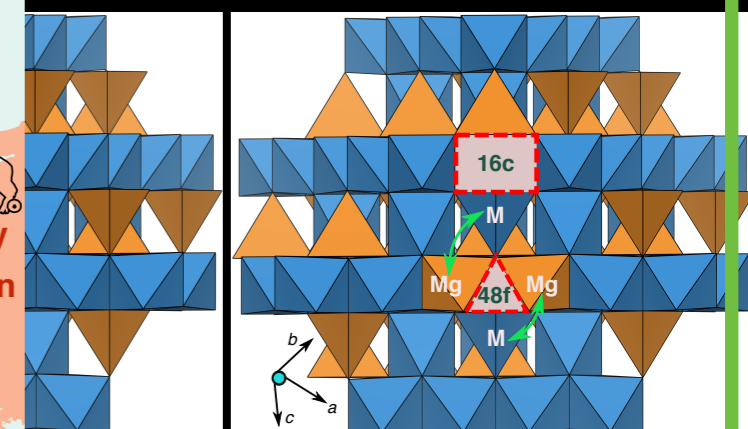
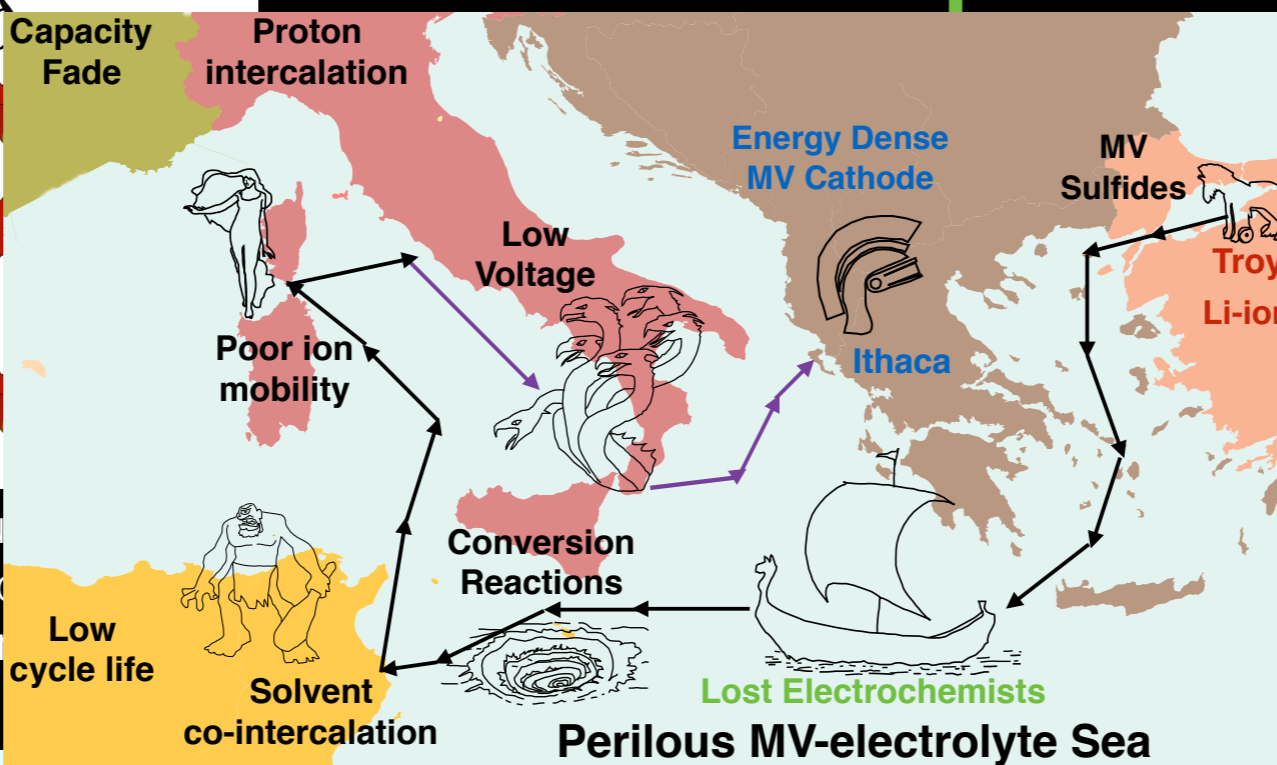
“The intercalation phase diagram of Mg in V_2O_5 from First-principles”,
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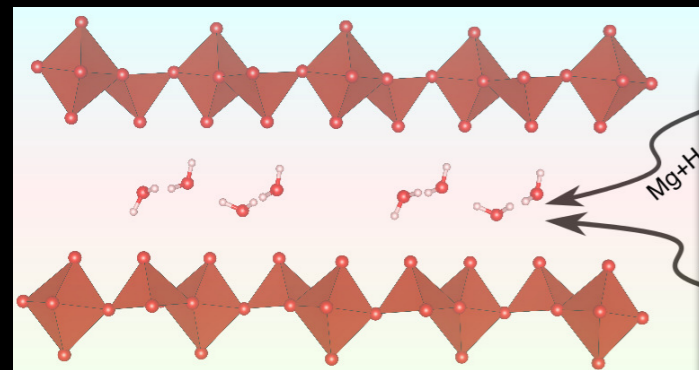
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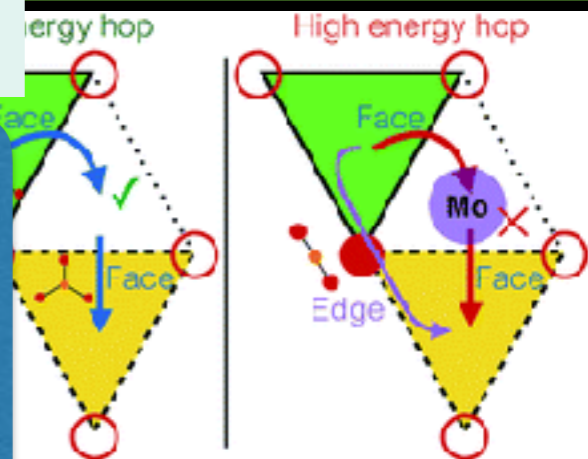


“Conversion on Mg mobility and energy in spinels”,
G.S. Gautam et al., **under review**



“Role of structural H_2O in intercalation electrodes: the case of Mg in nano-crystalline Xerogel- V_2O_5 ”,
G.S. Gautam et al., **Nano Lett.** 16, **2016**, 2426-2431

“Odyssey of Multivalent Cathode Materials: Open questions and Future Challenges”,
P. Canepa, G.S. Gautam et al., **Chem. Rev.** 117, **2017**, 4287-4341

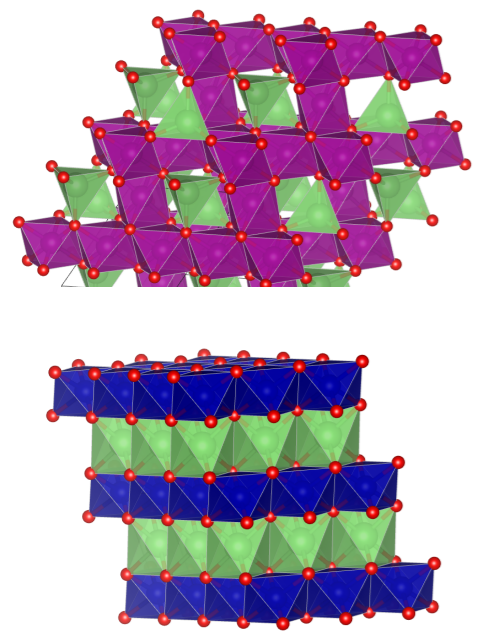


“Intermediate sites on bulk diffusion barriers: Mg intercalation in $Mg_2Mo_3O_8$ ”,
G.S. Gautam et al., **J. Mater. Chem. A** 4, **2016**, 17643-17648

Methods detour

How to calculate material properties?

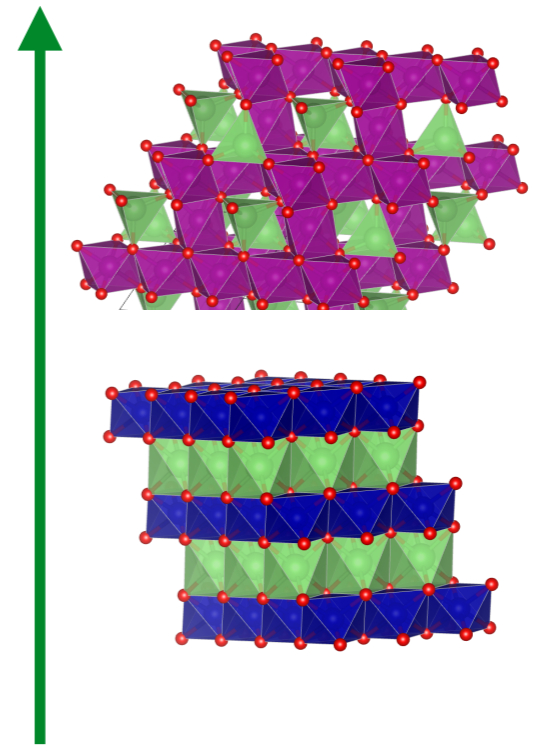
Thermodynamics: Voltages



$$H\psi = E\psi$$

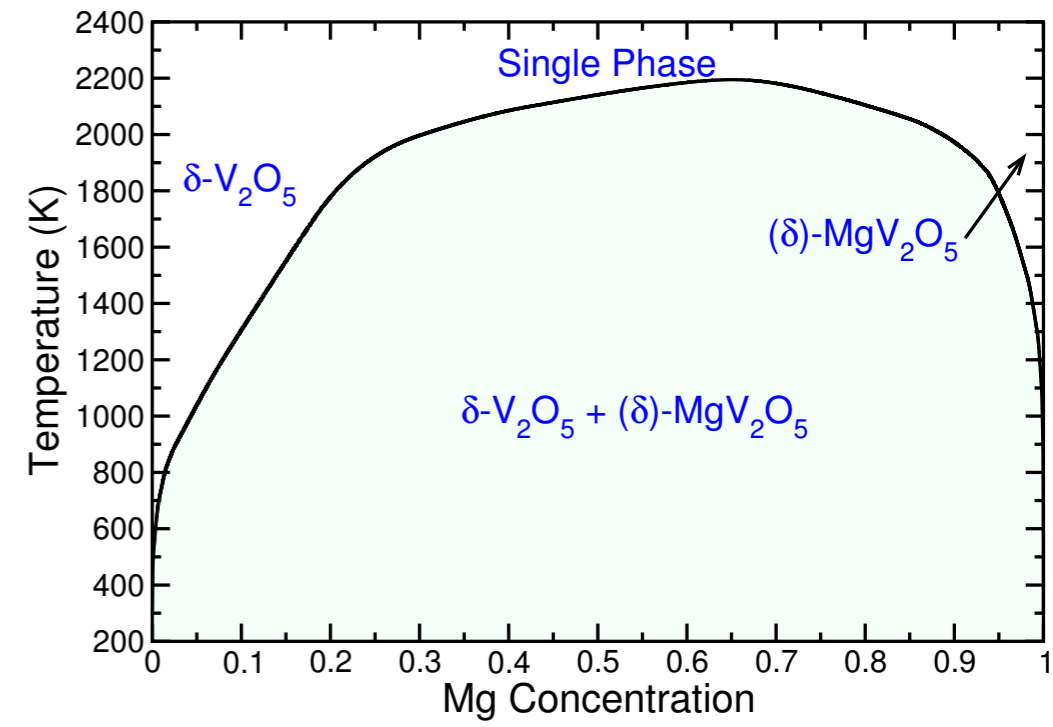


Free energy



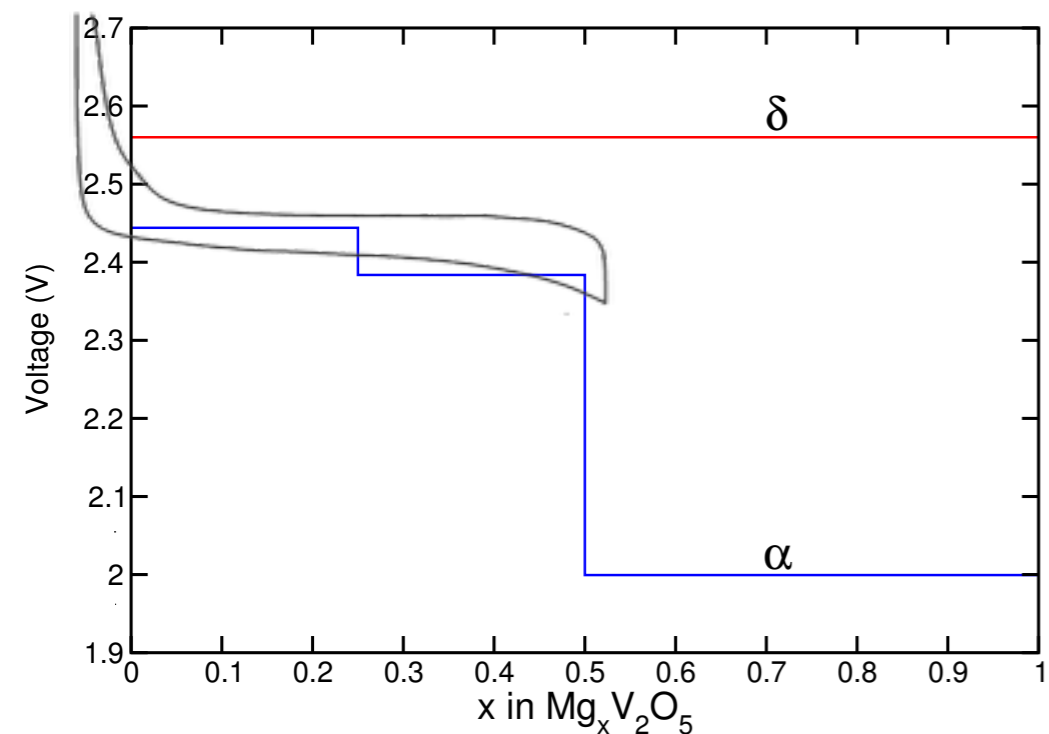
DFT+
lattice models
 x, T, p

Density Functional
Theory (DFT)¹⁻⁵



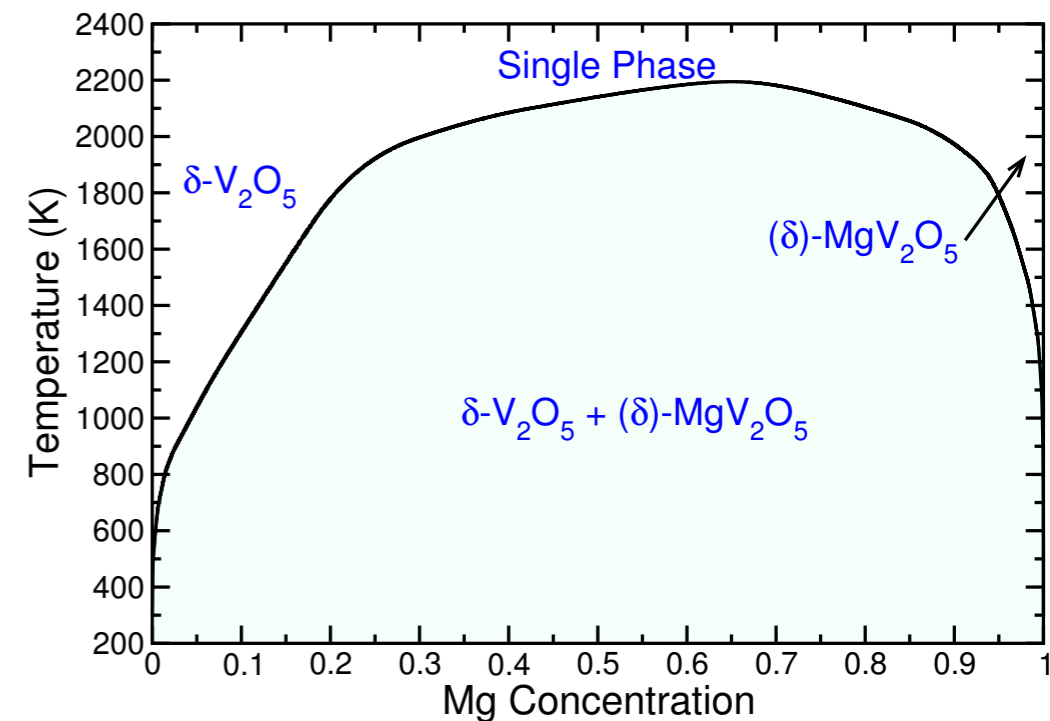
1. Kresse et al., Phys. Rev. B 1993
2. Kresse et al., Phys. Rev. B 1996
3. Perdew et al., Phys. Rev. Lett. 1996
4. Zhou et al., Phys. Rev. B 2004
5. Jain et al., Phys. Rev. B 2011

Thermodynamics: Voltages

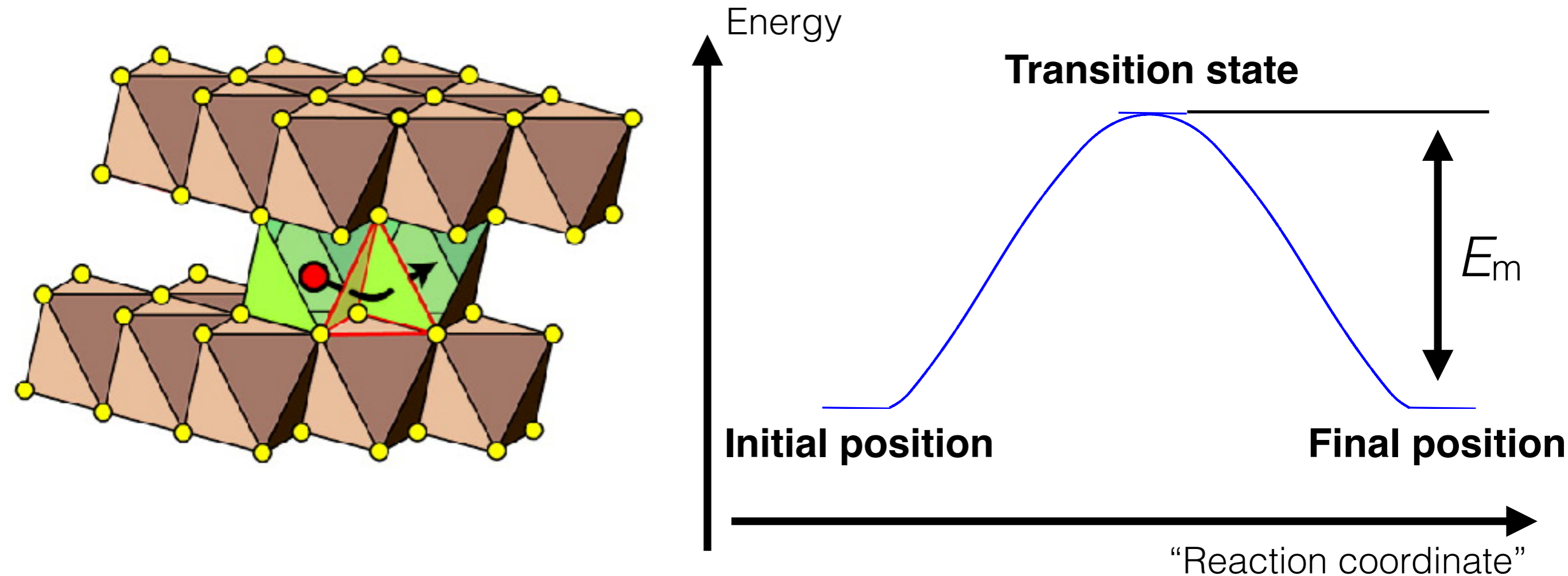


$$\Delta V = - \frac{\Delta G}{nF}$$

Nernst equation

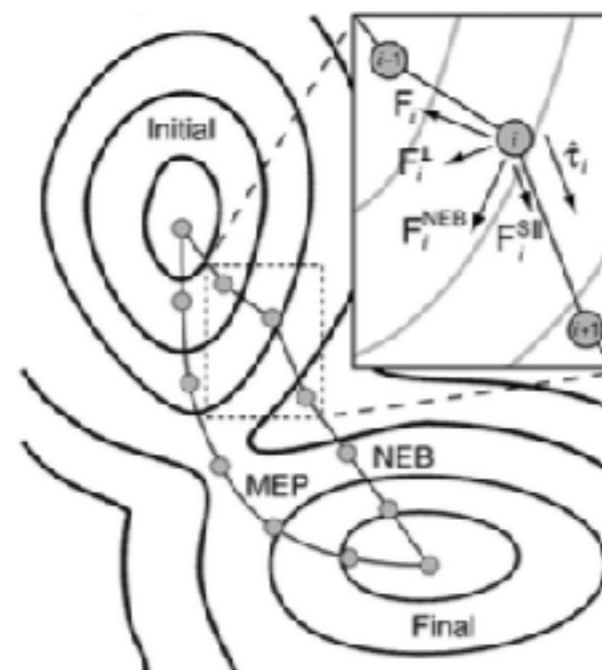


Kinetics: Mobility



$$D \propto \exp\left(-\frac{E_m}{kT}\right)$$

Diffusivity - Mobility - Flux - Conductivity



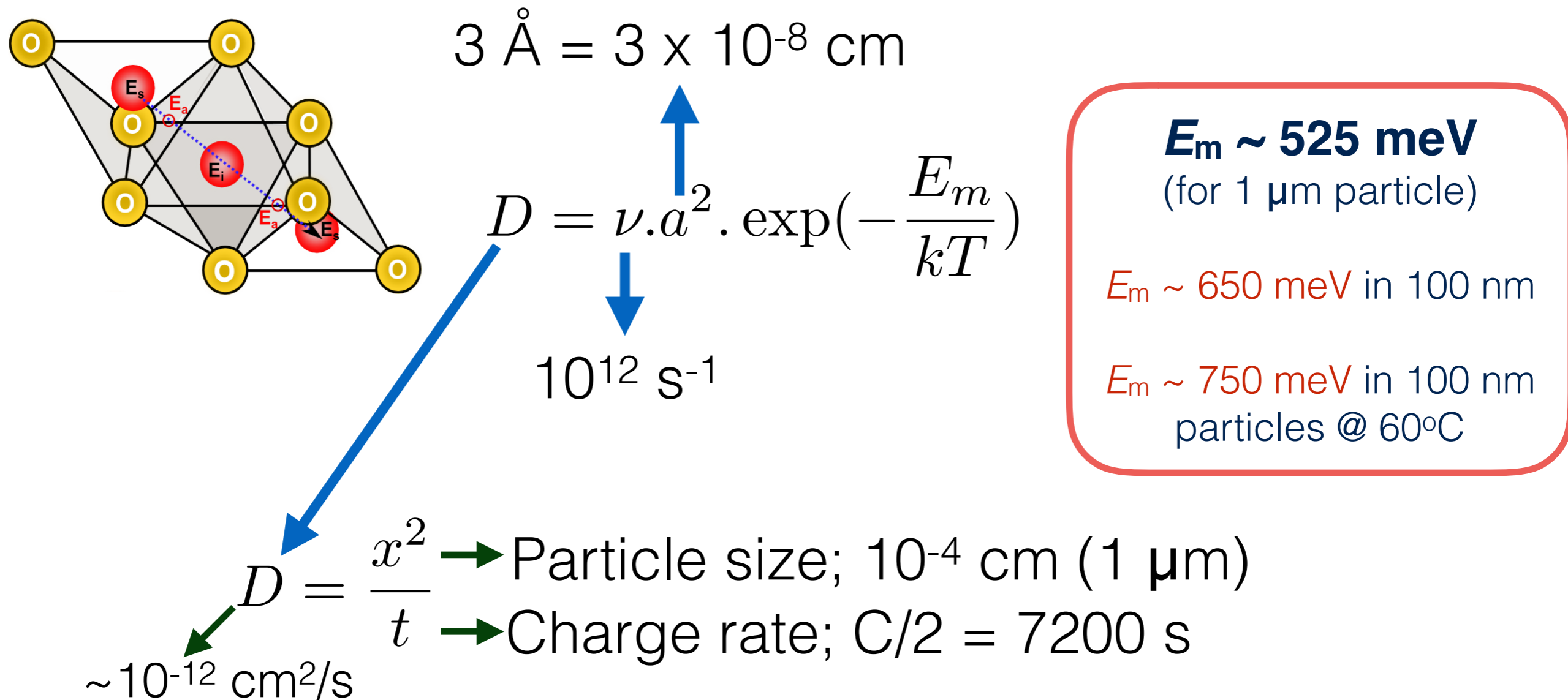
Nudged Elastic Band (NEB) method + DFT = E_m

