

# Role of exchange-correlation functionals in migration barrier predictions

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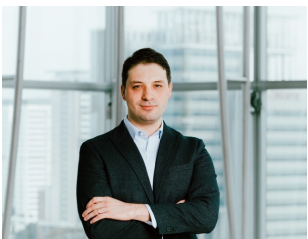
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# Acknowledgments



Reshma



+CaRe group



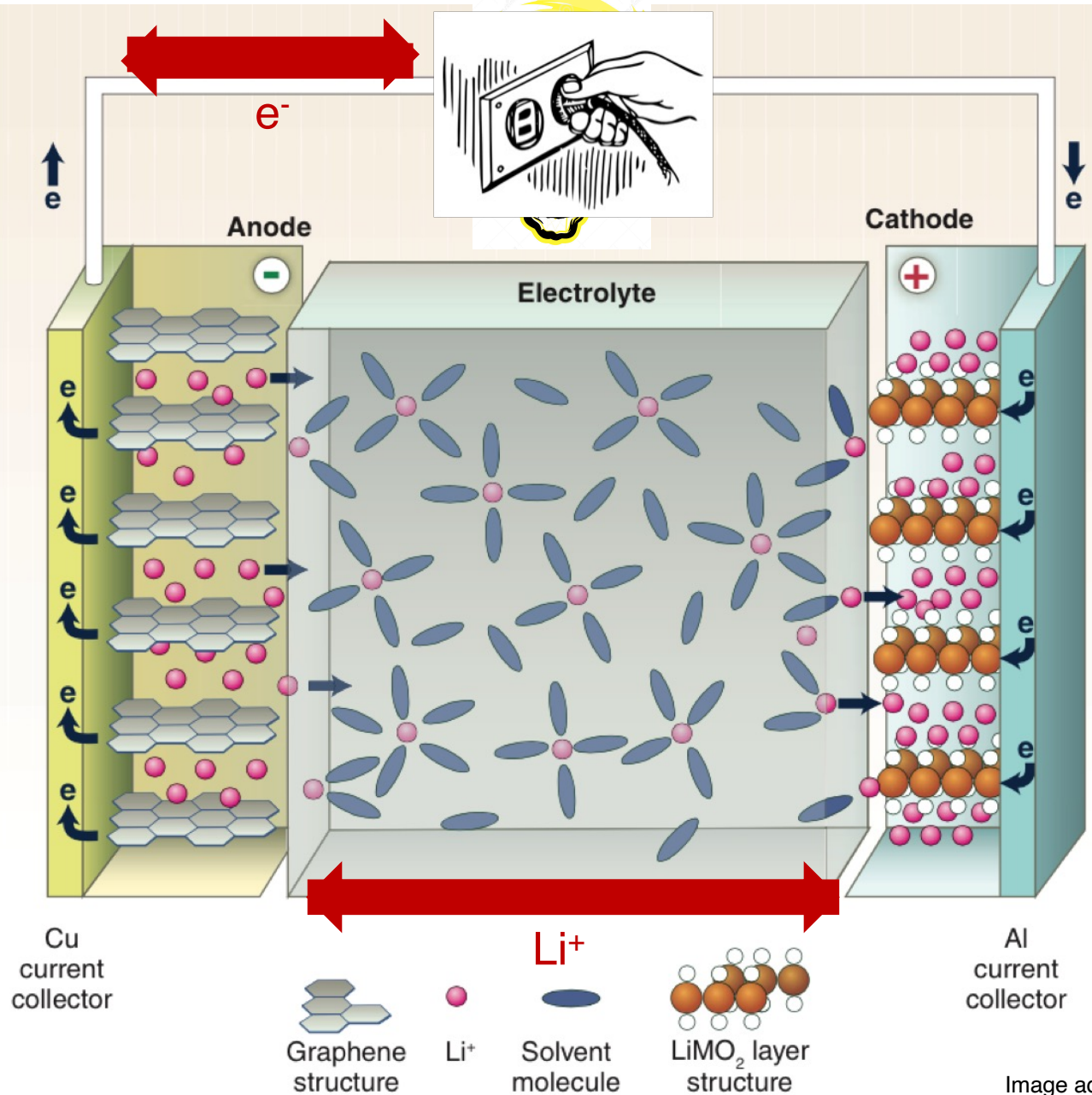
NSCC (Singapore)



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# How does the modern Li-ion battery work?