Cathode search

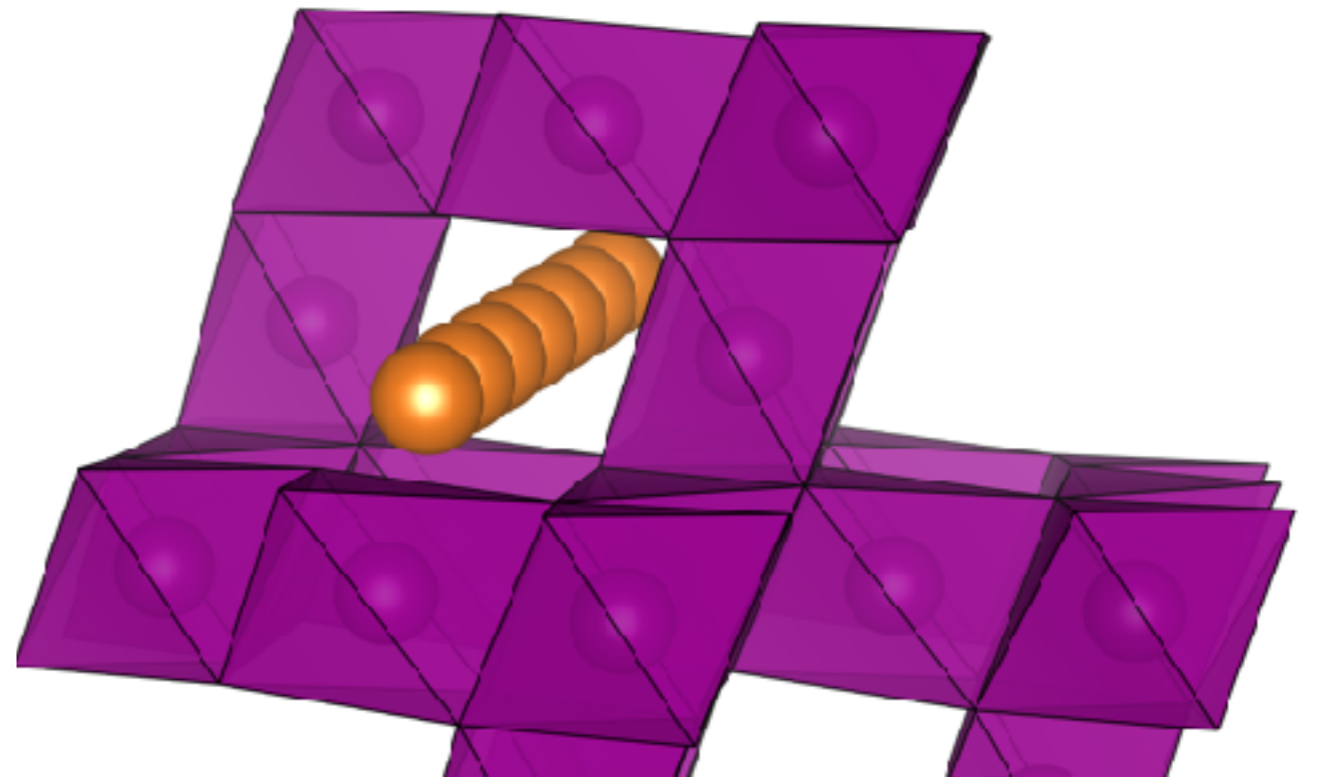
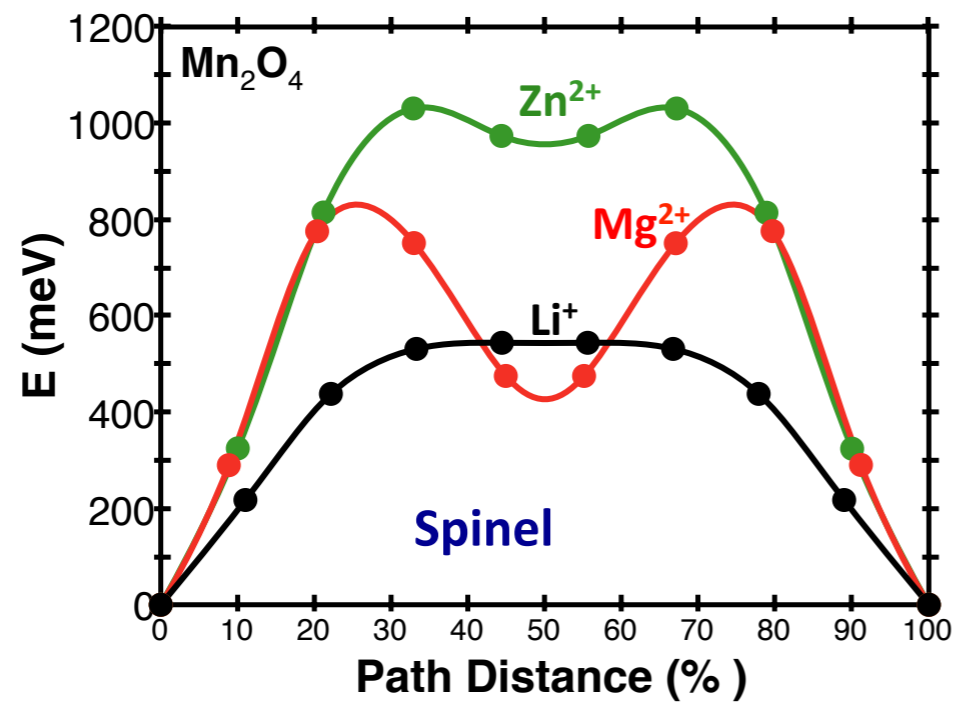
Z.Rong, R. Malik, P. Canepa, G.S. Gautam, *et al.*,
“Materials Design rules for multivalent ion mobility in intercalation structures”,
Chem. Mater. 27, **2015**, 6016-6021

Understanding MV mobility: Requirements

Order of magnitude estimates indicate a migration barrier requirement in the range of 500 — 700 meV range

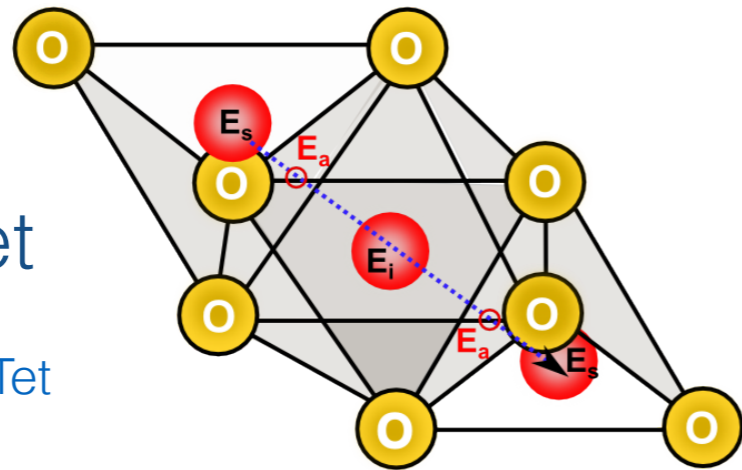


Trends differ with diffusion topology

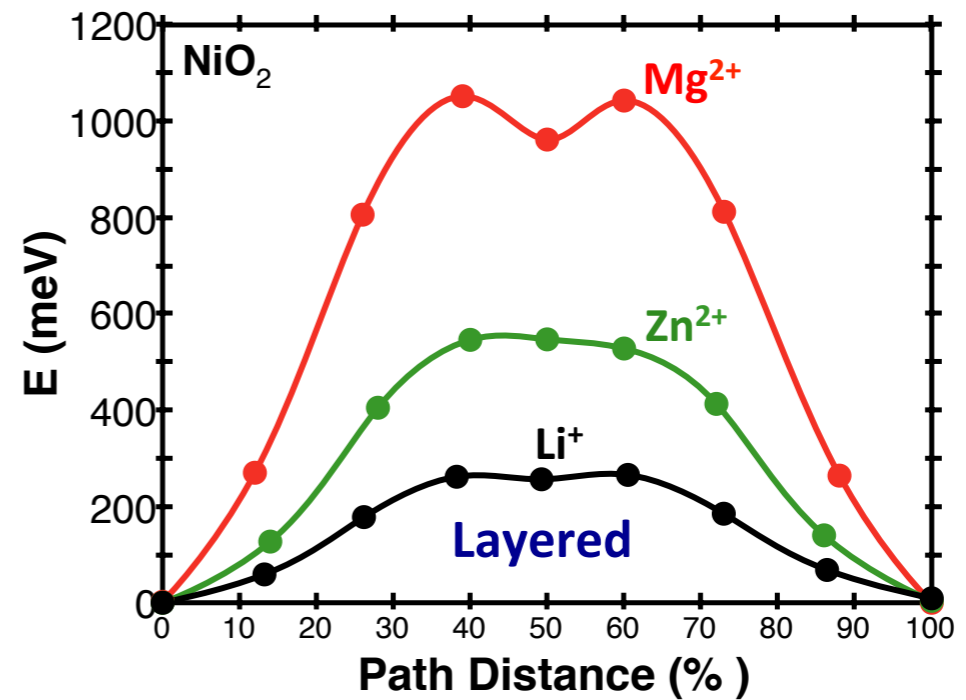
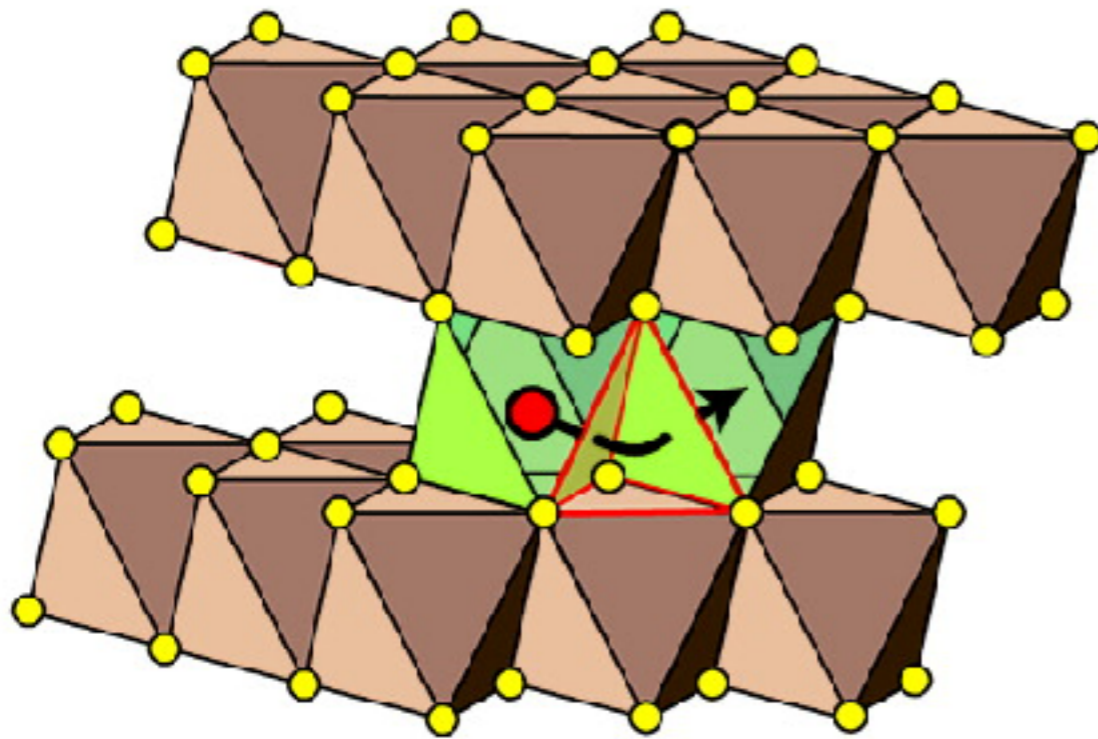


Tet - Oct - Tet

$$E_{\text{site}} = E_{\text{Oct}} - E_{\text{Tet}}$$

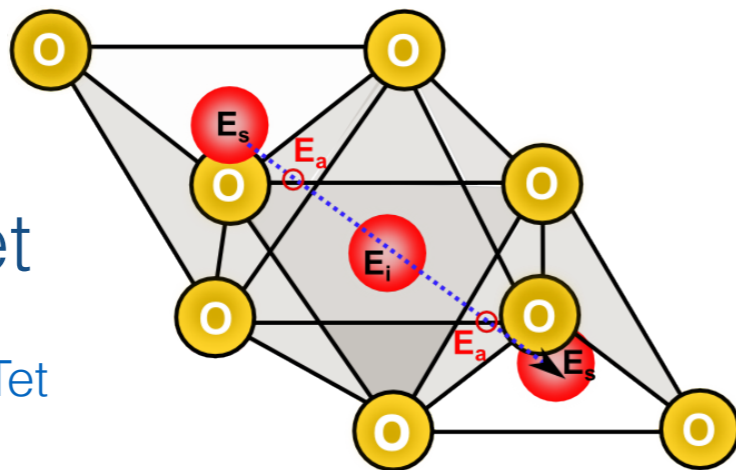


Trends differ with diffusion topology



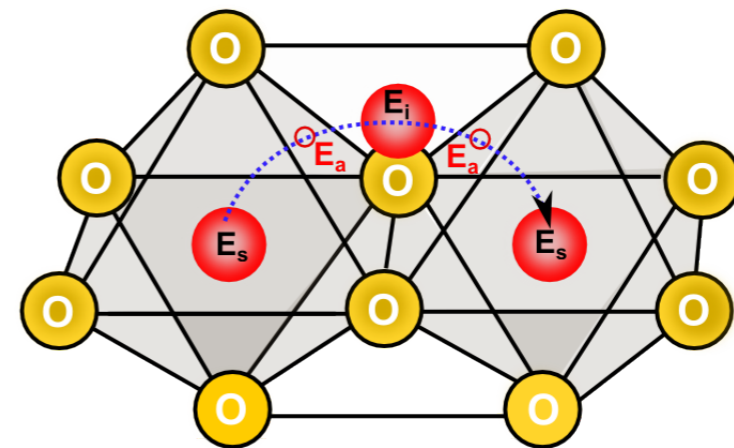
Tet - Oct - Tet

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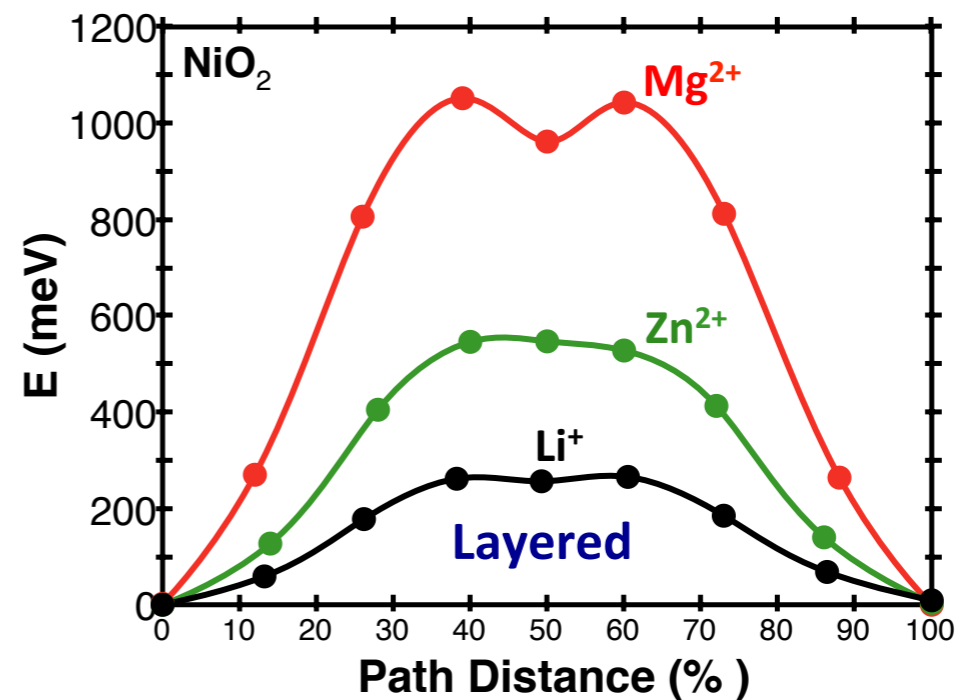
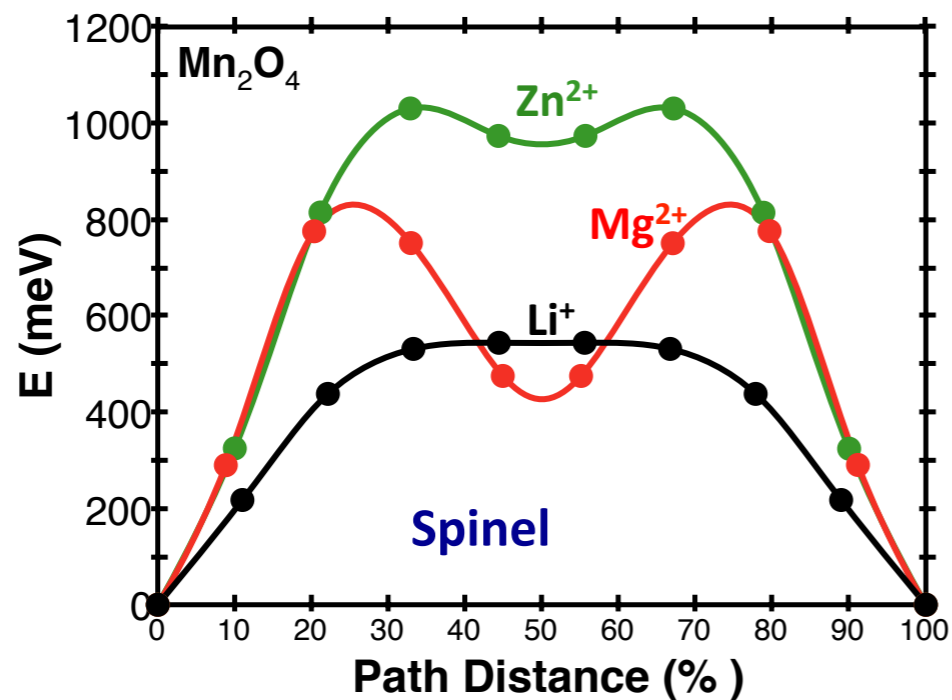


Oct - Tet - Oct

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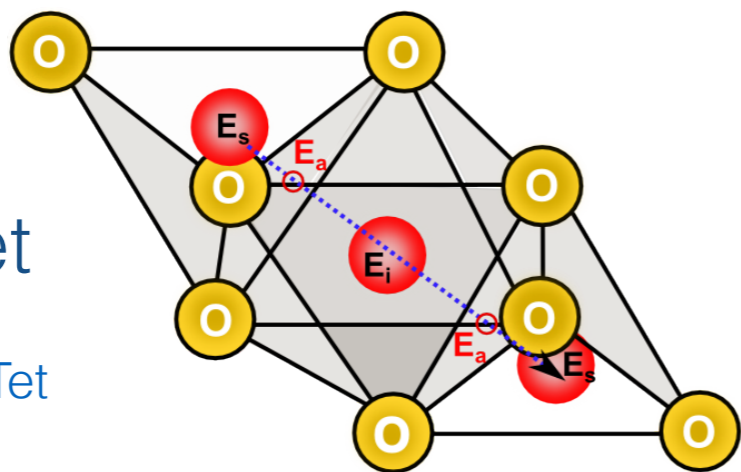


Trends differ with diffusion topology



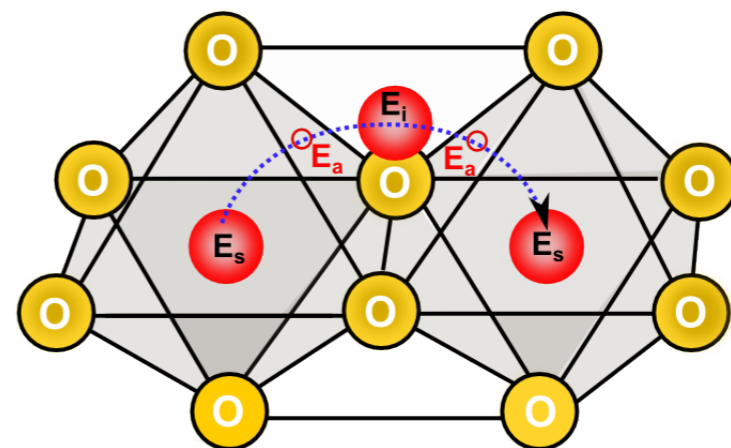
Tet - Oct - Tet

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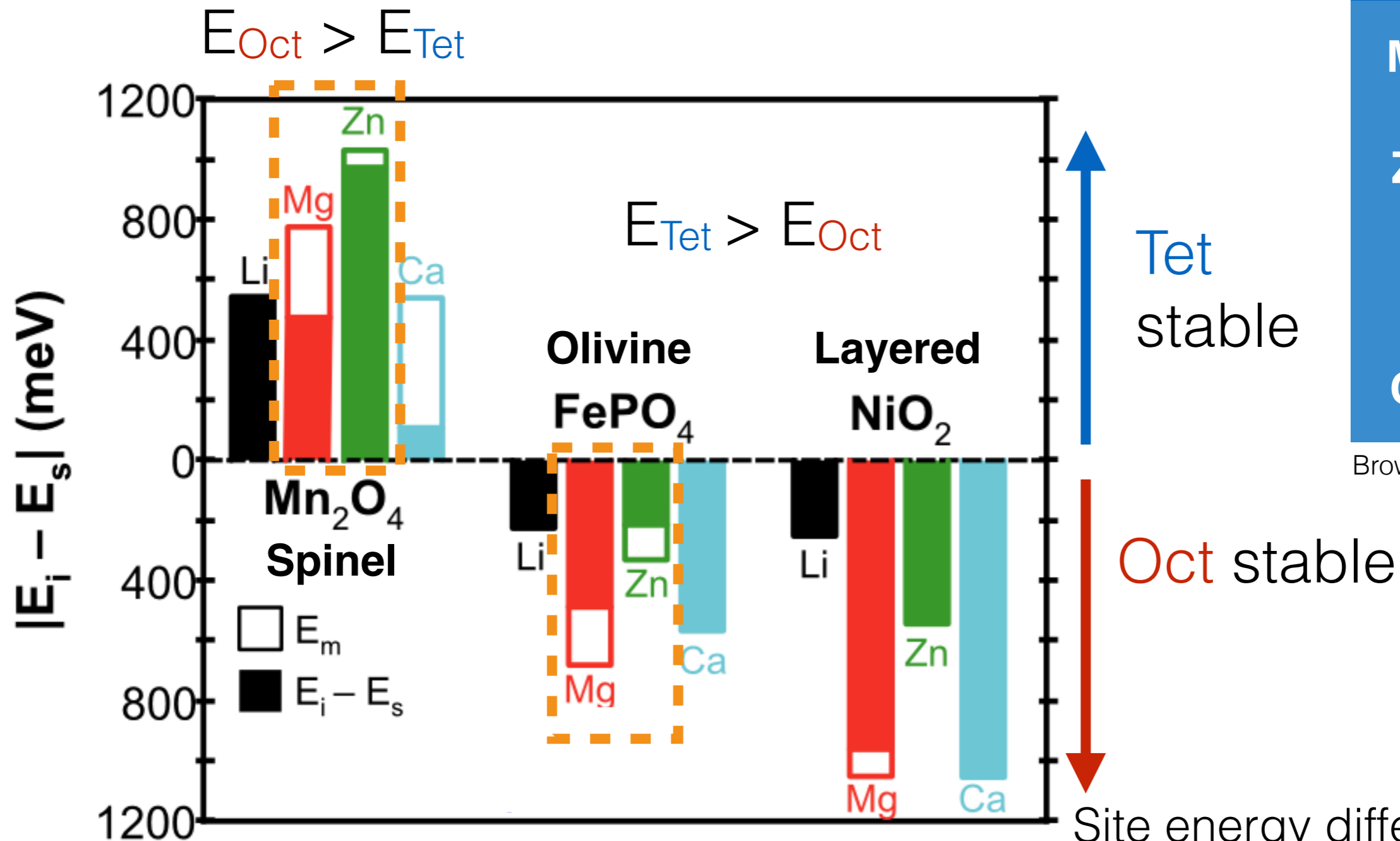
Oct - Tet - Oct

$$E_{\text{site}} = E_{\text{Tet}} - E_{\text{Oct}}$$



- General trend in E_m : $3^+ > 2^+ > 1^+$
- No universal trend in 2^+ (e.g., $\text{Mg}^{2+} > \text{Zn}^{2+}$)