**Voltage (V):** Potential to do work

**Capacity (mAh):** Amount of charge stored

Voltage\*capacity: Energy stored

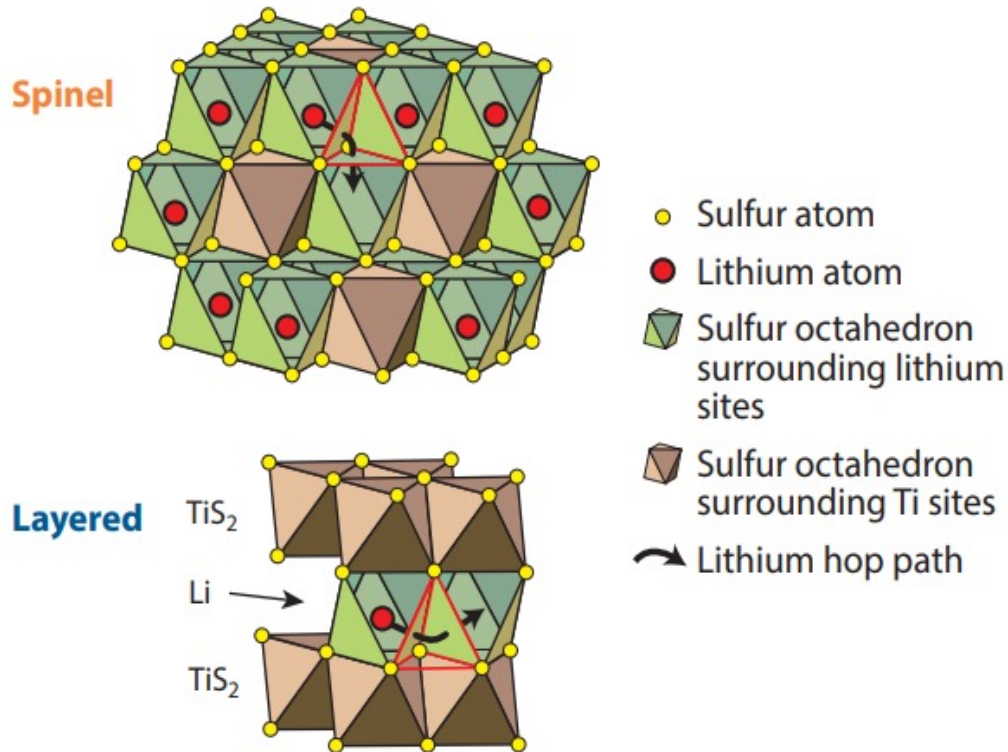
**Rate (C):** How fast can a battery be charged and discharged?

- Crucial for power performance

All performance metrics of a battery system are **material dependent:** anode, cathode, and electrolyte

Image adopted from B. Dunn et al., Science 2011

# Migration barriers govern rate performance in batteries



Intercalation electrodes/solid electrolytes: ionic diffusivity ( $D$ ) within the bulk a major factor in rate performance

$$D = D_o \exp\left(-\frac{E_m}{k_B T}\right)$$

$D_o$  : Diffusivity pre-factor (carrier concentration, correlations, etc.)