Coordination of the MV ion can be used as a screening criterion



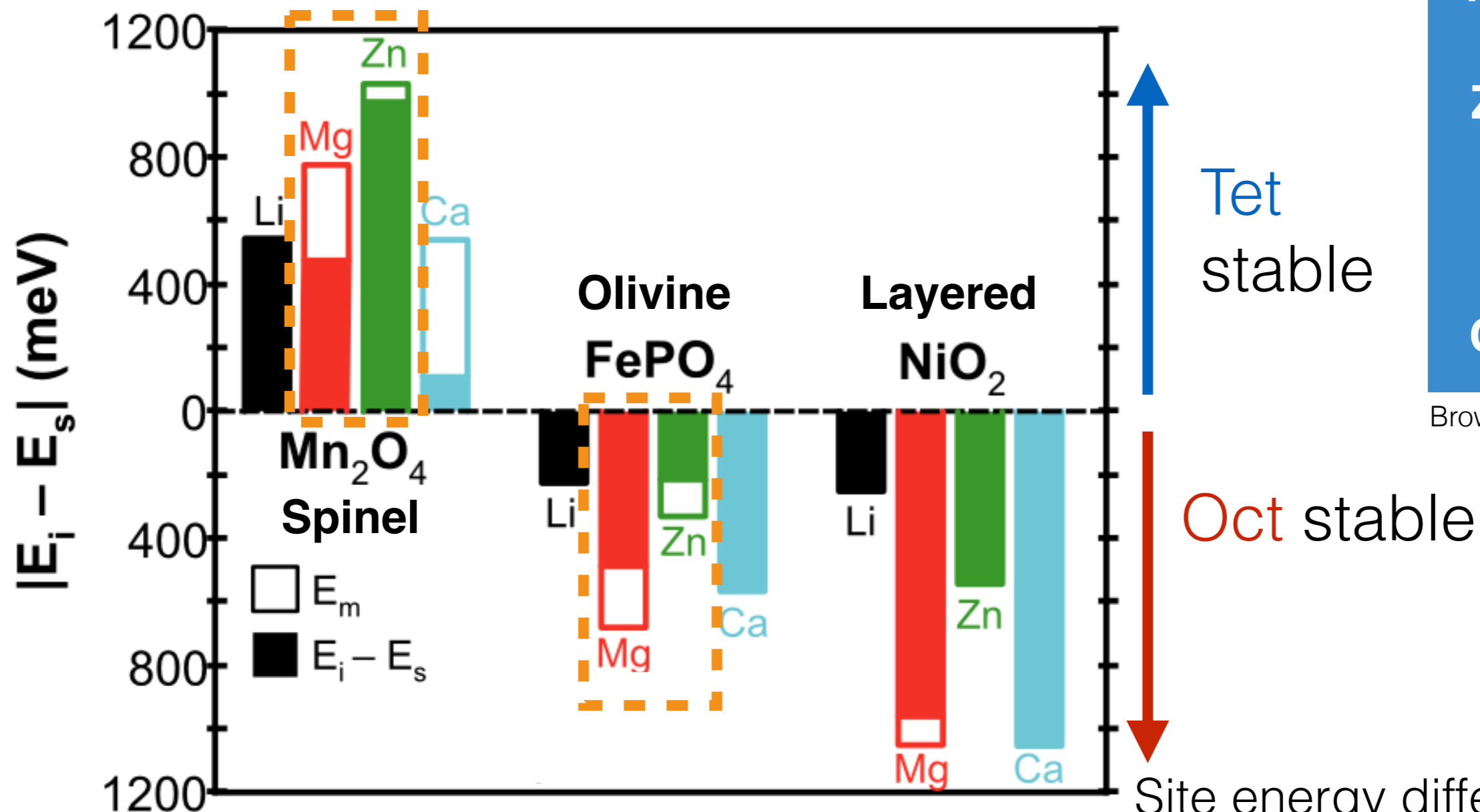
Ion	Coord. pref.
Mg	6
Zn	4
Li	4-6
Ca	6-8

Brown, Acta Cryst. (1988)

Coordination of the MV ion can be used as a screening criterion

MV ion-host pairs where MV ion resides in “**un-preferred**” coordination will have higher mobility

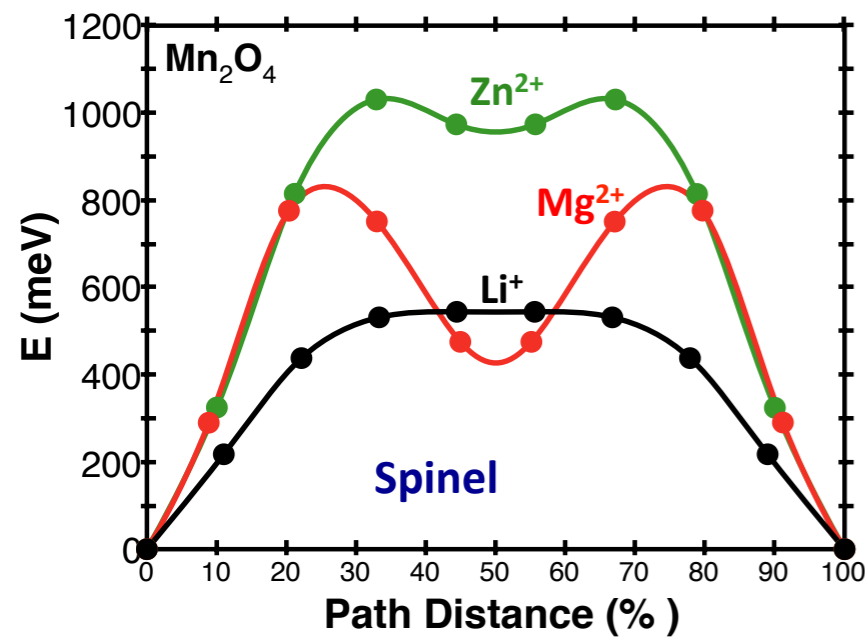
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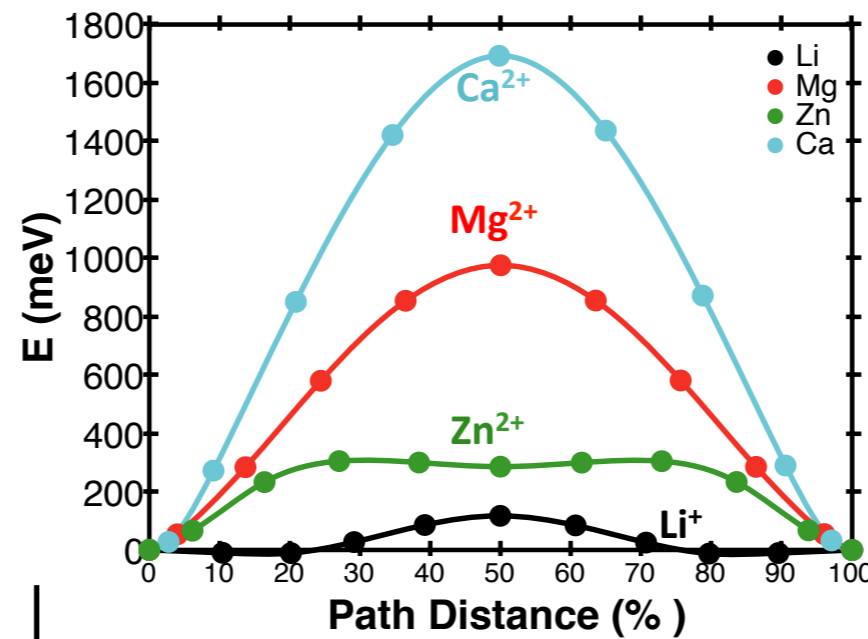
Brown, Acta Cryst. (1988)

Site energy difference is a good proxy to actual barrier

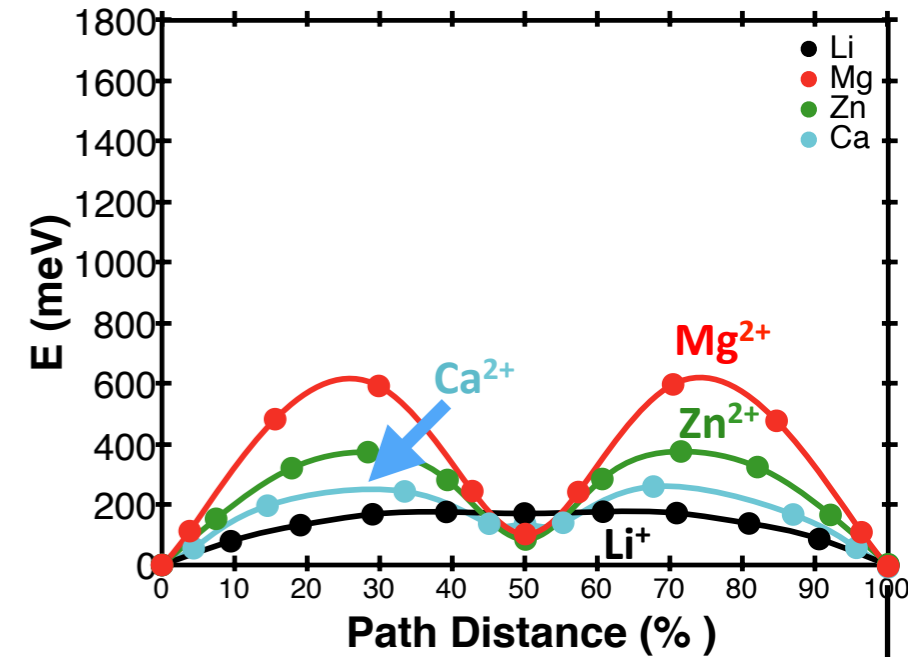
“Smaller” $\Delta(\text{Coordination})$ is also important



Spinel Mn₂O₄



α



δ

Orthorhombic V₂O₅

Coordination change:

4-6-4

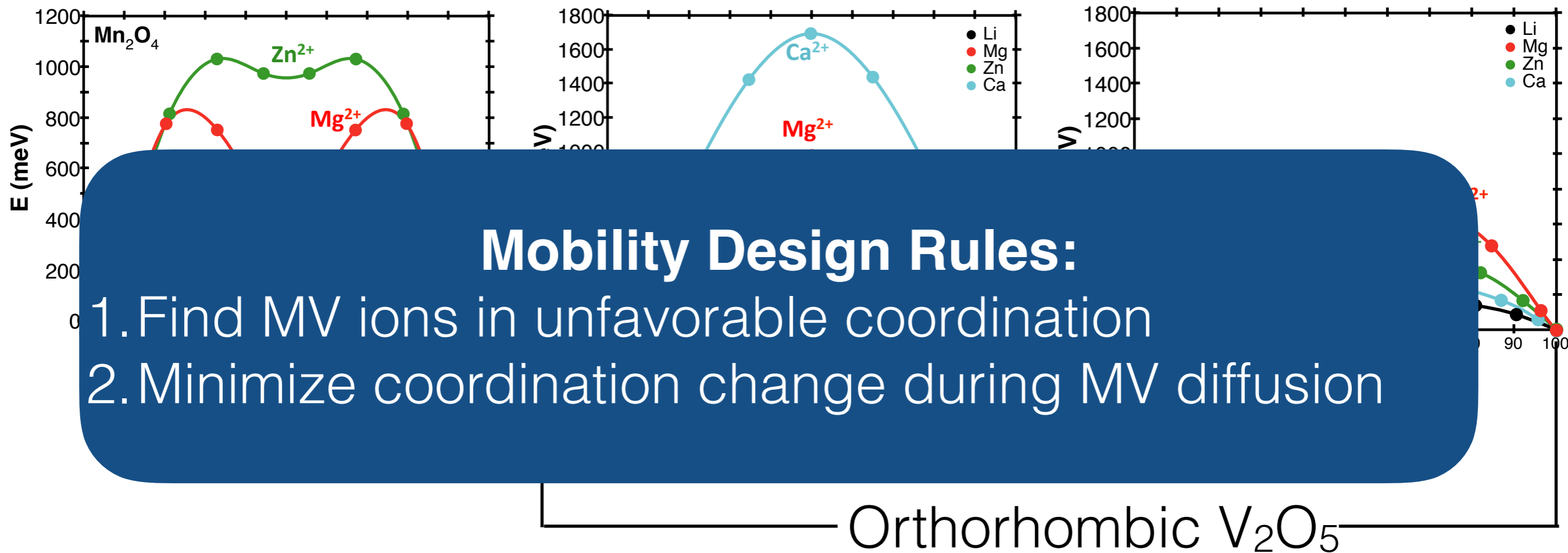
8-3-8

6-5-6

Host structures with “**smaller**” coordination change will have better MV mobility

More on spinels shortly...

“Smaller” $\Delta(\text{Coordination})$ is also important



Coordination change:

4-6-4

8-3-8

6-5-6

Host structures with “**smaller**” coordination change will have better MV mobility

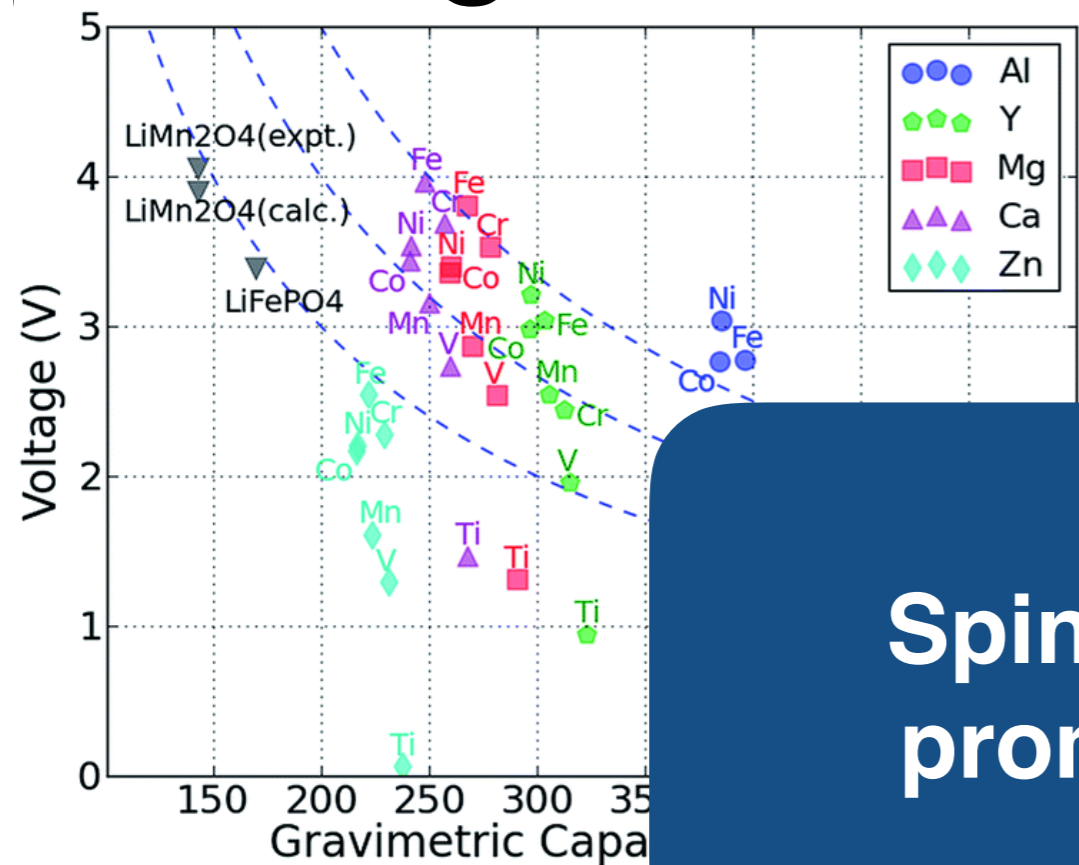
More on spinels shortly...

Spinel in MV batteries

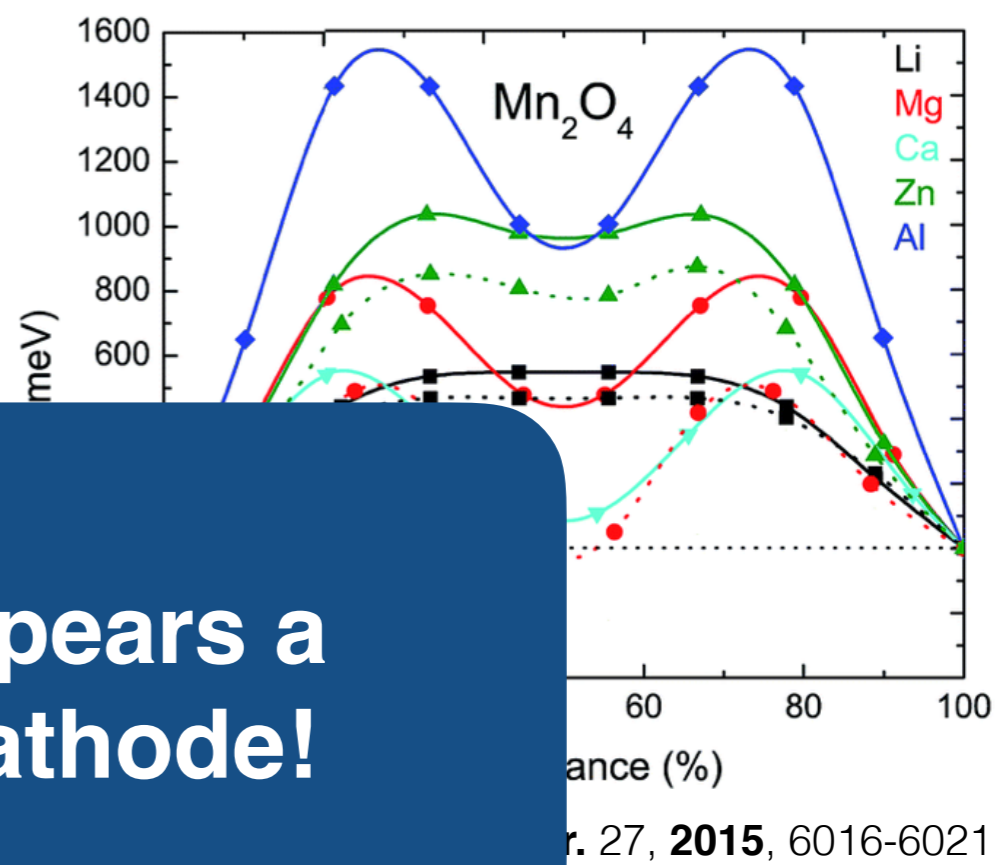
Inversion in MgMn_2O_4

G.S. Gautam *et al.*, “Influence of spinel inversion on Mg mobility and electrochemistry in spinels”, under review

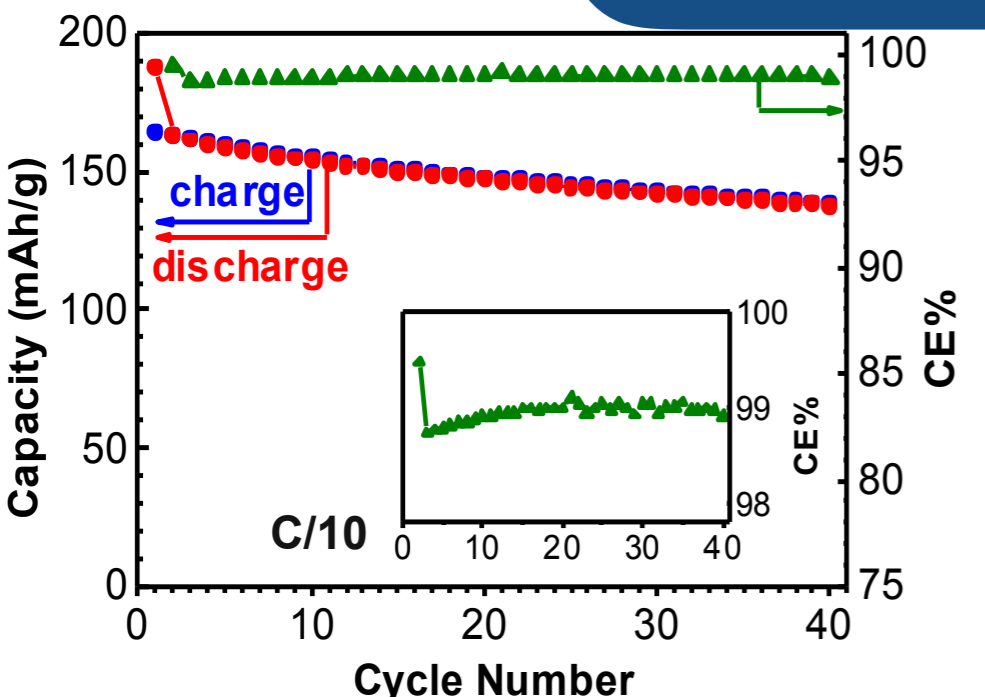
Spinel have promising energy densities and Mg^{2+} mobility



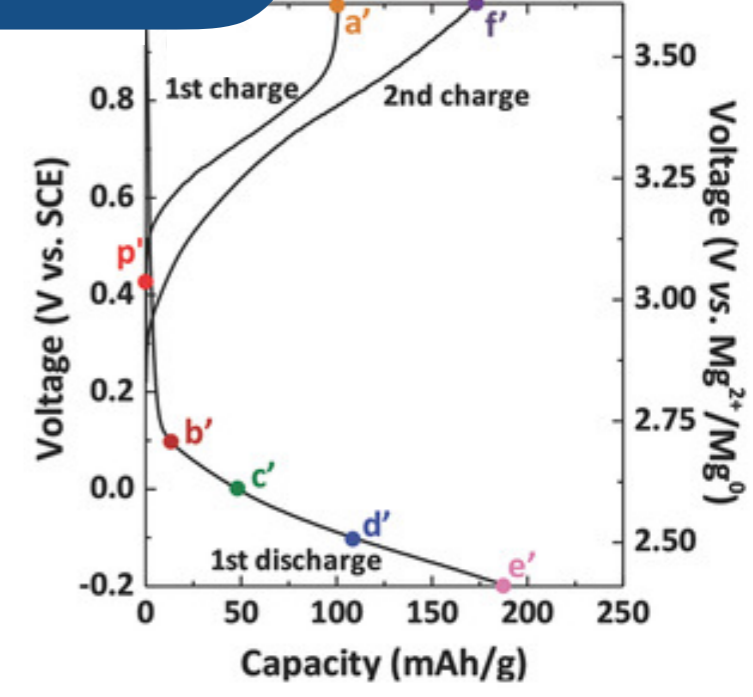
Spinel- Mn_2O_4 appears a promising Mg cathode!



Liu *et al.*, **Energy Environ. Sci.** 9, 2016, 2273-2277



Sun *et al.*, **Energy Environ. Sci.** 9, 2016, 2273-2277



Kim *et al.*, **Adv. Mater.** 27, 2015, 3377-3384

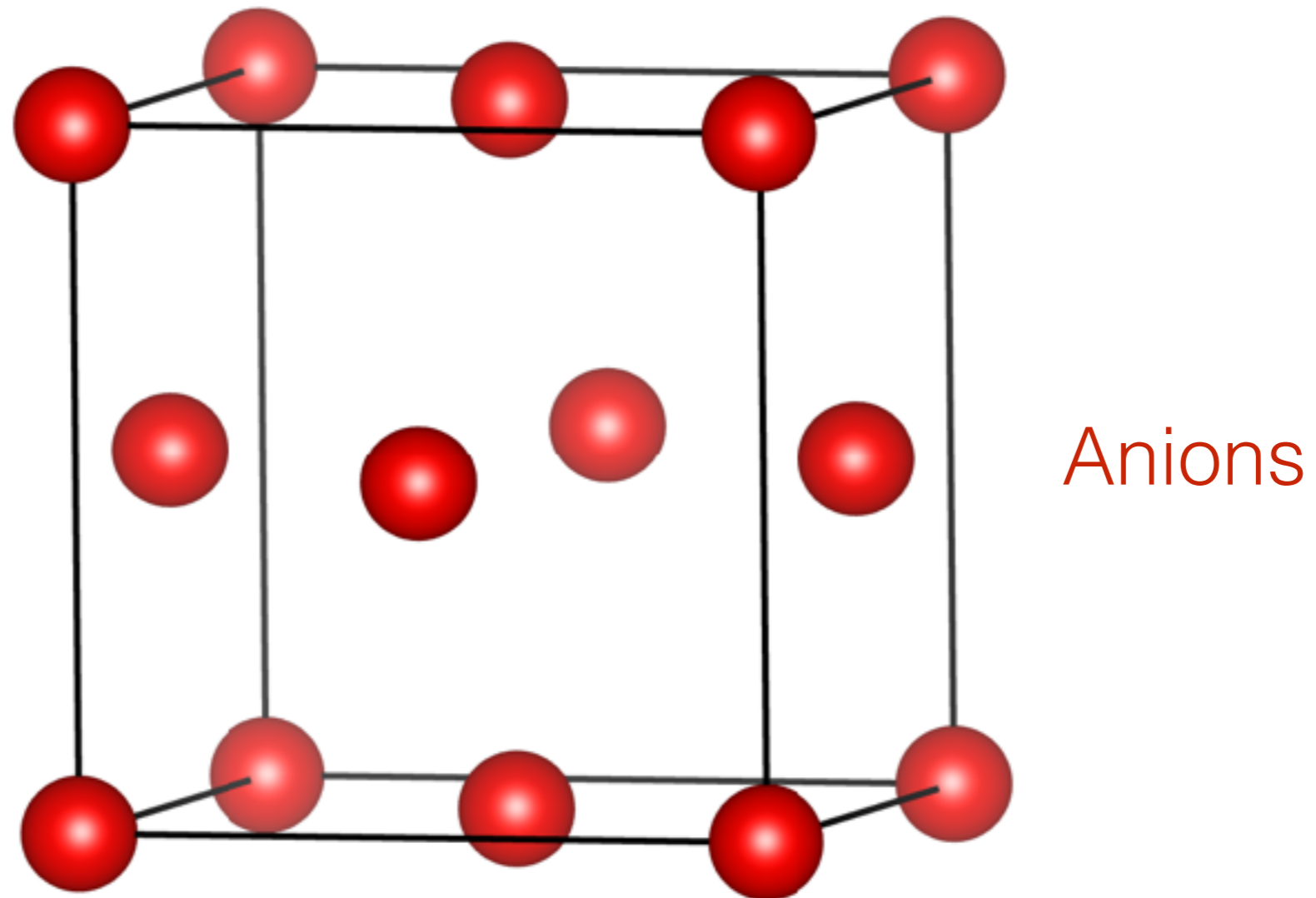
Normal spinel structure: AM_2X_4

A = Mg

M = Mn

X = O

Vac = Vacancy



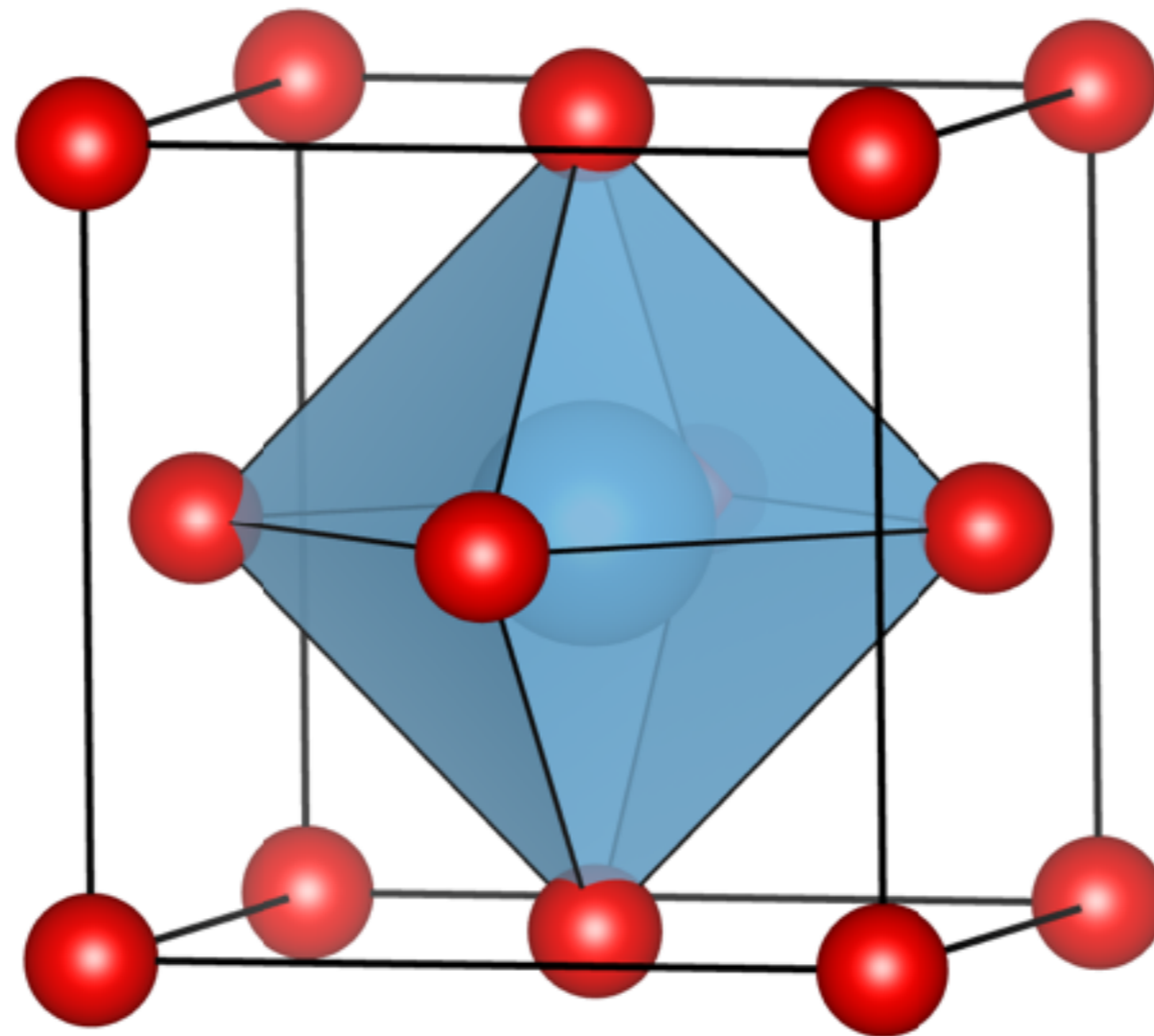
Normal spinel structure: AM_2X_4

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= (# Anions)

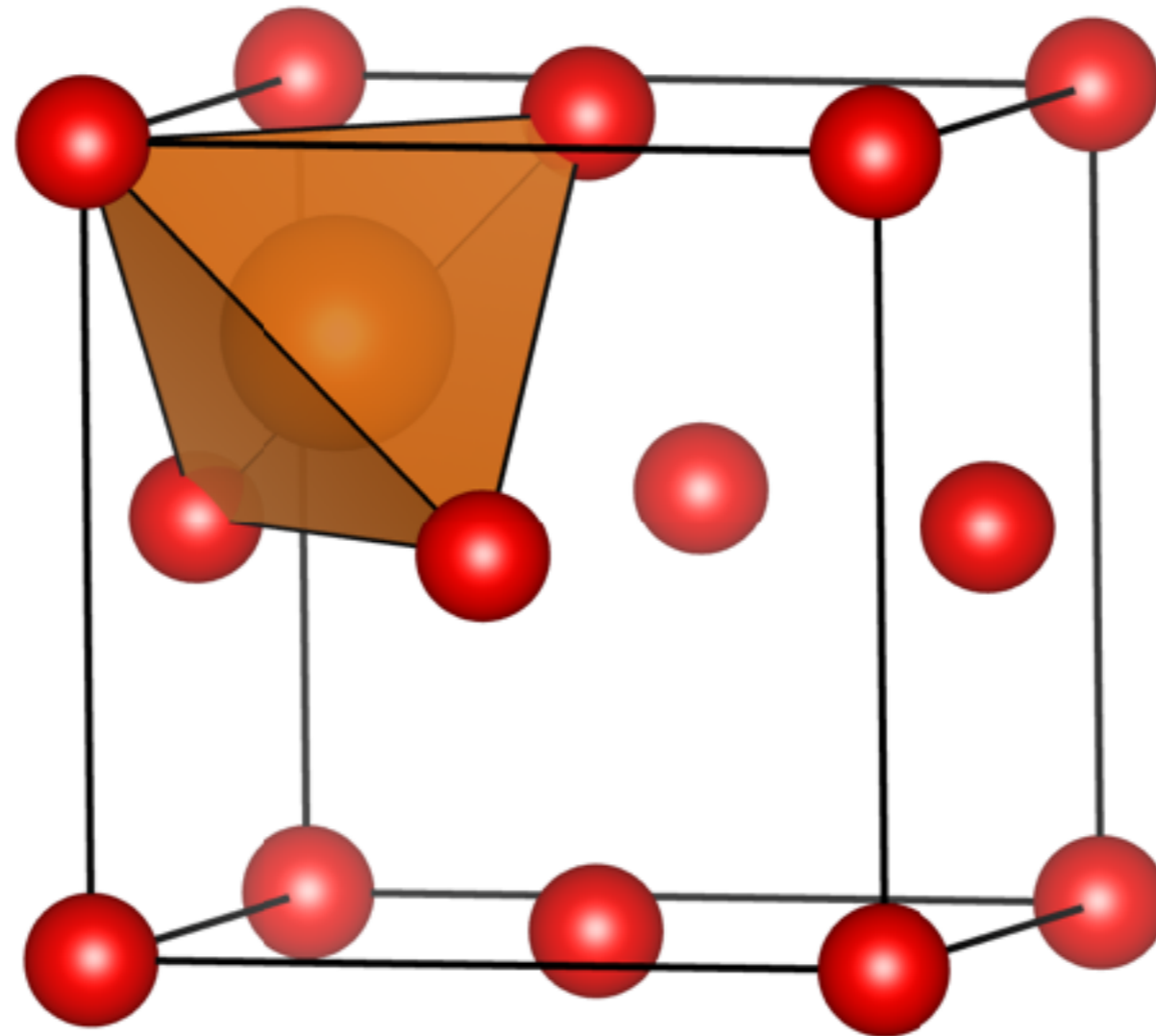
Normal spinel structure: AM_2X_4

A = Mg

M = Mn

X = O

Vac = Vacancy



= 2*(# Anions)

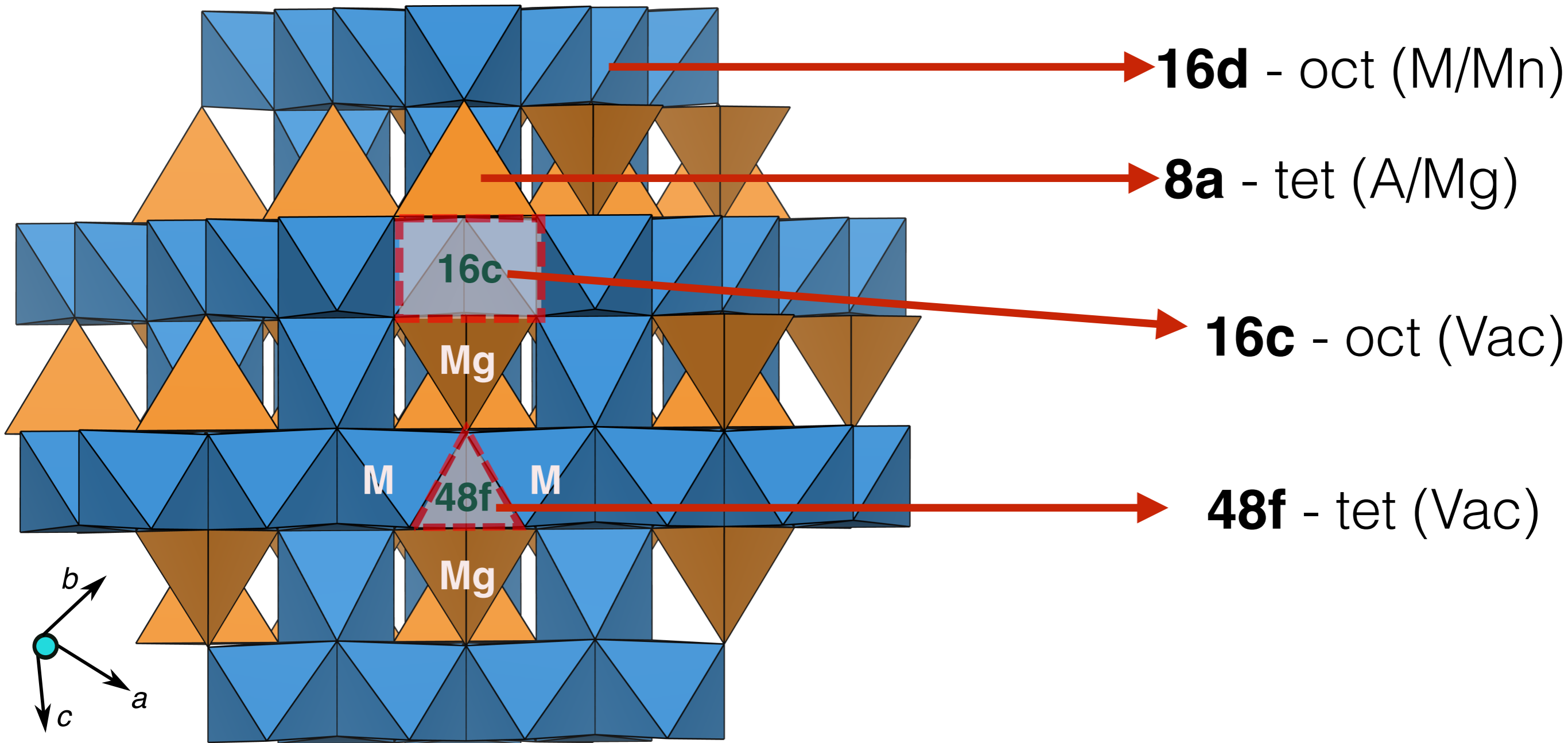
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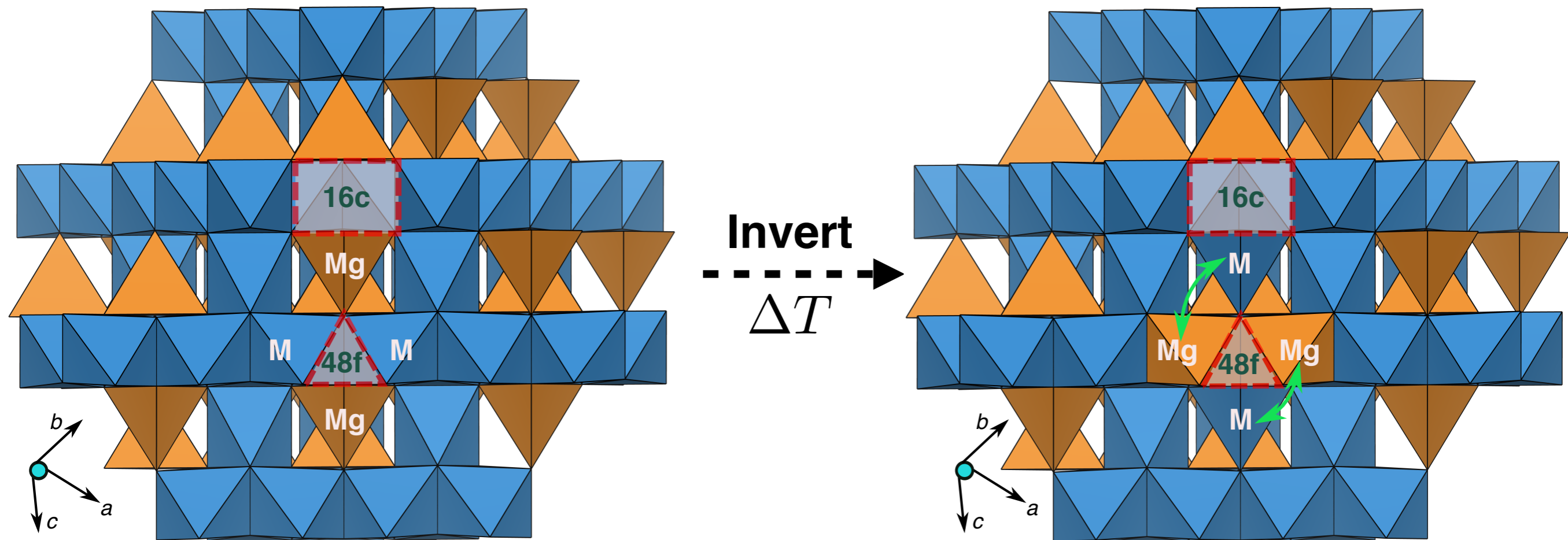
X = O

Vac = Vacancy



MgMn₂O₄ spinel afflicted by inversion

Inversion: Exchange of A and M sites in AM₂X₄



Degree of inversion (*i*): Fraction of 8a (tet) sites occupied by M

$$i \sim 0.2 - 0.6^{6,7} \text{ in MgMn}_2\text{O}_4$$

6. Irani *et al.*, **J. Phys. Chem. Solids** 23, 1962, 711-727

7. Malavasi *et al.*, **J. Solid State Chem.** 166, 2002, 171-176

MgMn₂O₄ spinel afflicted by inversion

Inversion: Exchange of A and M sites in AM₂X₄



How does inversion affect Mg mobility?

Degree of inversion (i): Fraction of 8a (tet) sites occupied by M

$$i \sim 0.2 - 0.6^{6,7} \text{ in MgMn}_2\text{O}_4$$

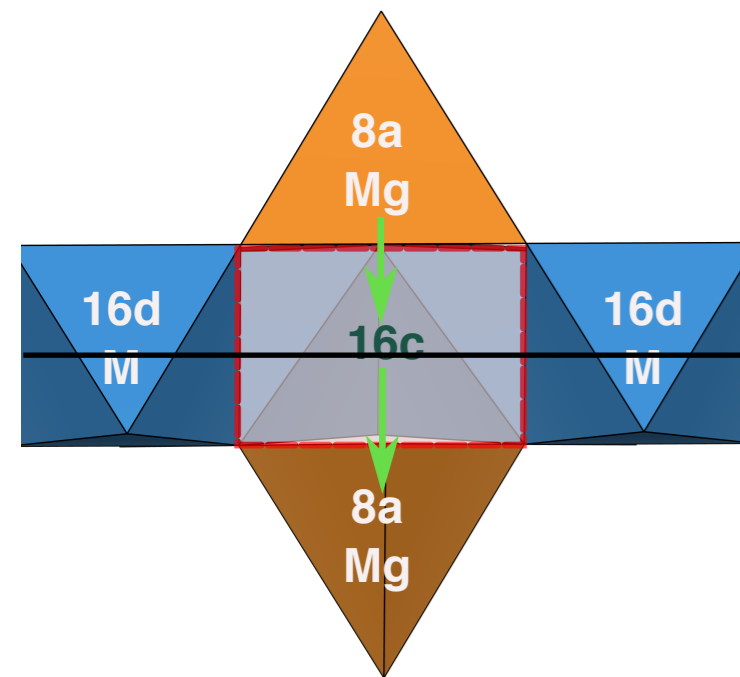
6. Irani *et al.*, **J. Phys. Chem. Solids** 23, **1962**, 711-727