$k_B$ : Boltzmann constant

$T$ : Temperature

$E_m$ : Migration barrier

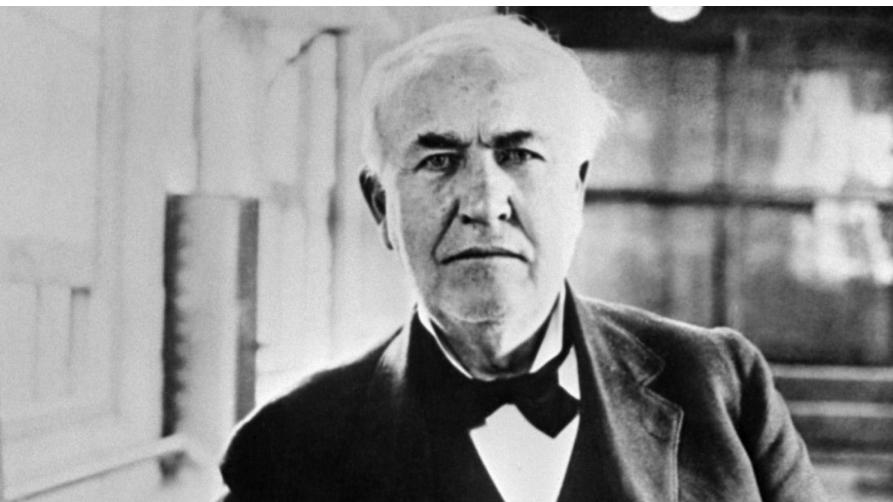
Migration barrier: dominant factor determining diffusivity

Experimental measurements of  $E_m$ : variable-temperature impedance spectroscopy, variable temperature nuclear magnetic resonance, etc.

Computational predictions of  $E_m$ : ab initio molecular dynamics, nudged elastic band (NEB)

How accurate are computational predictions?

# Methods: Edison vs. Iron Man



**Trial and error** of candidates in a lab

**Simulate and identify** candidates  
(on a transparent touch screen preferably)



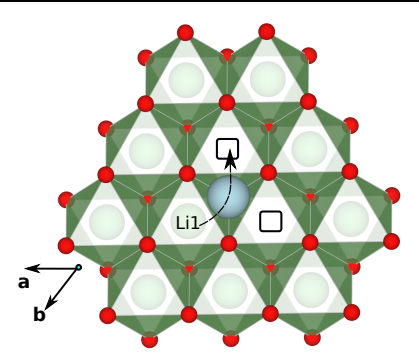
Density functional theory (DFT): [Approximately] predict material properties

- Thermodynamic, kinetic, and electronic properties

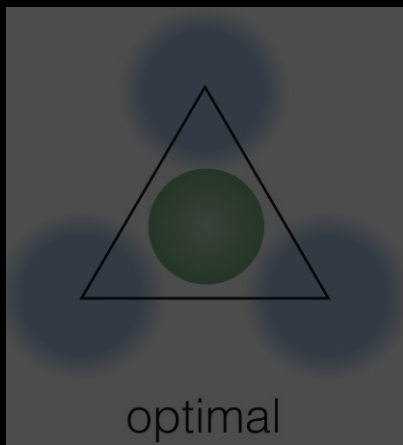
Machine learning: learn from predictions to make better predictions



# Objectives

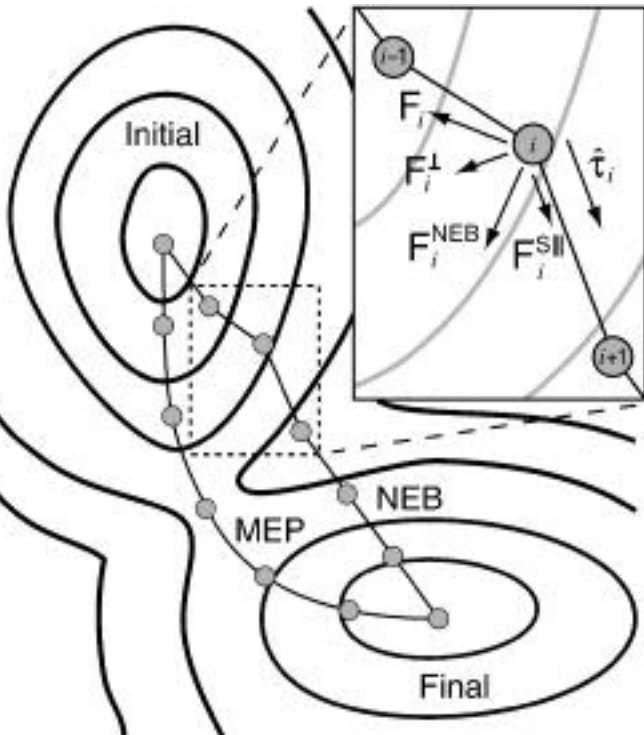


Assessing accuracy of computational predictions of ionic mobility



Examples of cathode discovery in Ca-batteries, including ionic mobility as a criterion