7. Malavasi *et al.*, **J. Solid State Chem.** 166, **2002**, 171-176

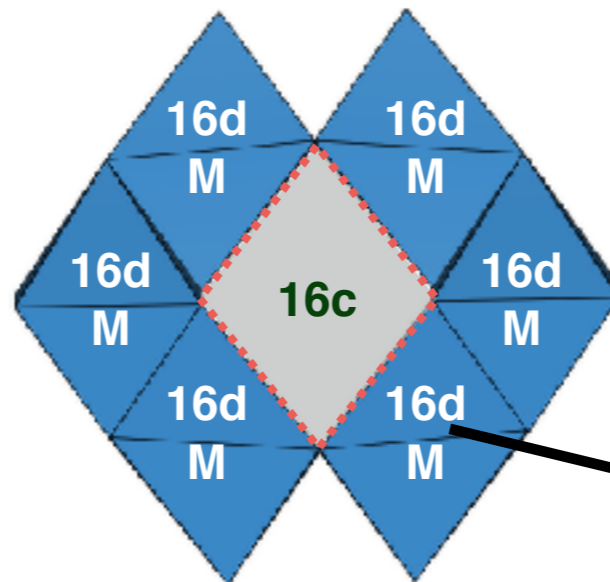
Inversion: distinct local cation decorations

tet \rightarrow tet hops

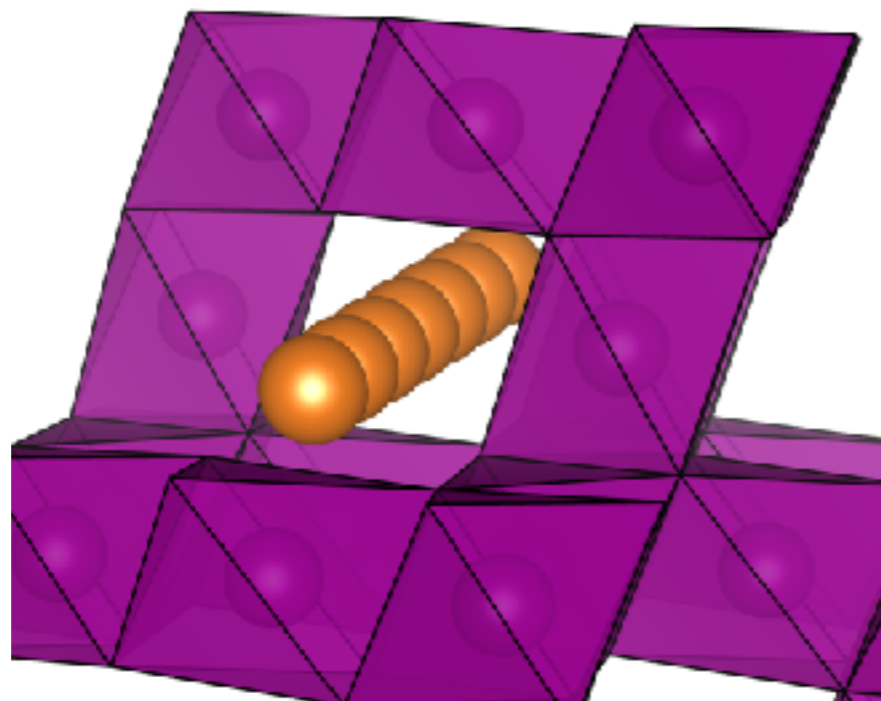
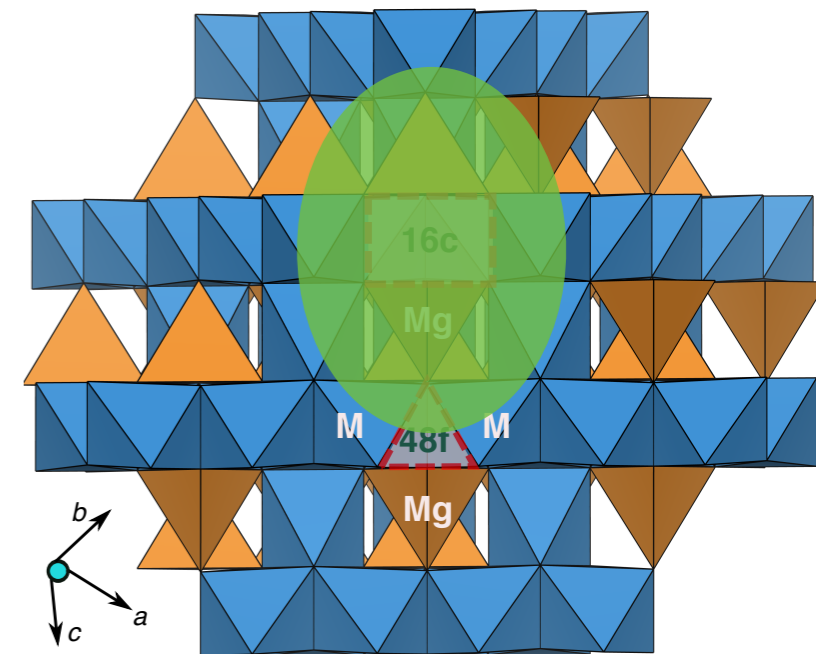
Hop 1



tet
↓
oct
↓
tet



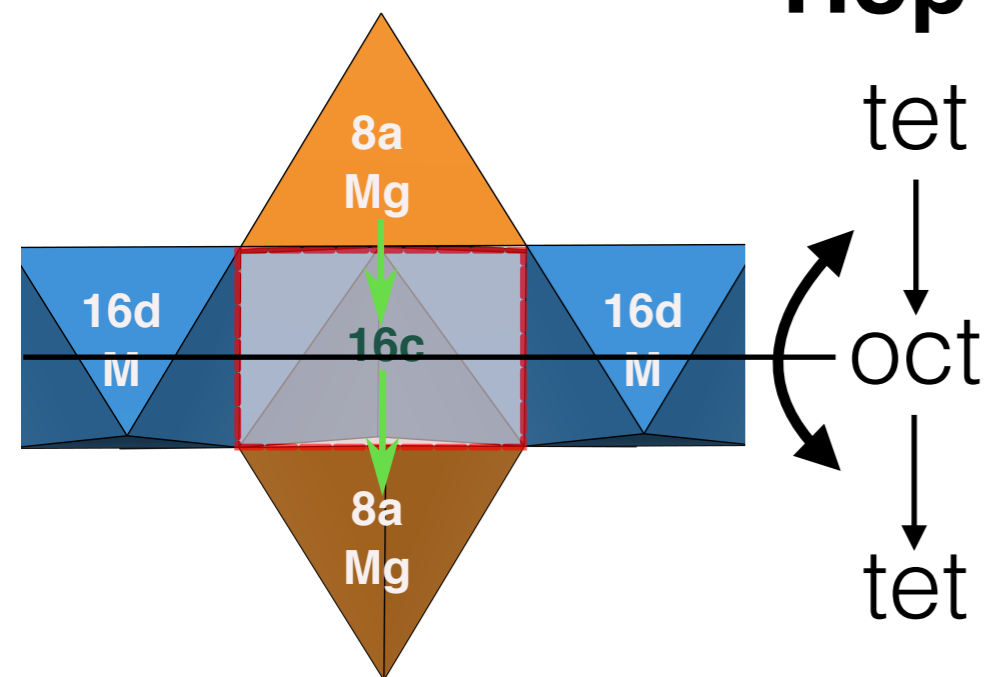
16d "ring" sites



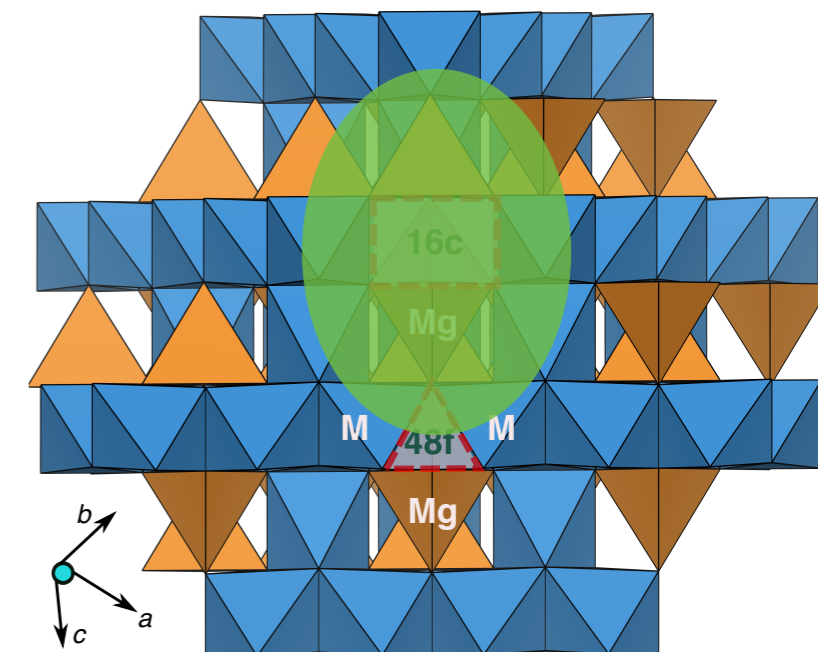
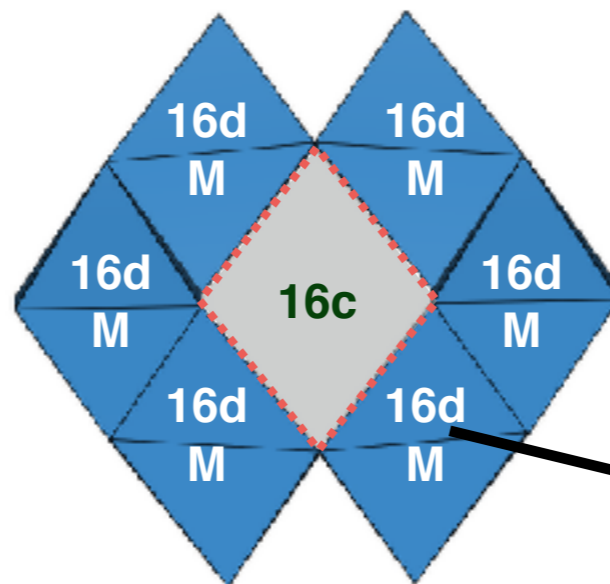
Inversion: distinct local cation decorations

tet \rightarrow tet hops

Hop 1

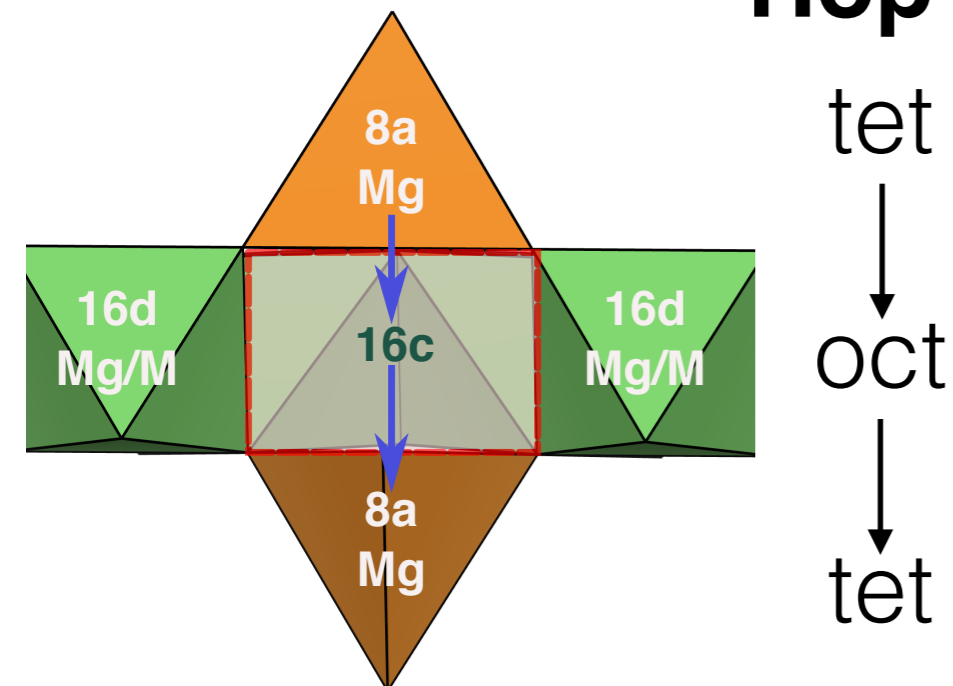


tet
oct
tet

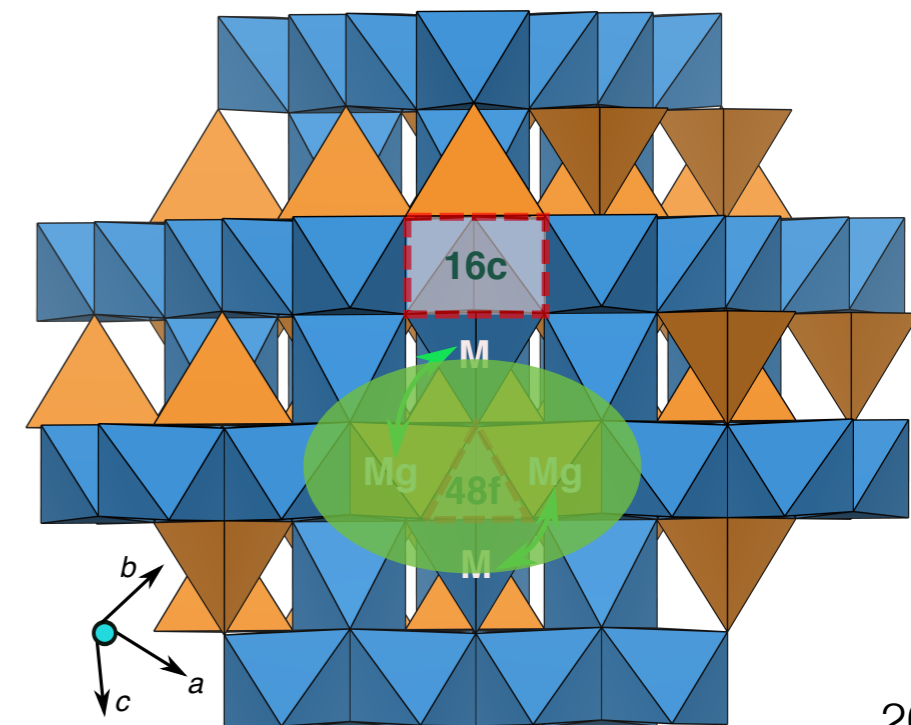
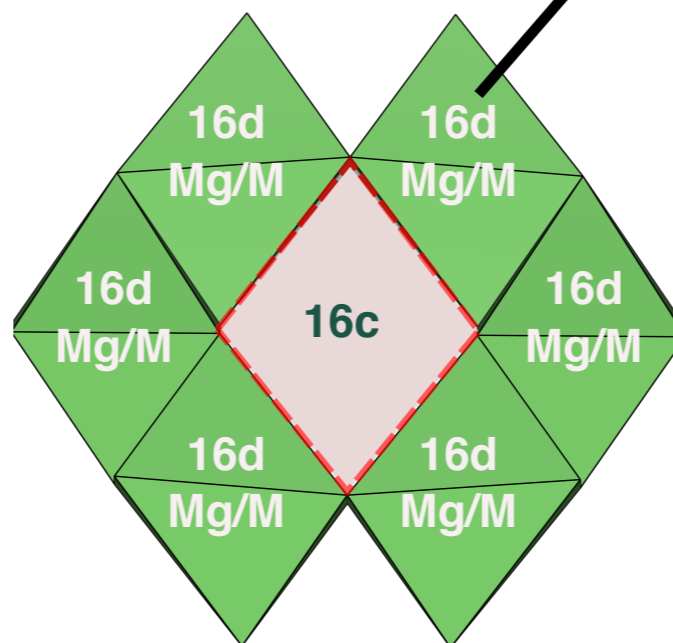


16d "ring" sites

Hop 2



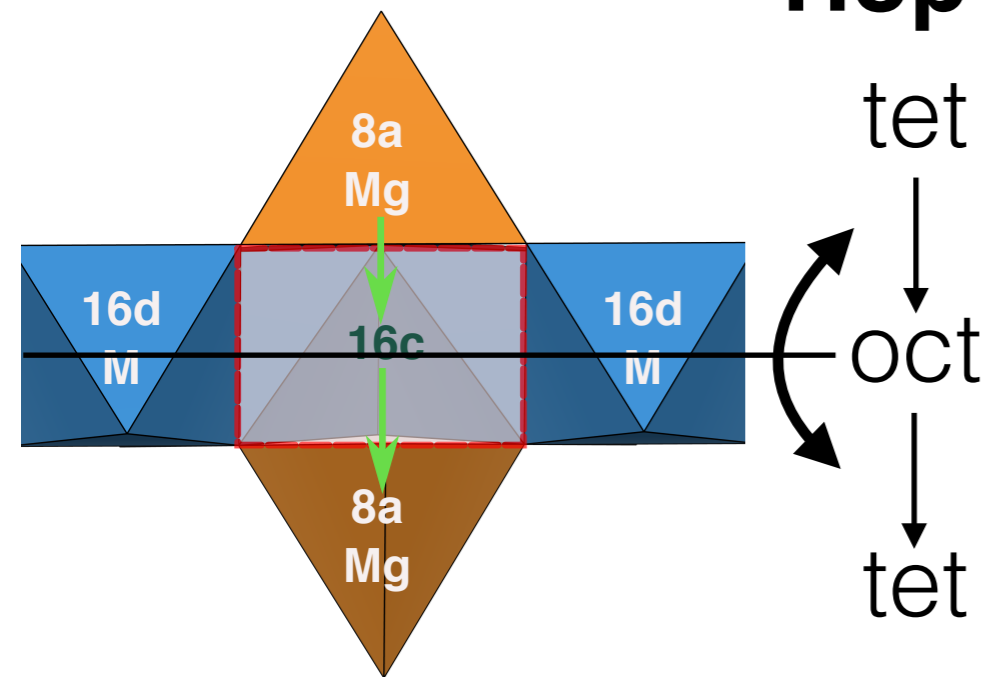
tet
oct
tet



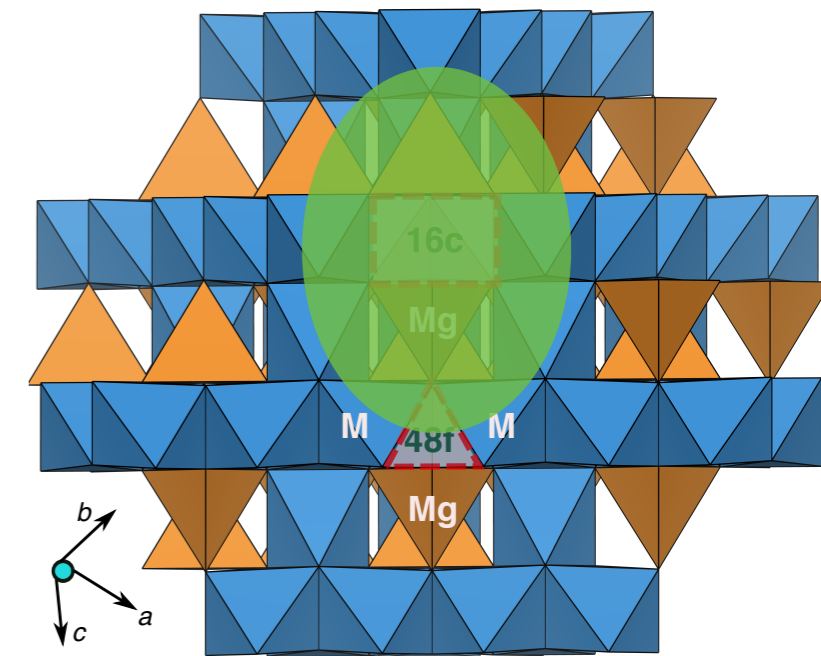
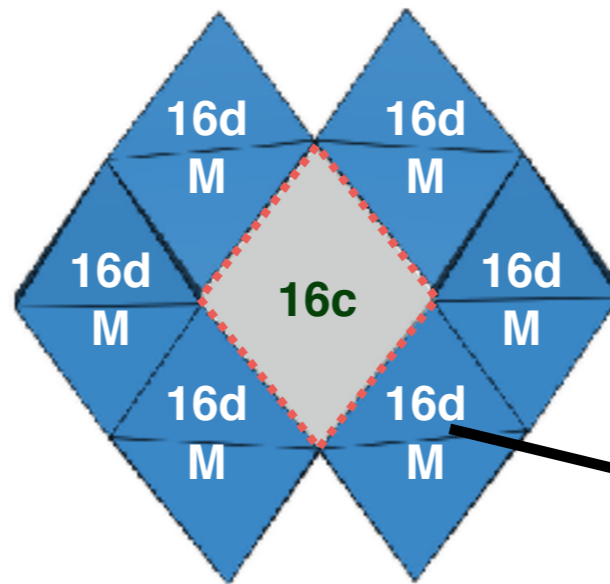
Inversion: distinct local cation decorations

tet \rightarrow tet hops

Hop 1

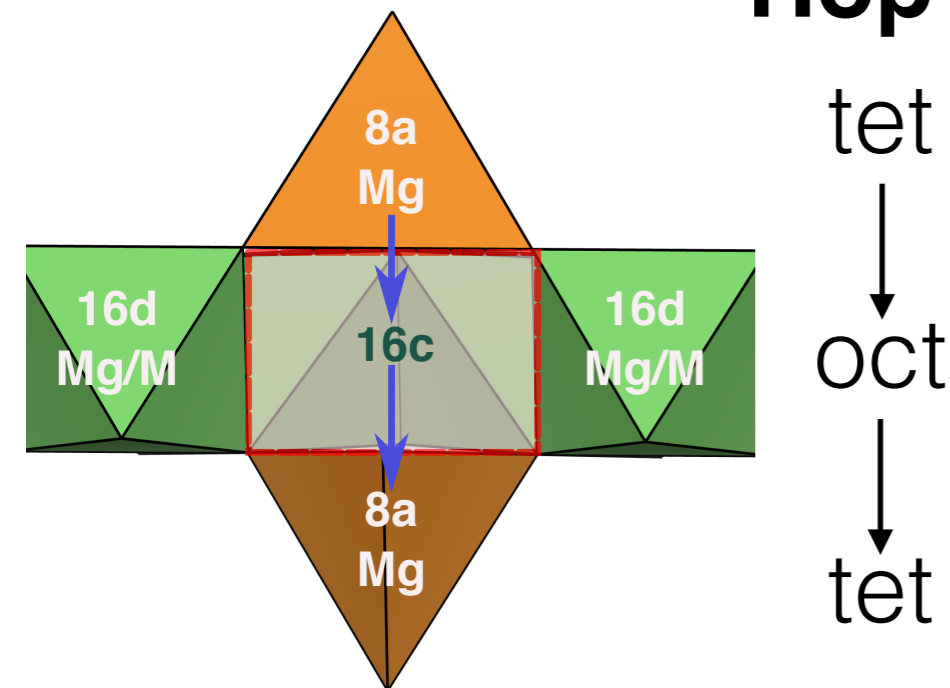


tet
↓
oct
↓
tet

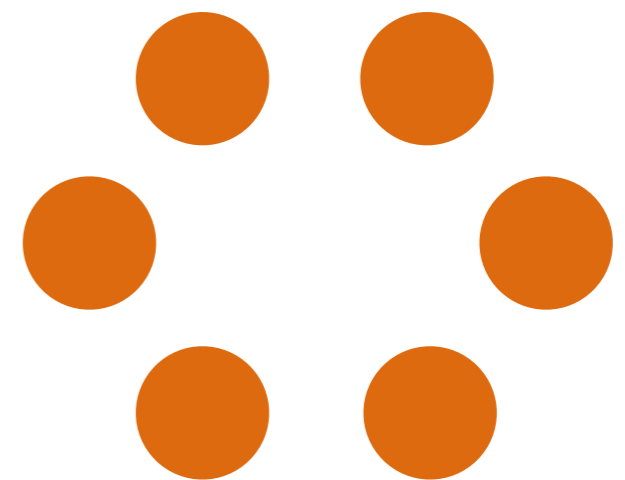
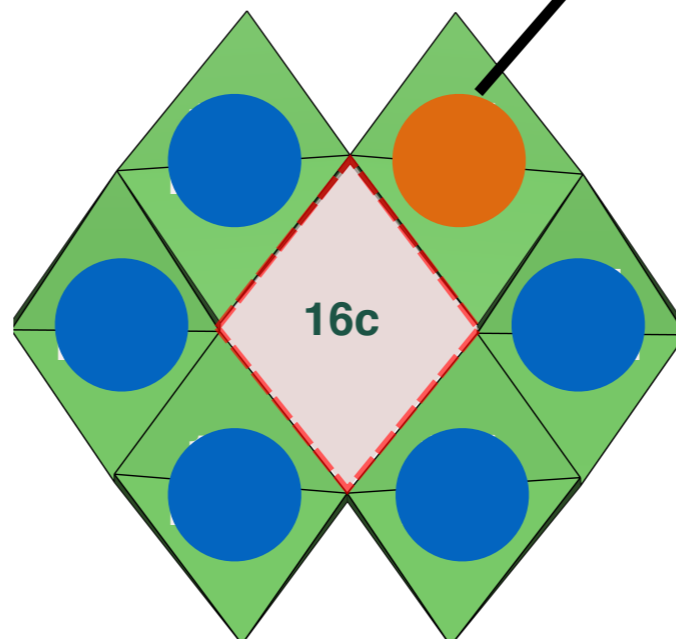


16d "ring" sites

Hop 2

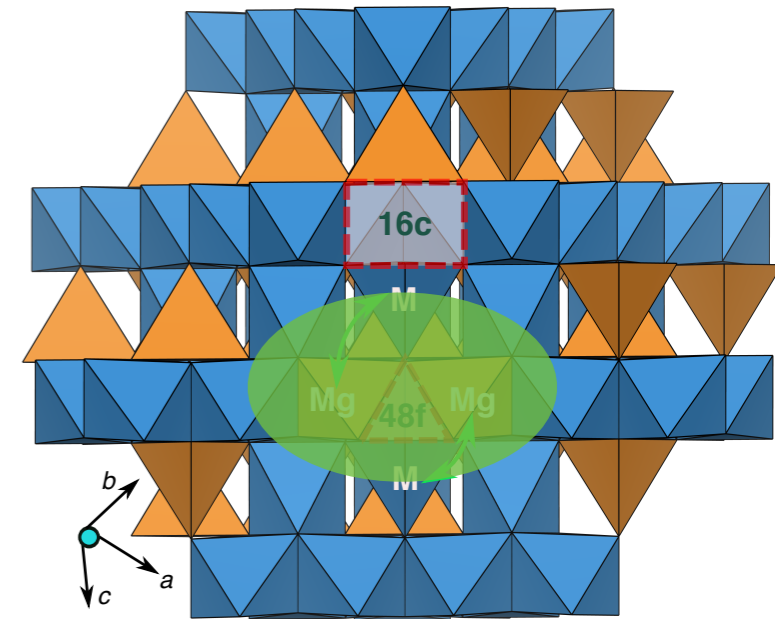


tet
↓
oct
↓
tet



Inversion: distinct local cation decorations

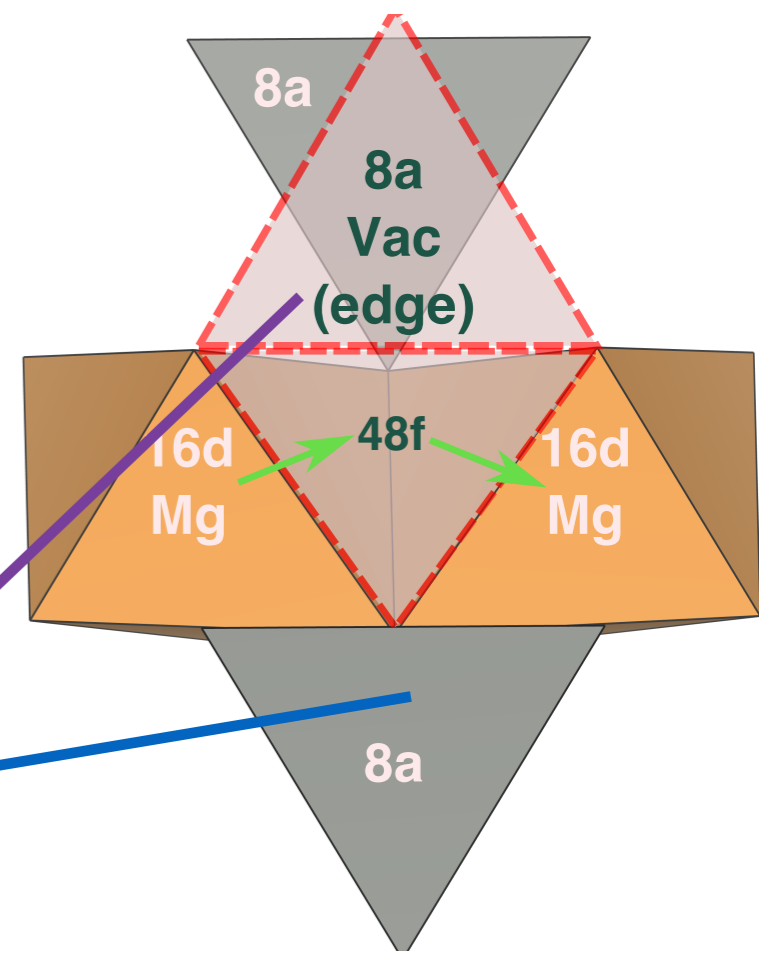
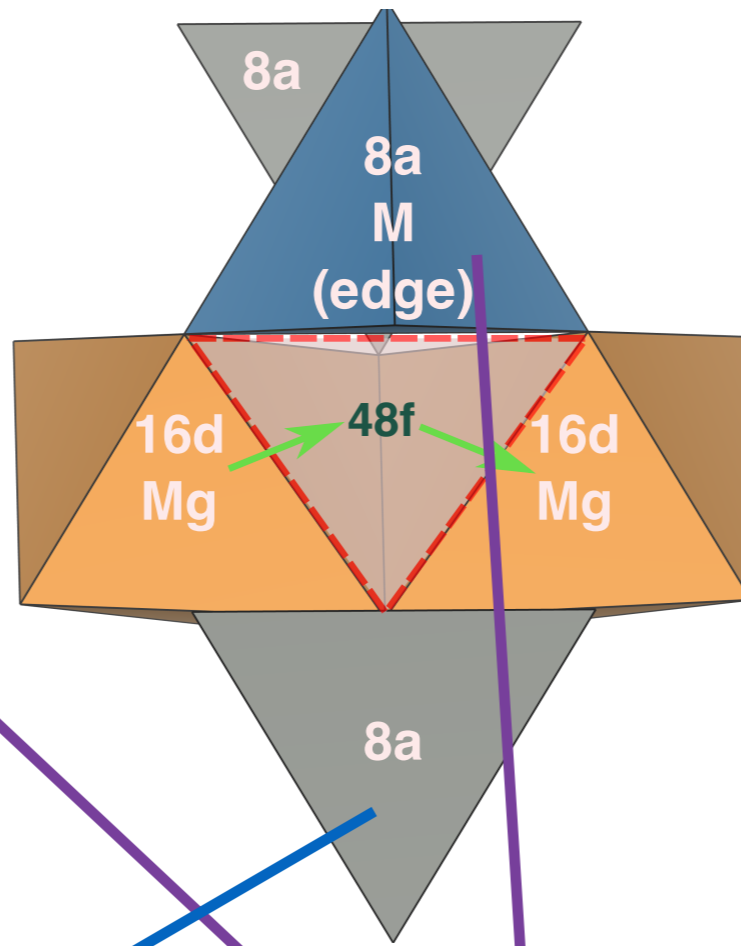
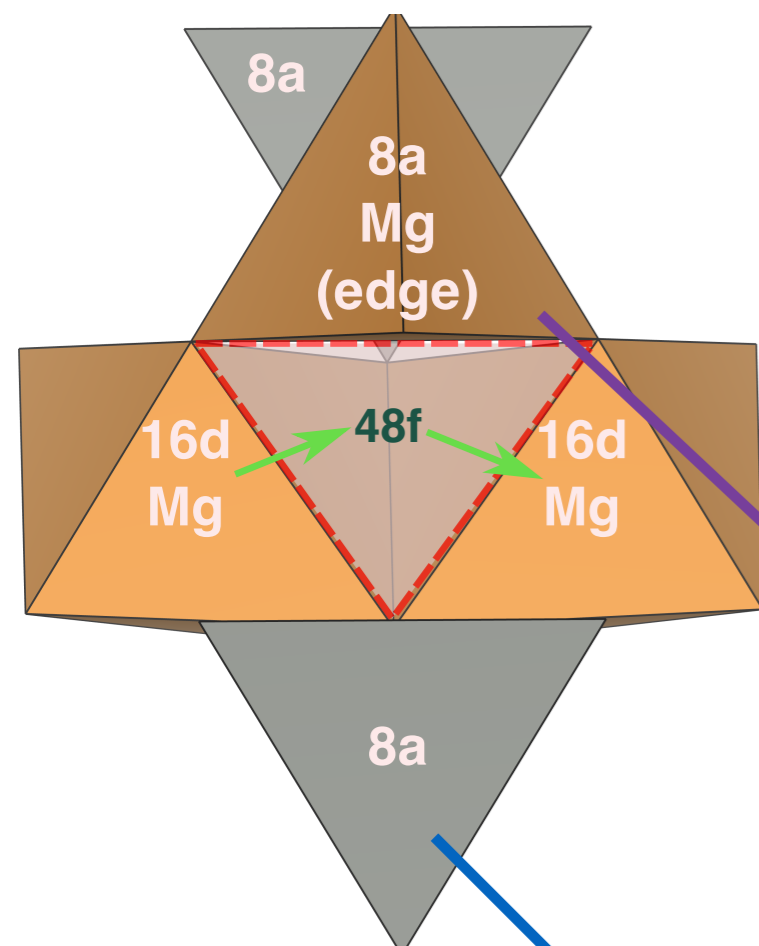
oct \rightarrow oct hops



Hop 3

Hop 4

Hop 5

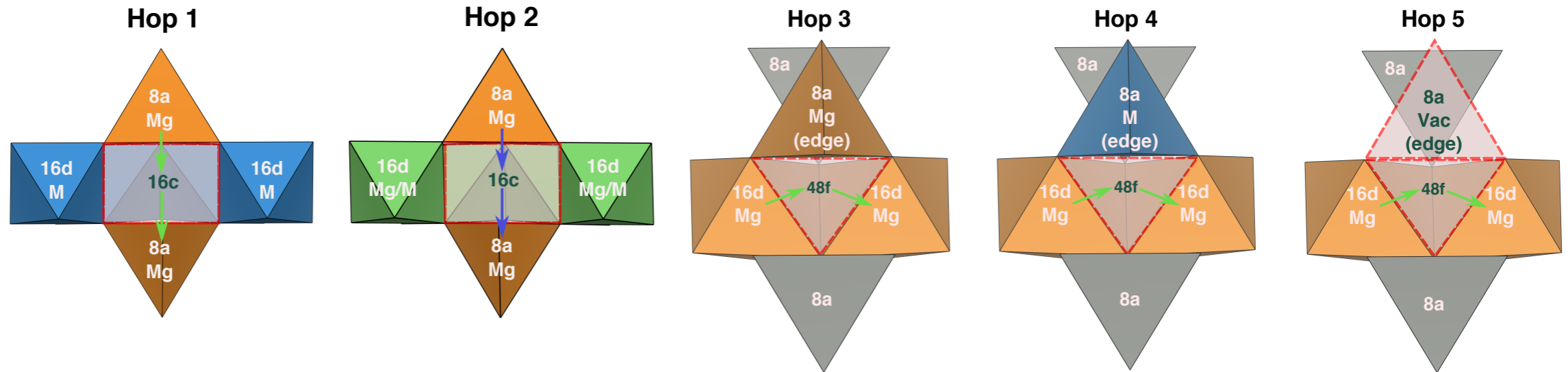


“Corner”-8a

“Edge”-8a

oct \rightarrow tet \rightarrow oct

Notations Summary



Topology	tet-oct-tet	tet-oct-tet	oct-tet-oct	oct-tet-oct	oct-tet-oct
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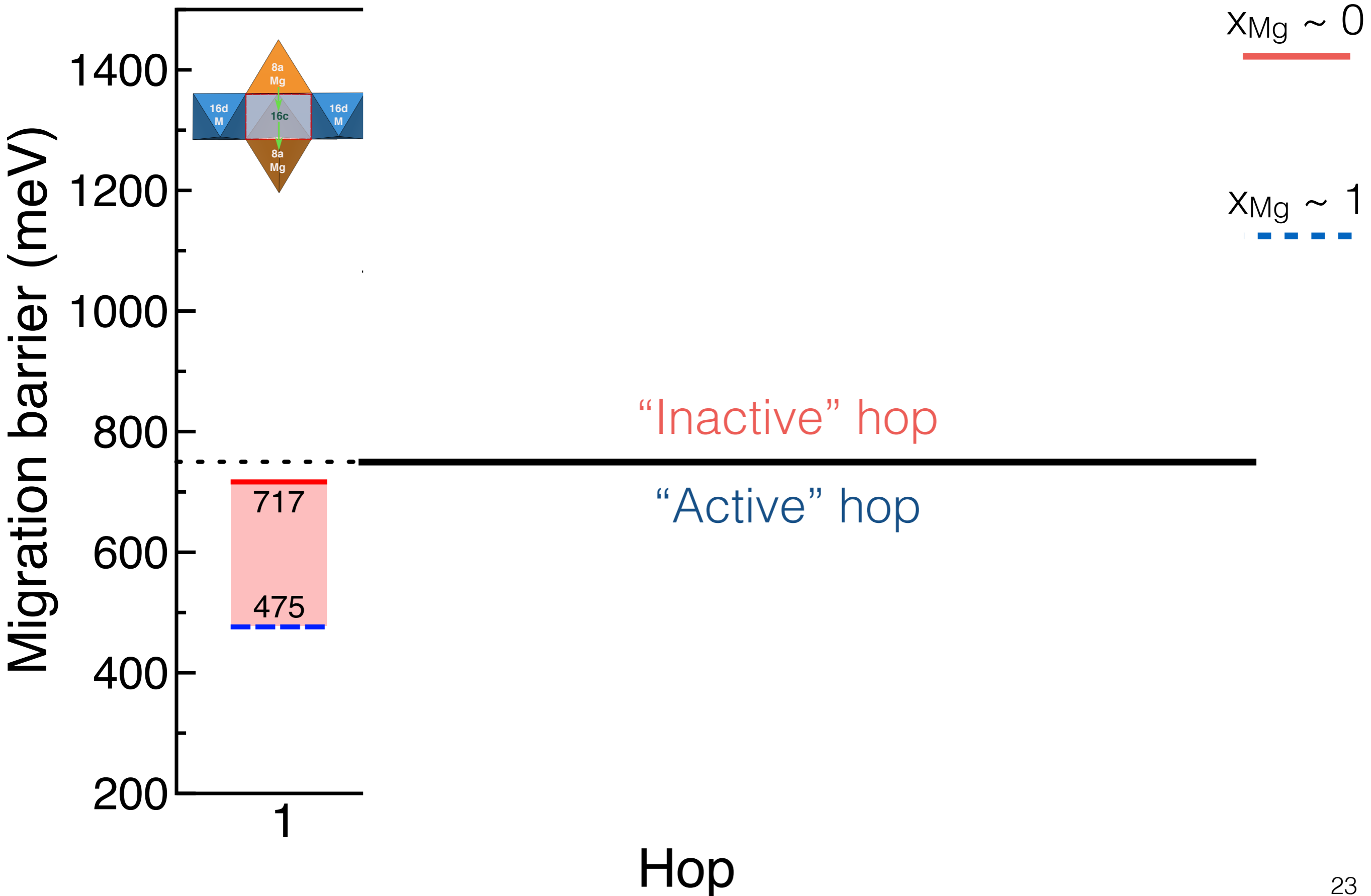
Stable site	8a	8a	16d	16d	16d
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Intermediate site	16c	16c	48f	48f	48f
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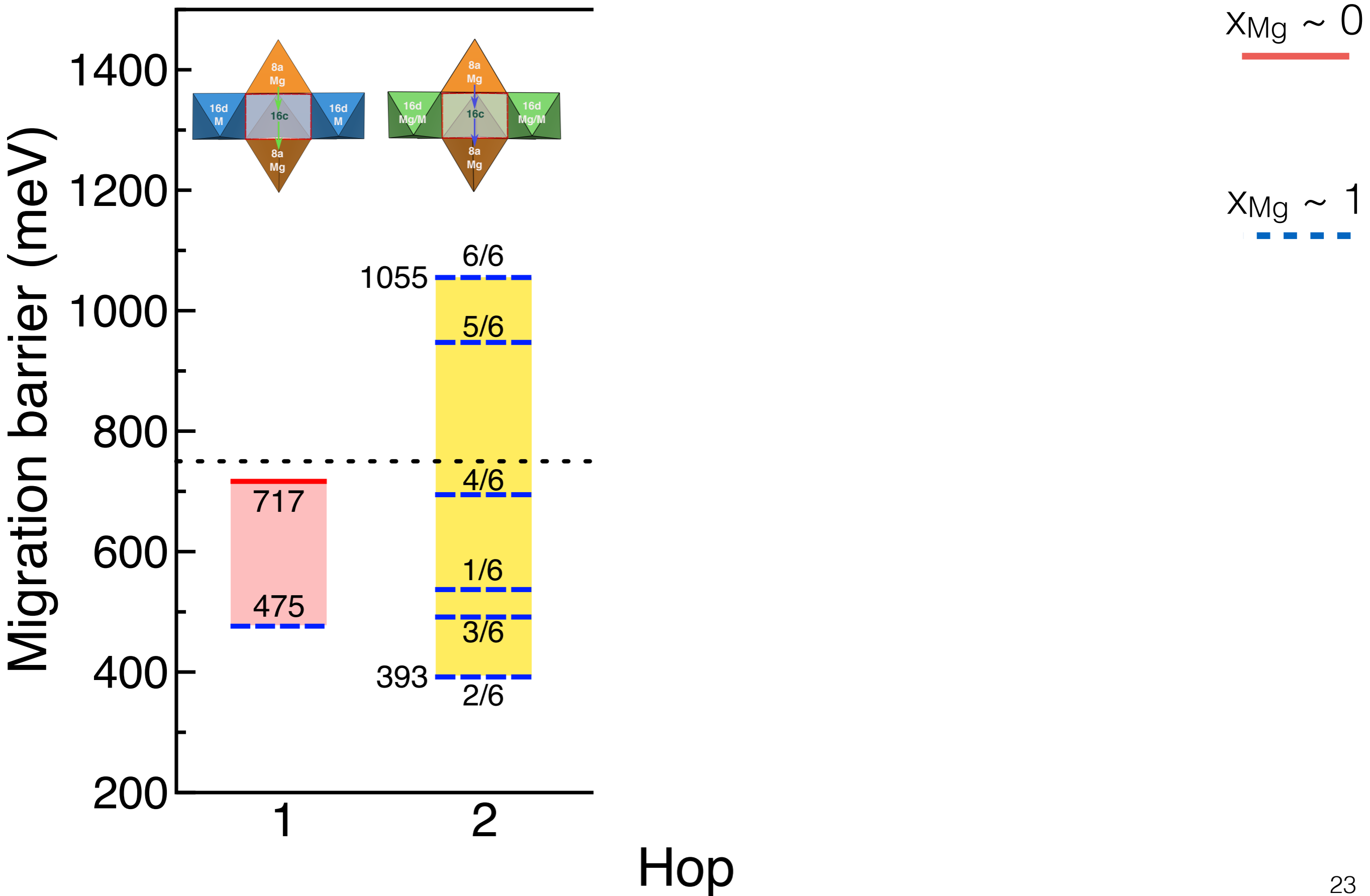
Edge neighbors	“Ring” 16d (Mn)	“Ring” 16d (Mg/Mn)	“Edge” 8a (Mg)	“Edge” 8a (Mn)	“Edge” 8a (Vac)
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Corner neighbors	—	—	“Corner” 8a (Mg/Mn/Vac)	“Corner” 8a (Mg/Mn/Vac)	“Corner” 8a (Mg/Mn/Vac)
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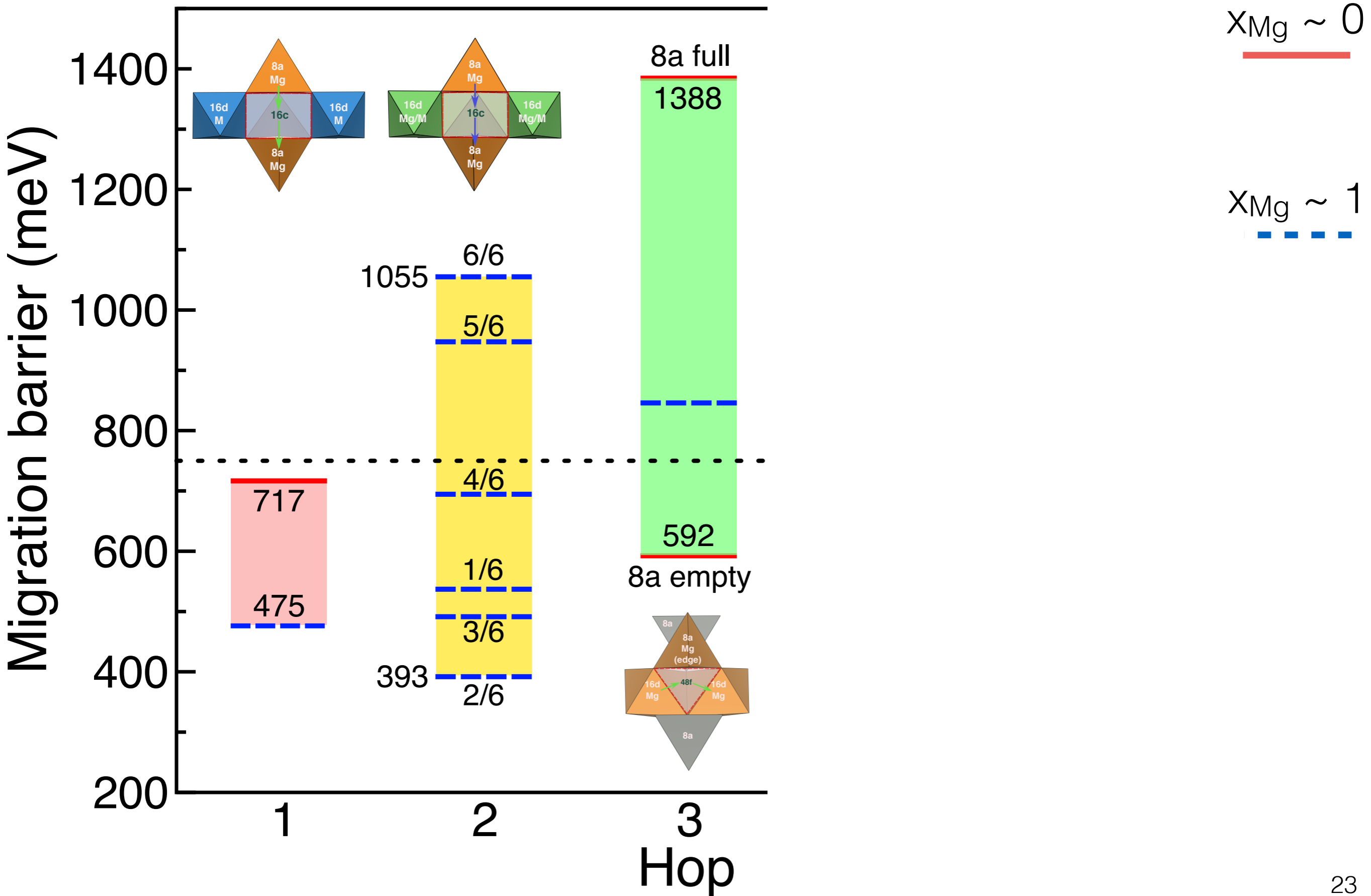
Barriers in MgMn_2O_4