# NEB and computational handles for ionic mobility



NEB: can estimate  $E_m$  for an ionic hop from one "stable" site to another

Saddle-point finder by optimizing forces "perpendicular" to the potential energy surface

NEB often used with DFT: approximations within DFT affect NEB estimates

Uniform background charge (NE or ne)  
[solid electrolytes]

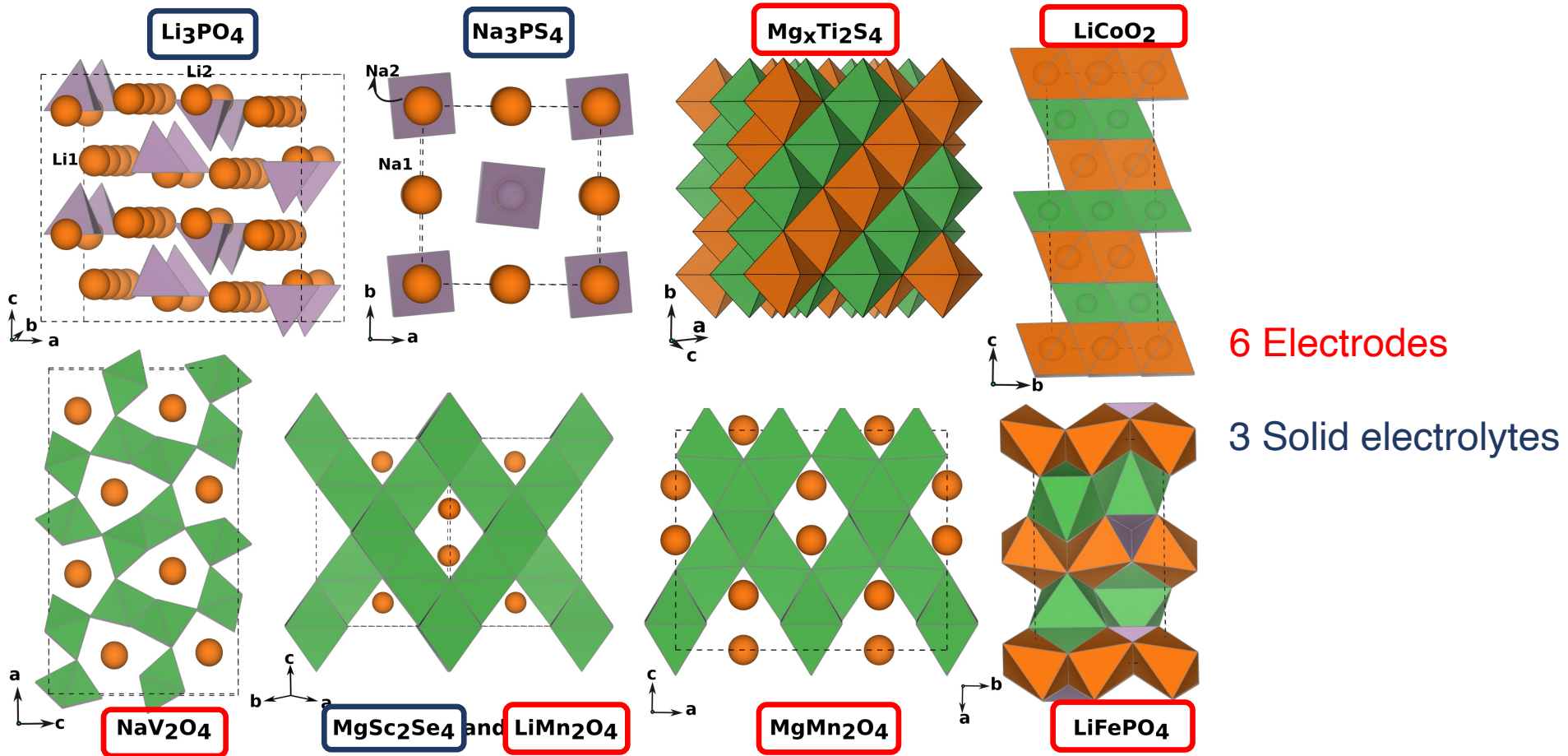
Climbing image (CI) approximation  
[solid electrolytes]

Exchange-correlation (XC) choice

- Generalized gradient approximation (GGA)
  - Perdew-Burke-Ernzerhof<sup>2</sup>
- Strongly constrained and appropriately normed (SCAN)<sup>2</sup>
- Hubbard  $U$  corrected frameworks<sup>3</sup>: GGA+ $U$ , SCAN+ $U$ 
  - For electrodes only

1. J.P. Perdew et al., Phys. Rev. Lett. 77, 3865 (1996)
2. J. Sun et al., Phys. Rev. Lett. 115, 036402 (2015)
3. V. Anisimov et al., Phys. Rev. B 44, 943 (1991)

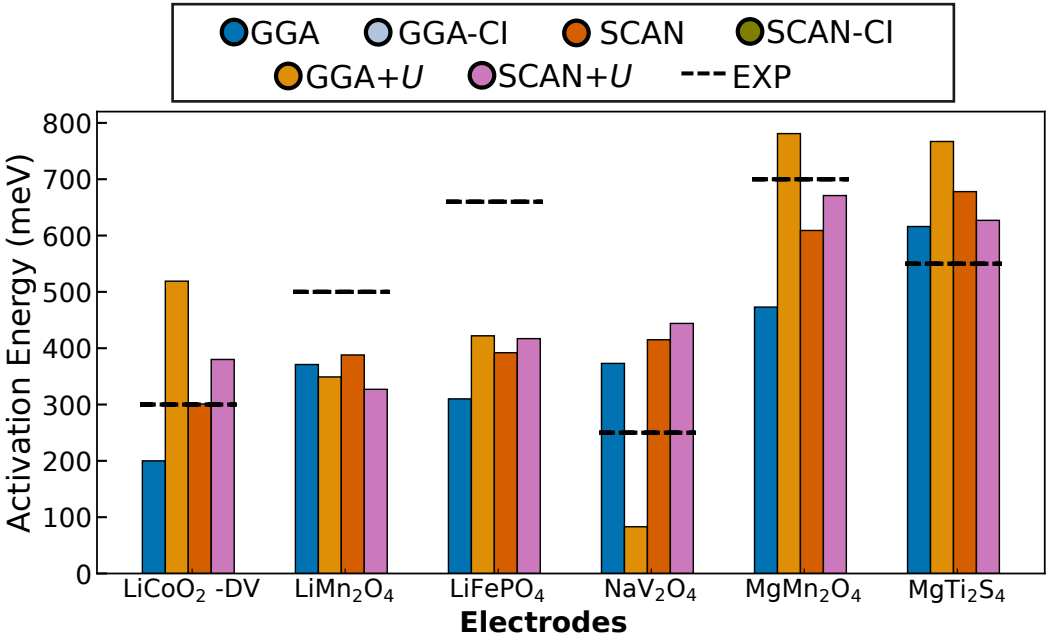
# 9 distinct systems considered



- Why these systems?
- Availability of experimental data
  - Heterogeneity of intercalation ion
  - Diversity of structural frameworks



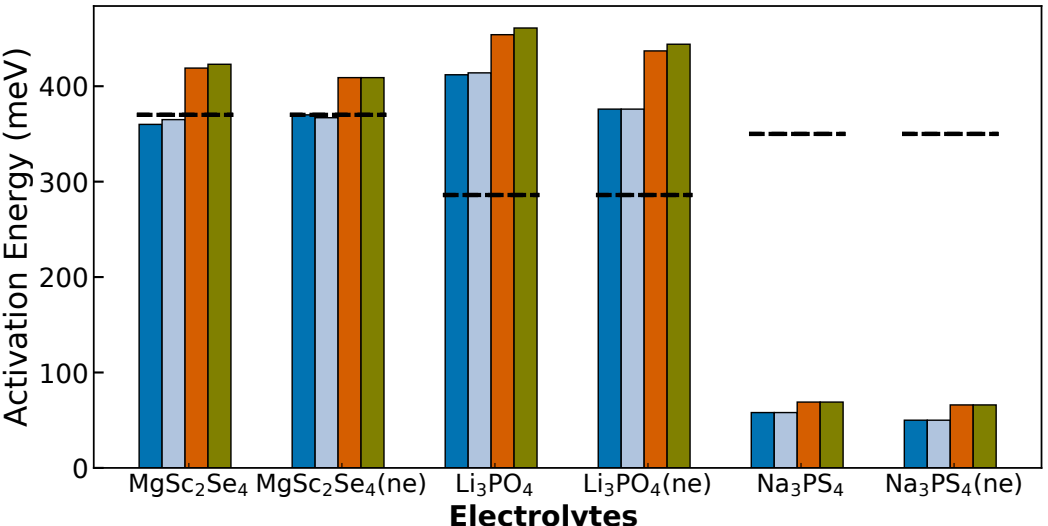
# SCAN exhibits better numerical accuracy, *on average*