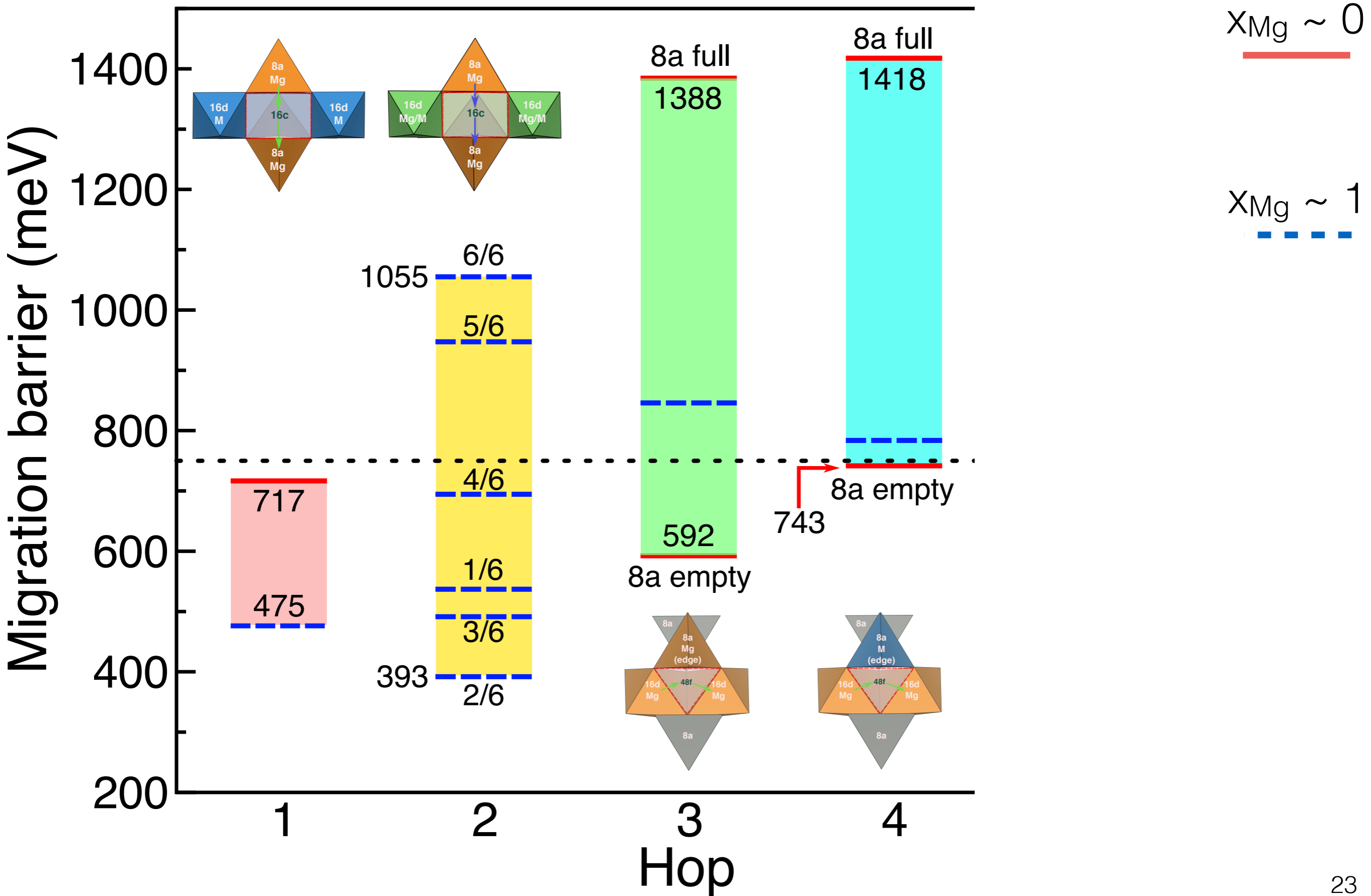
Barriers in MgMn₂O₄



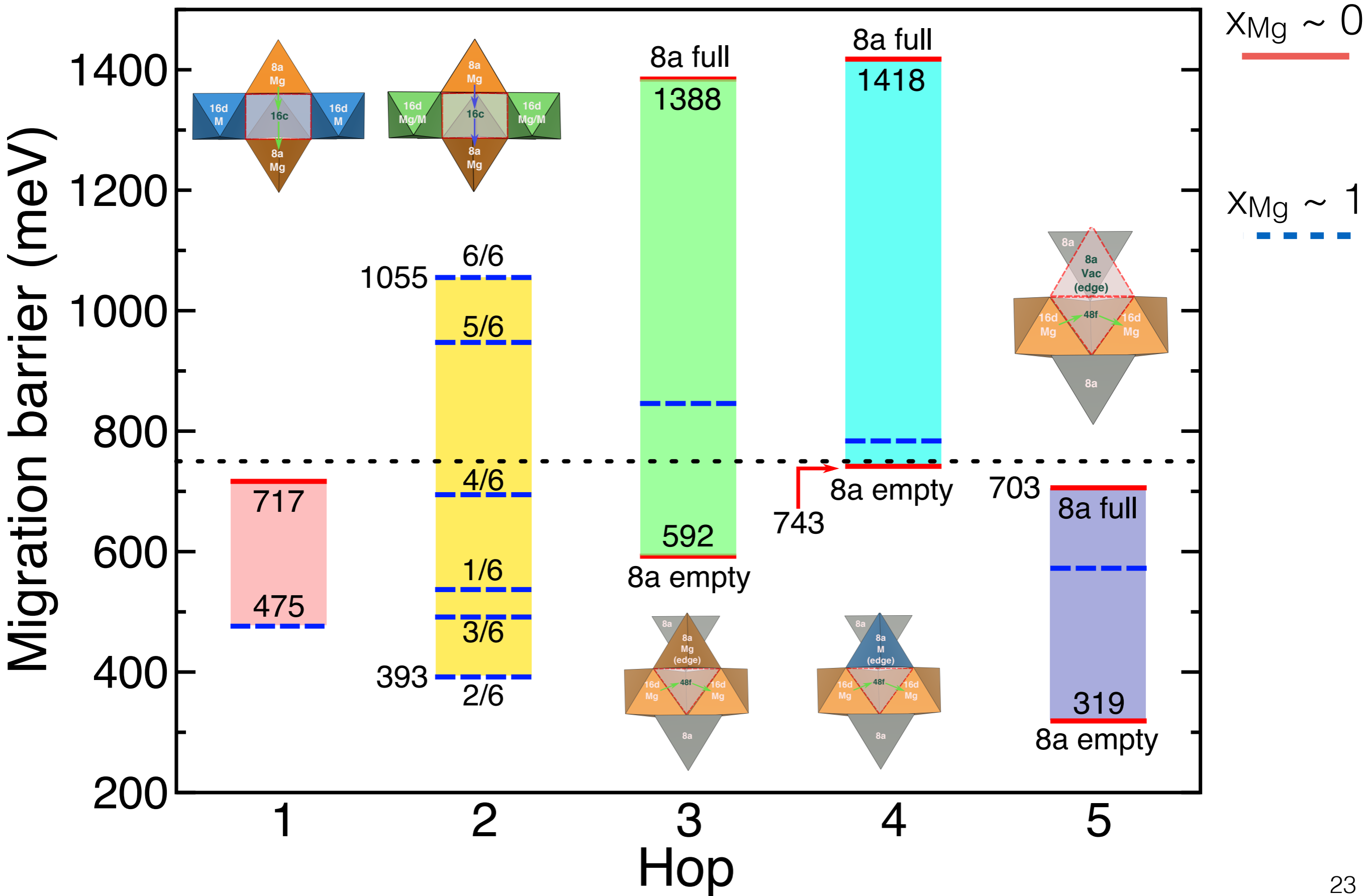
Barriers in MgMn_2O_4



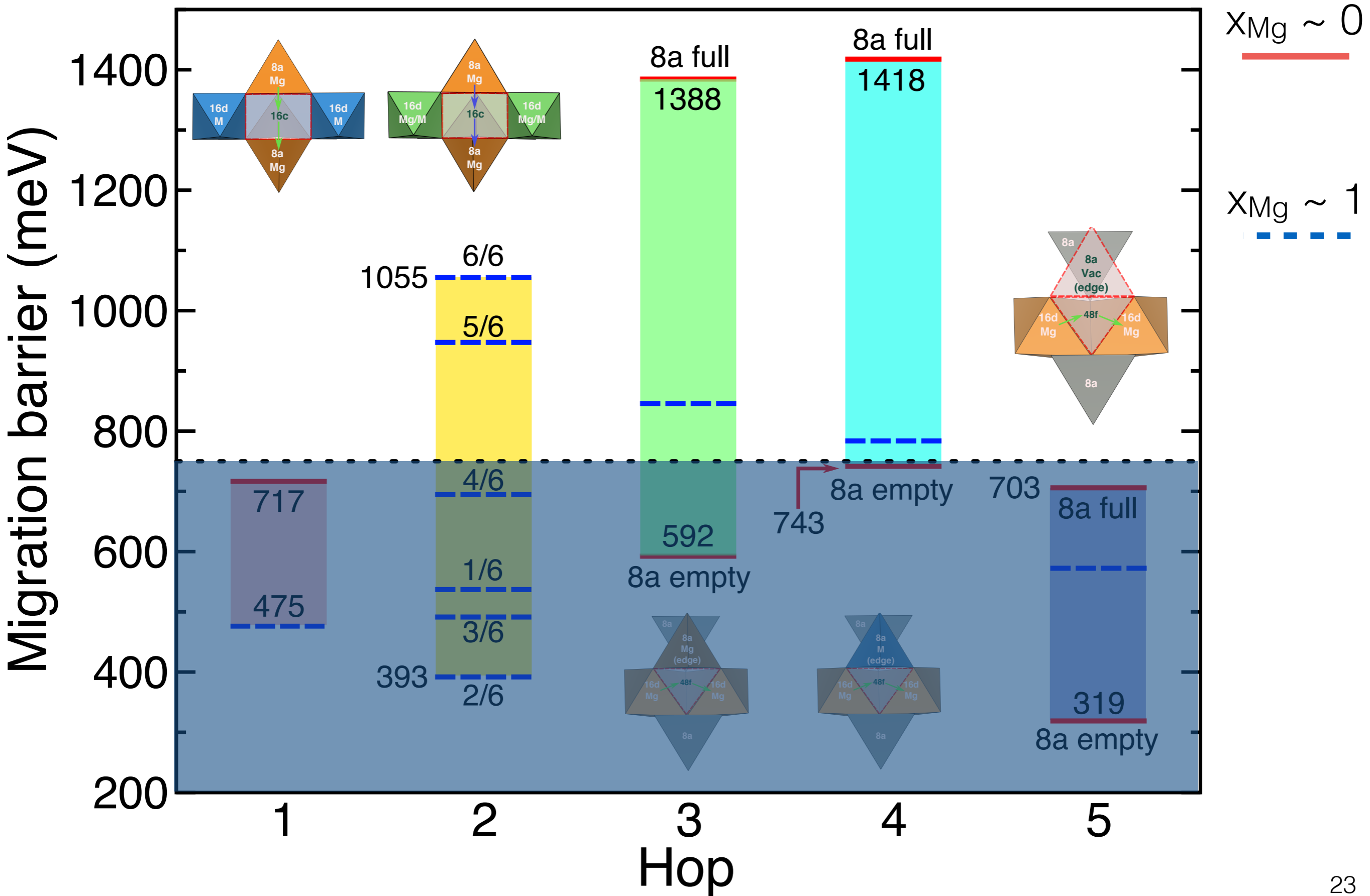
Barriers in MgMn_2O_4



Barriers in MgMn_2O_4



Barriers in MgMn_2O_4

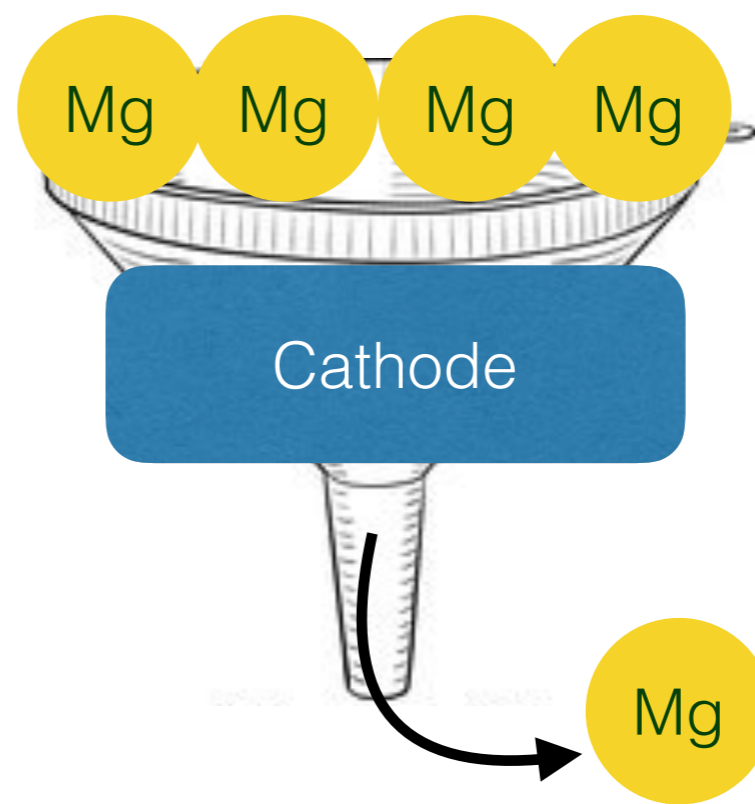
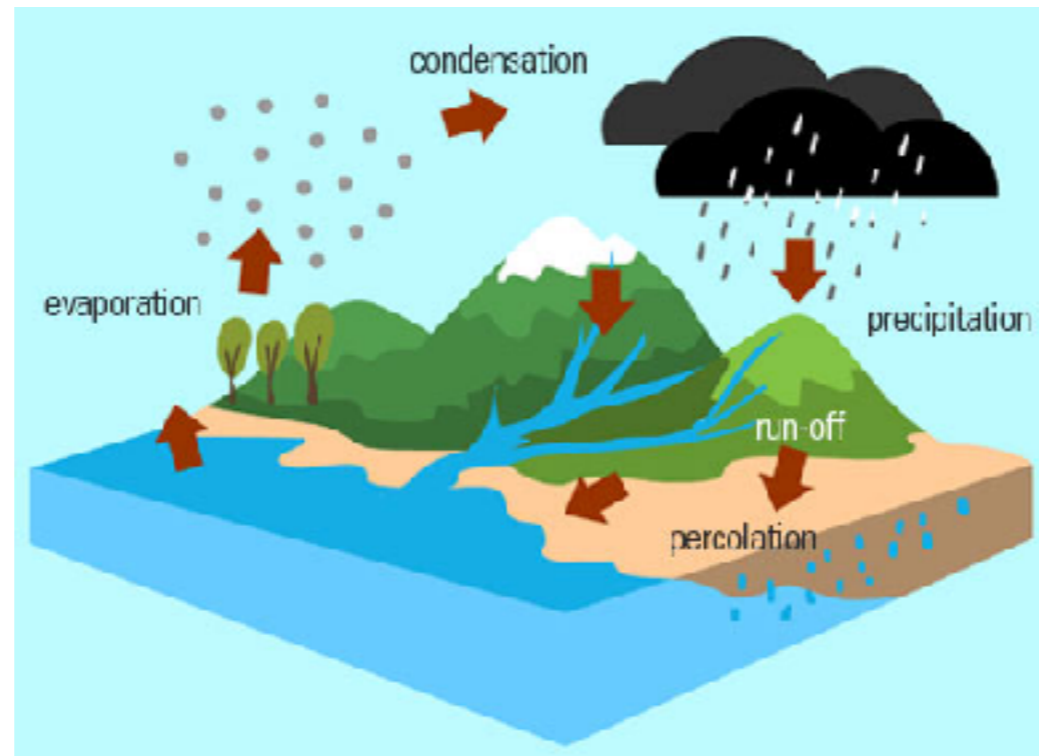


I understand barriers in local environments, but what happens to macroscopic Mg migration?

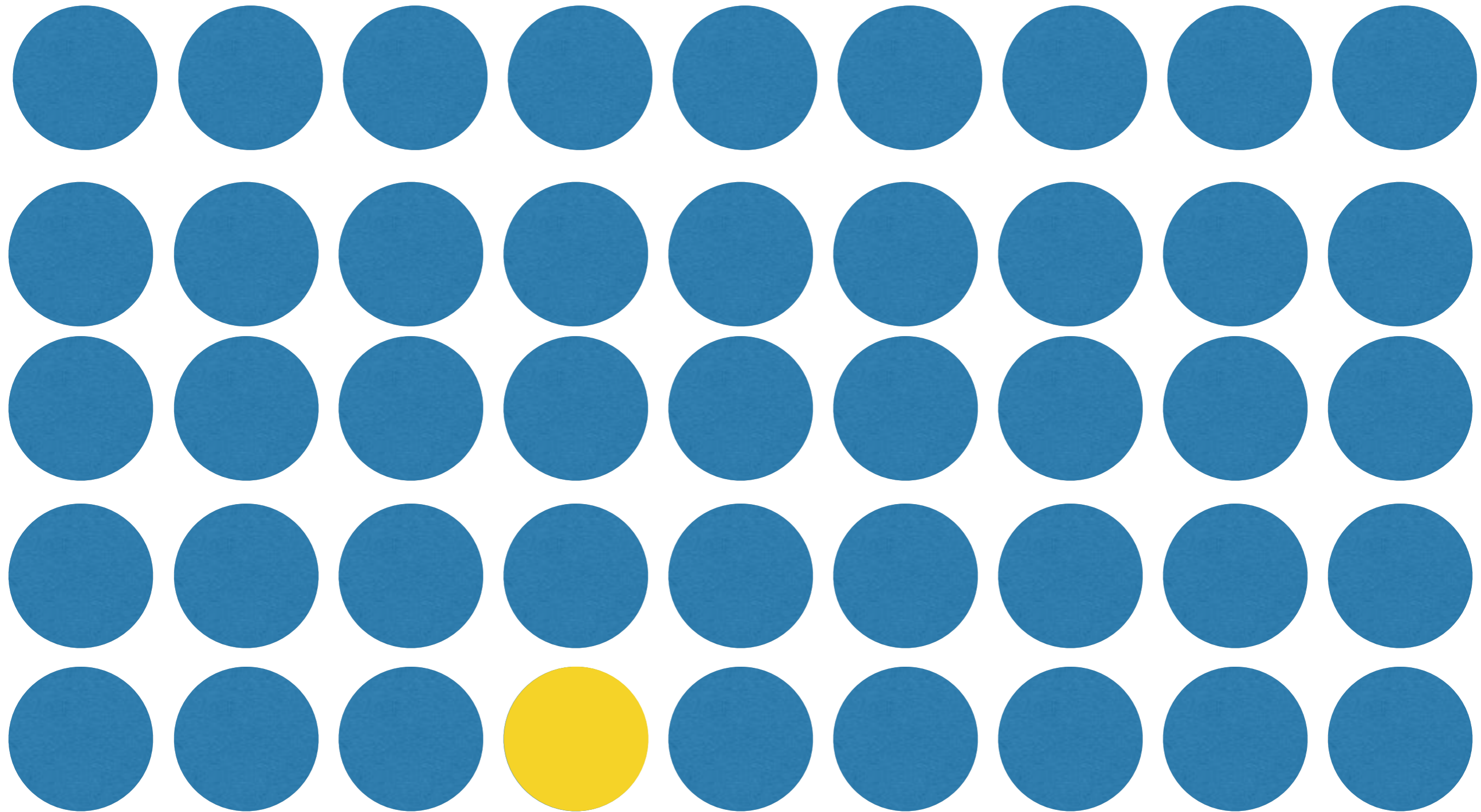
Ground water

Coffee

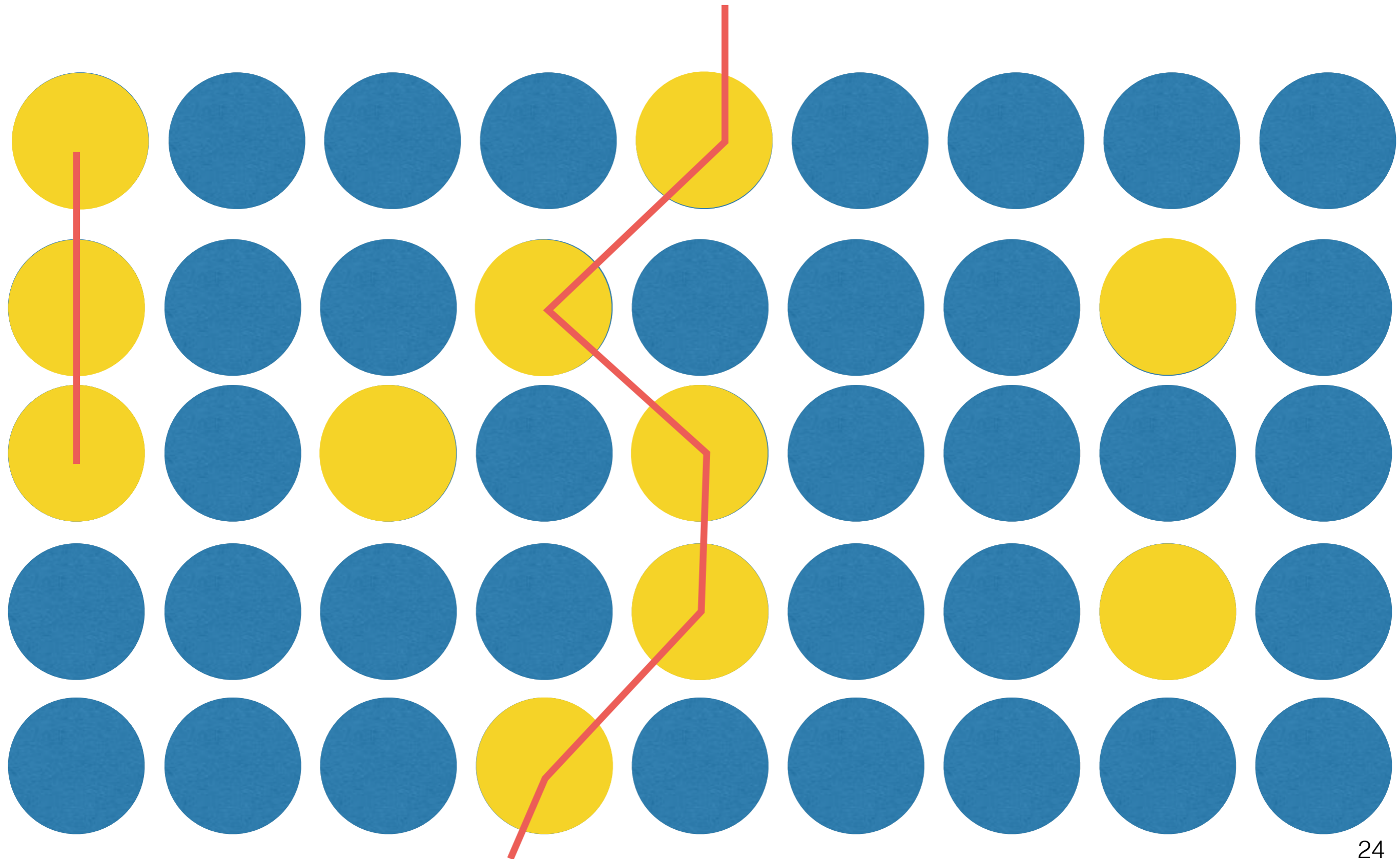
Percolation in nature



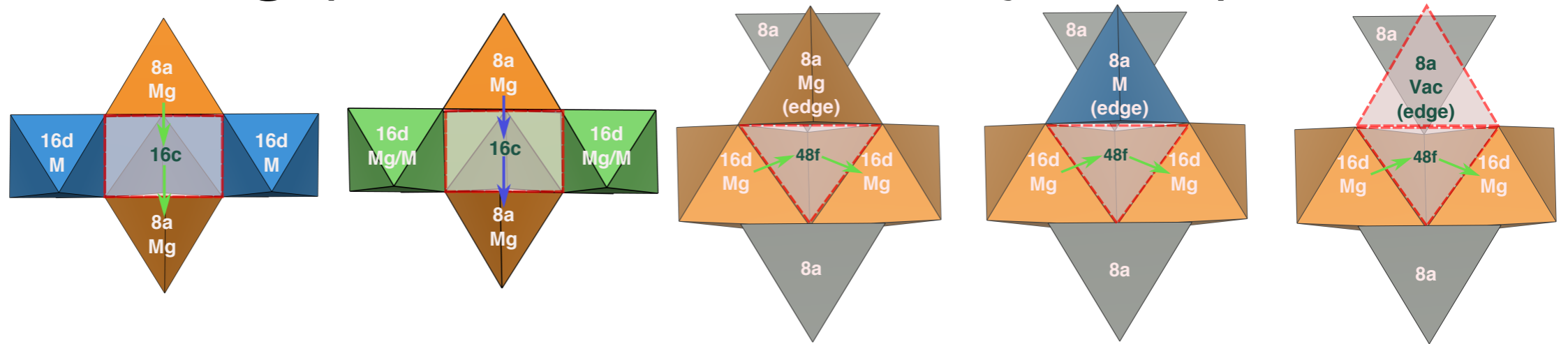
I understand barriers in local environments, but what happens to macroscopic Mg migration?



I understand barriers in local environments, but what happens to macroscopic Mg migration?