MAE of SCAN (140 meV) lower than other XC frameworks (>145 meV)

SCAN  $E_m >$  GGA  $E_m$

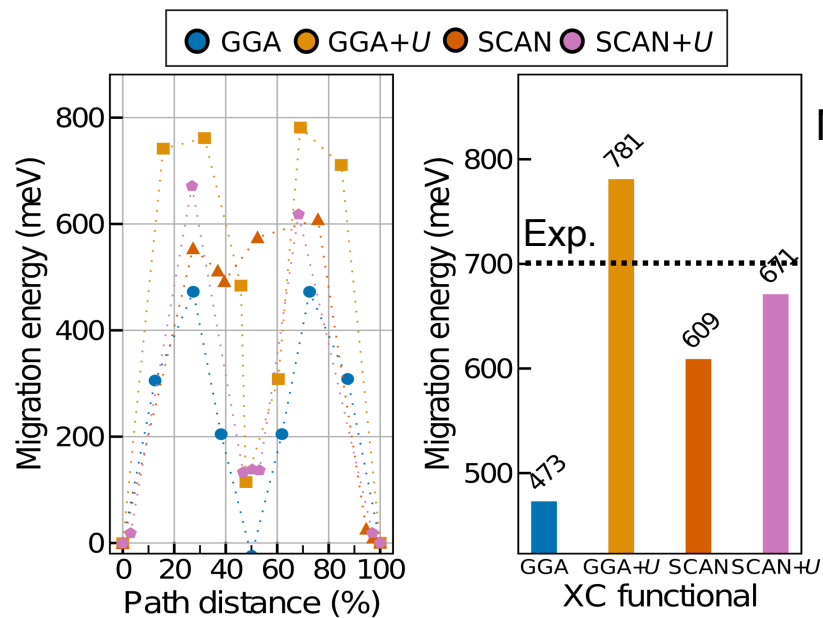
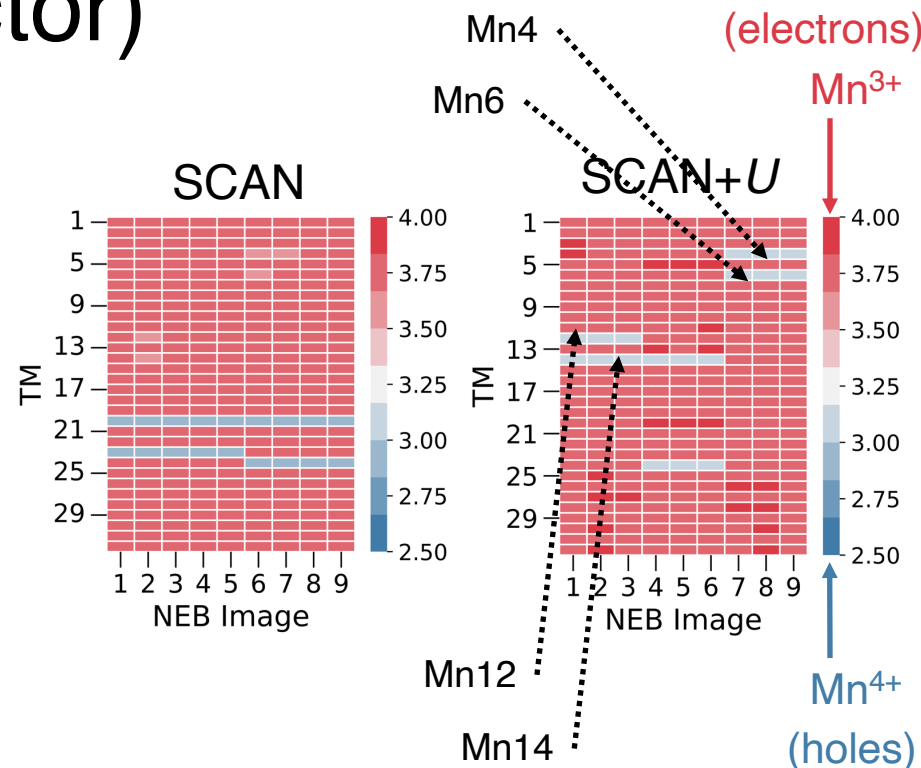
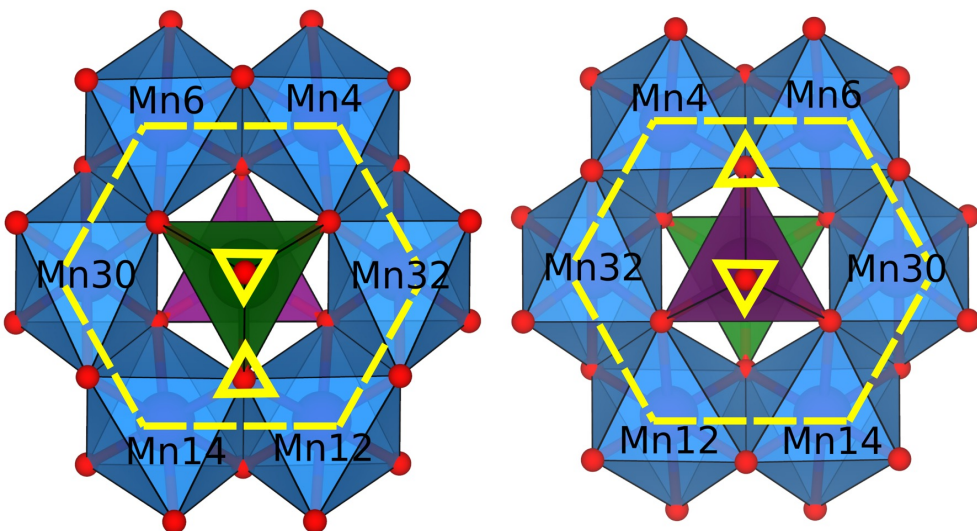
SCAN+U  $E_m <$  GGA+U  $E_m$