Translating percolation theory to spinels



Hop

1

2

3

4

5

Topology

tet-oct-tet

tet-oct-tet

oct-tet-oct

oct-tet-oct

oct-tet-oct

MgMn₂O₄ : 750 meV

Open under

Always open

Up to 4/6 Mg
in ring

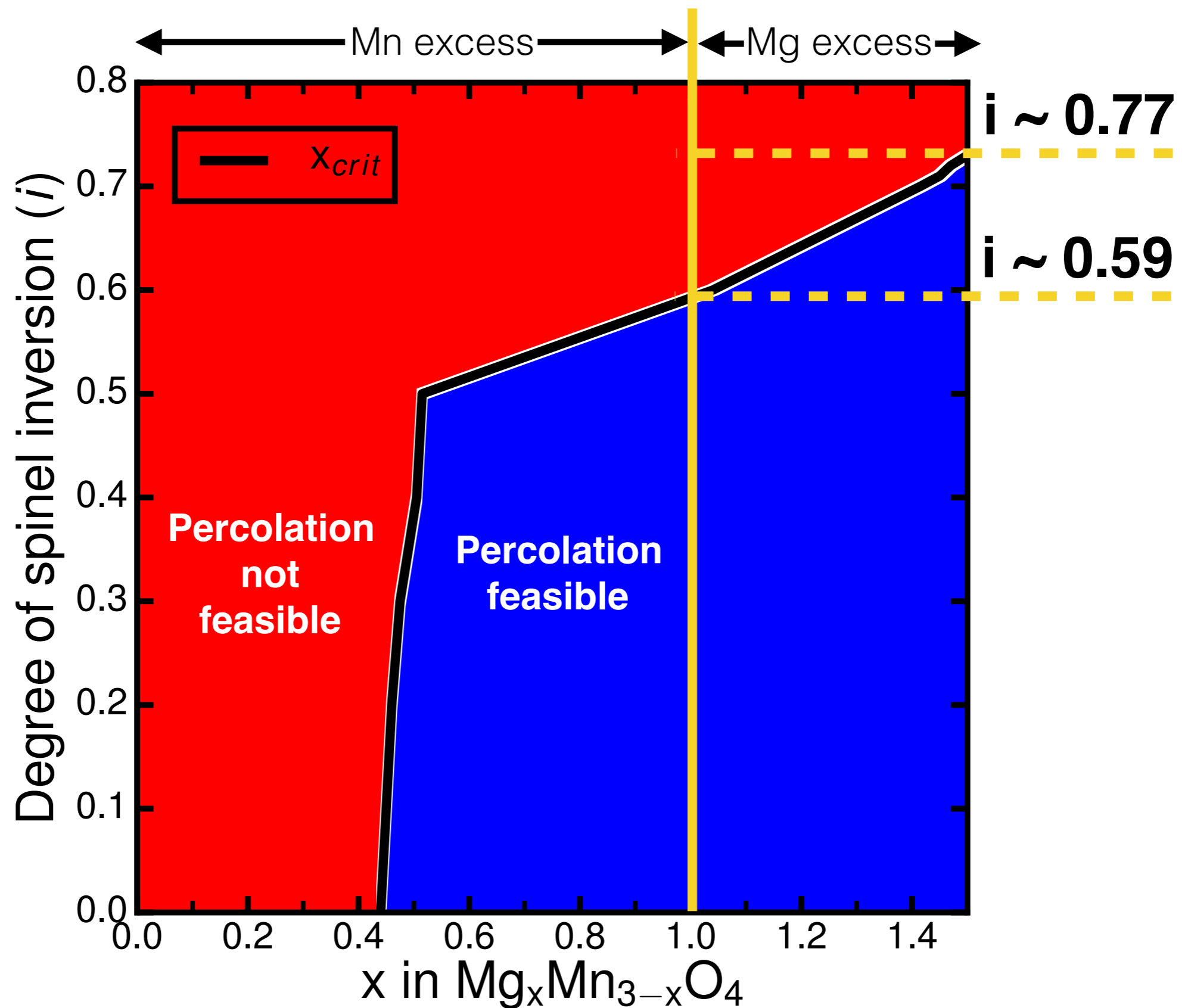
Corner 8a
vacant

Closed

Always open

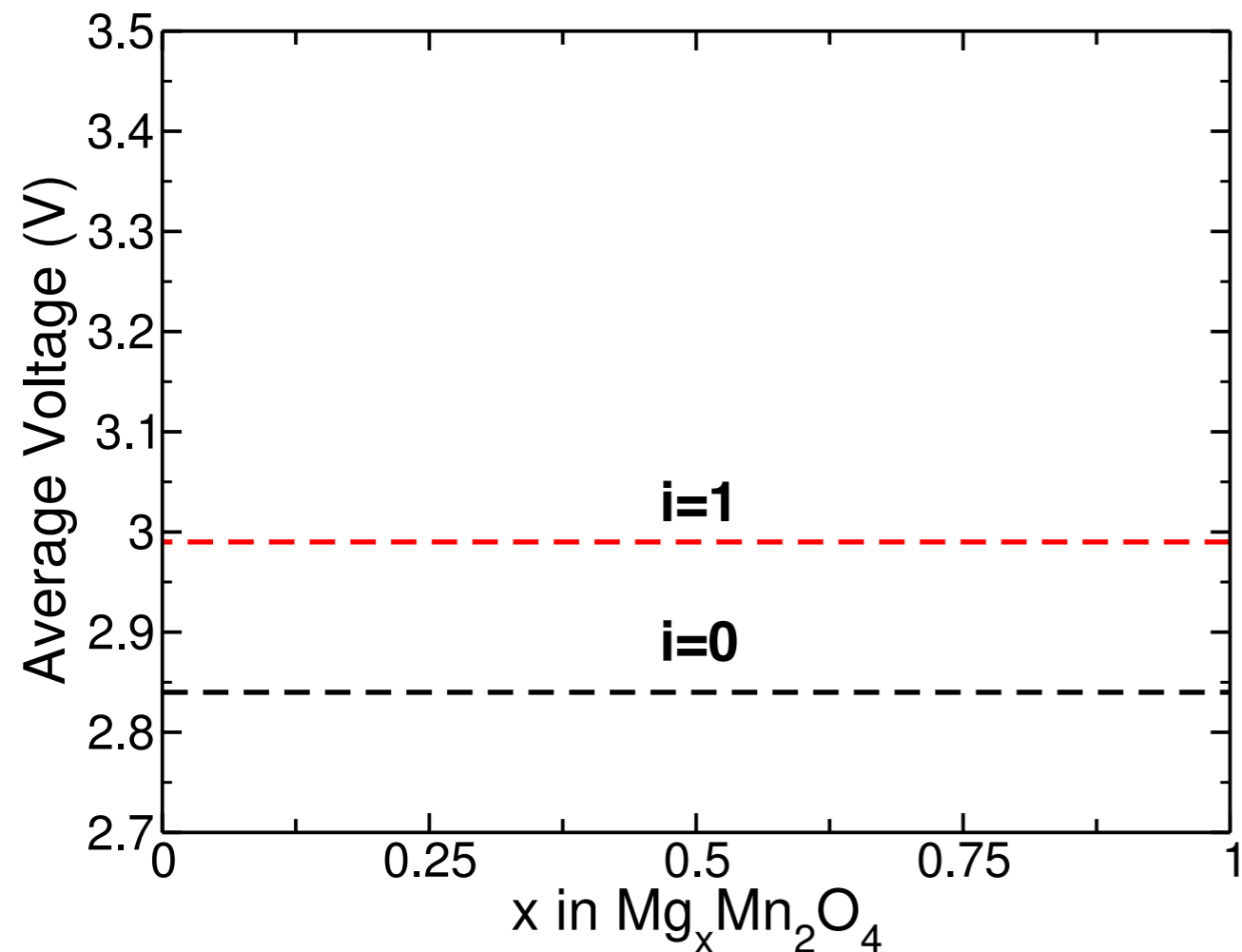
Perform Monte-Carlo simulations

Stoichiometric MgMn_2O_4 should percolate up to $\sim 59\%$ inversion



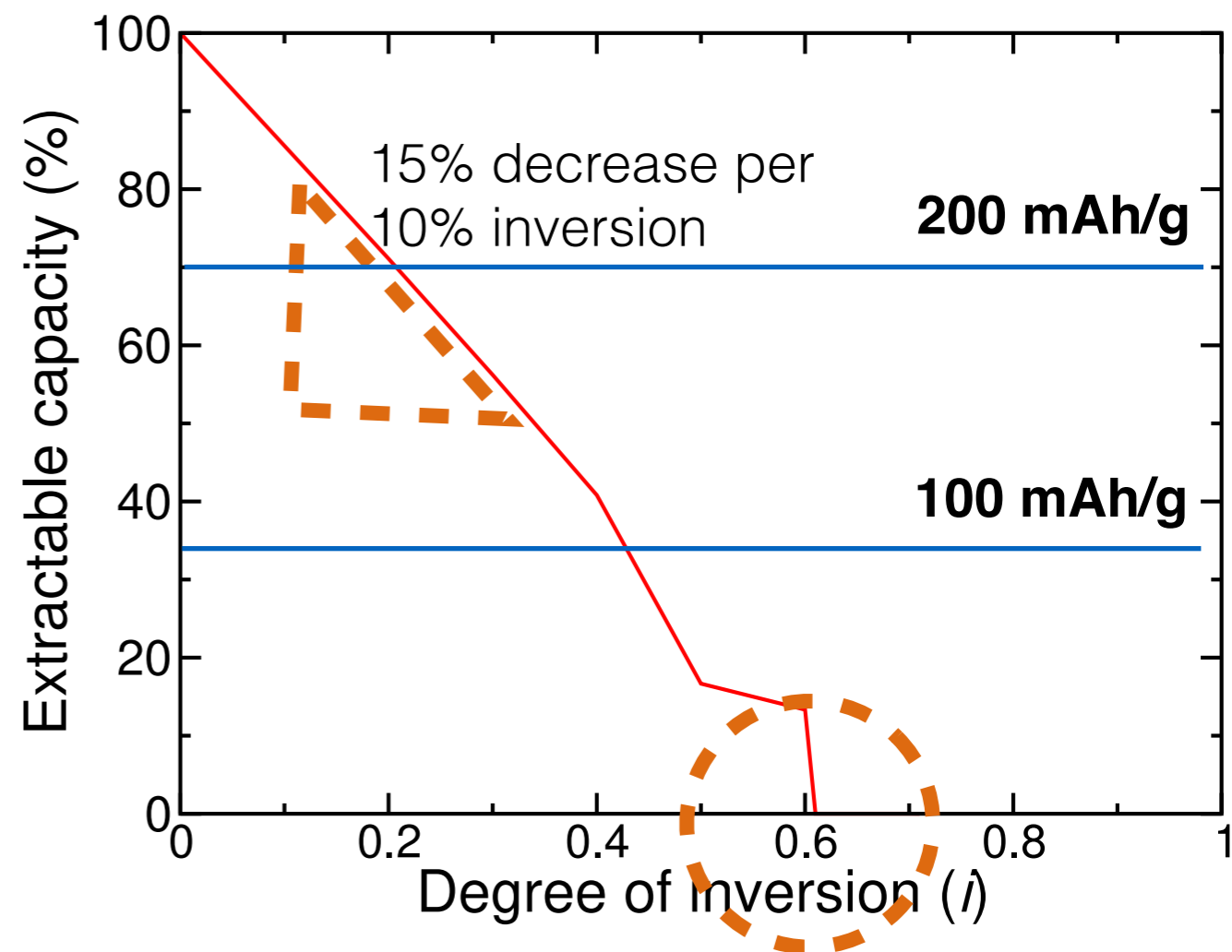
Does inversion affect electrochemical properties?

No inversion: 2.84 V
Fully inverted: 2.99 V



Inversion increases average voltage

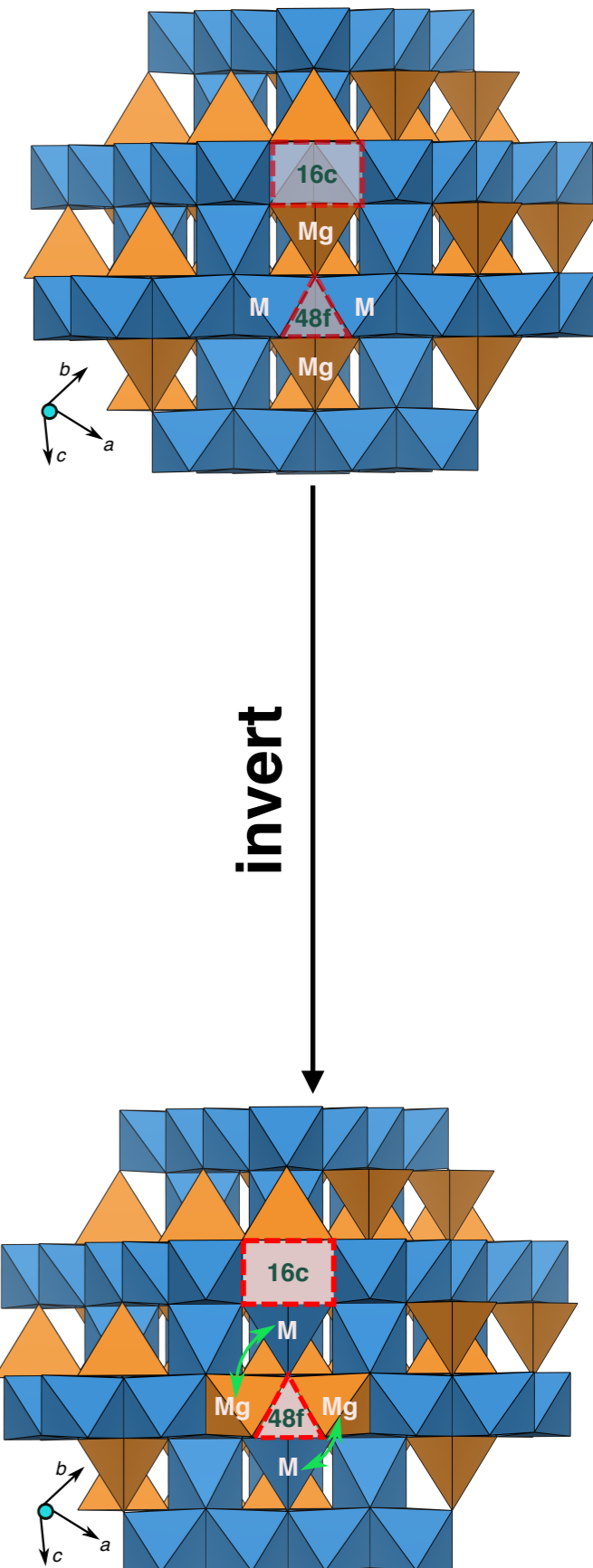
Theoretical capacity for MgMn_2O_4 : 270 mAh/g



Inversion decreases extractable capacity significantly

Summary

- MgMn_2O_4 (cathode) is susceptible to spinel inversion
- Several cation arrangements under inversion
- Inversion can open and close Mg diffusion channels
- Percolation simulations: stoichiometric MgMn_2O_4 should percolate Mg up to 59%
- Inversion raises the average voltage but detrimentally reduces extractable capacity

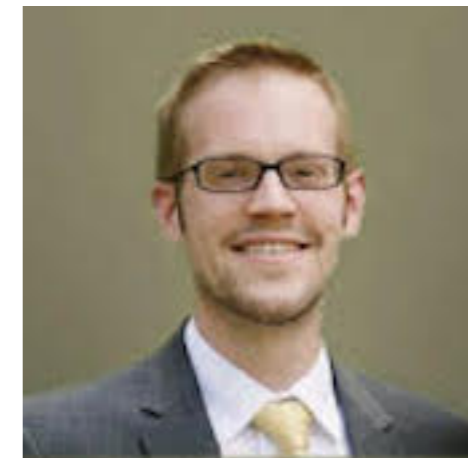
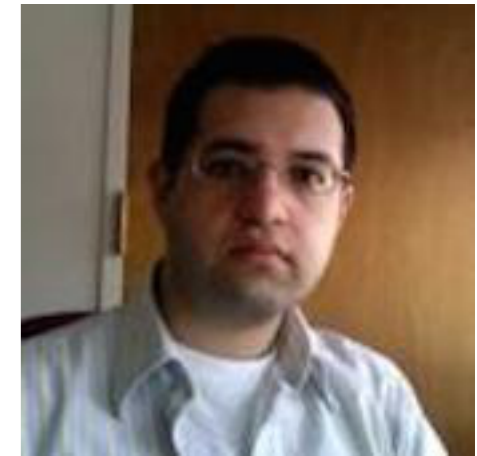
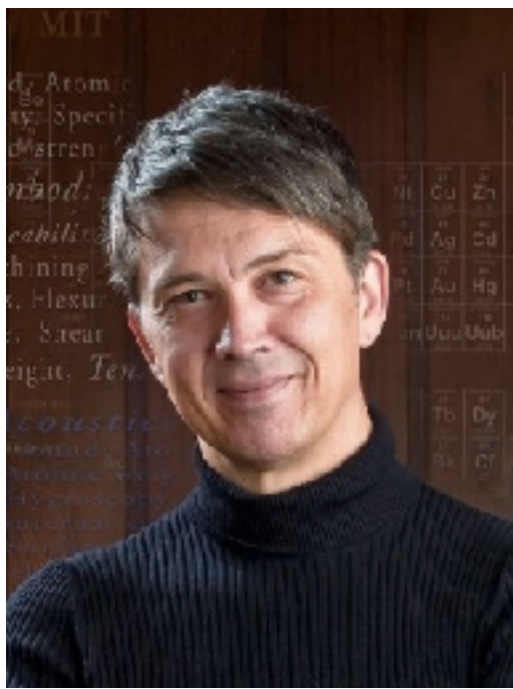


Conclusions

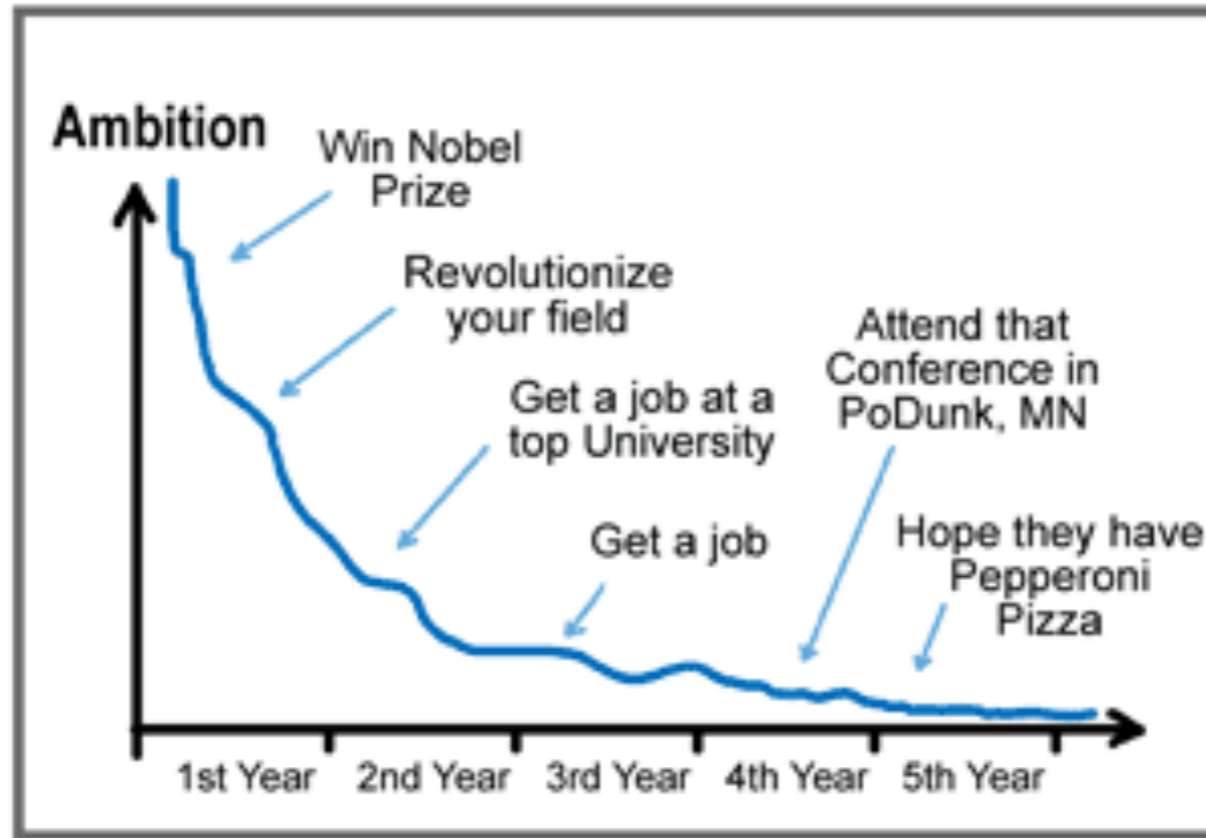


- Multivalent batteries have the potential to achieve high energy densities than current state-of-the-art Li-ion technology
 - But the development has been hindered by the **lack of energy-dense cathodes and efficient electrolytes**
- Cathode design challenge: design a high voltage, high capacity material with high Mg mobility
 - **Coordination preferences** can be a good guiding principle in finding fast diffusers
- Spinel structures are important in the Multivalent space, but may be prone to **inversion**
 - Spinel structures host Mg in an “un-preferred” tetrahedral environment
 - Inversion will affect both Mg mobility and electrochemical properties
- Develop more robust design rules/parameters to **identify frameworks with fast Mg mobility**
 - High-pressure, post-spinel phases hold the key?

Thanks!

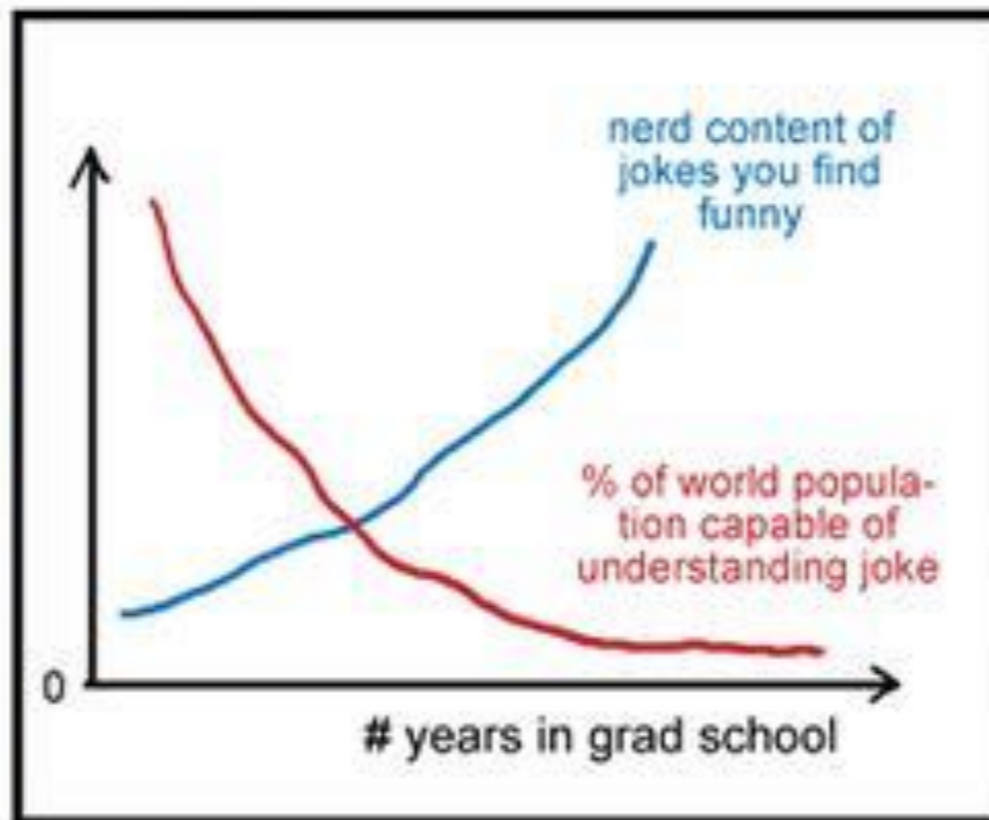


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