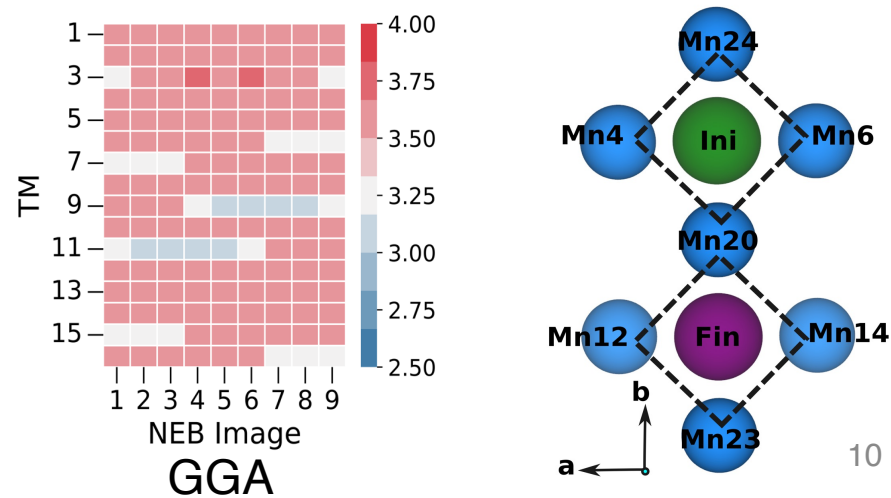
Addition of ne or Cl doesn't seem to matter for solid electrolytes

Let's look at some specific cases

# MgMn<sub>2</sub>O<sub>4</sub> (semiconductor) GGA underestimates

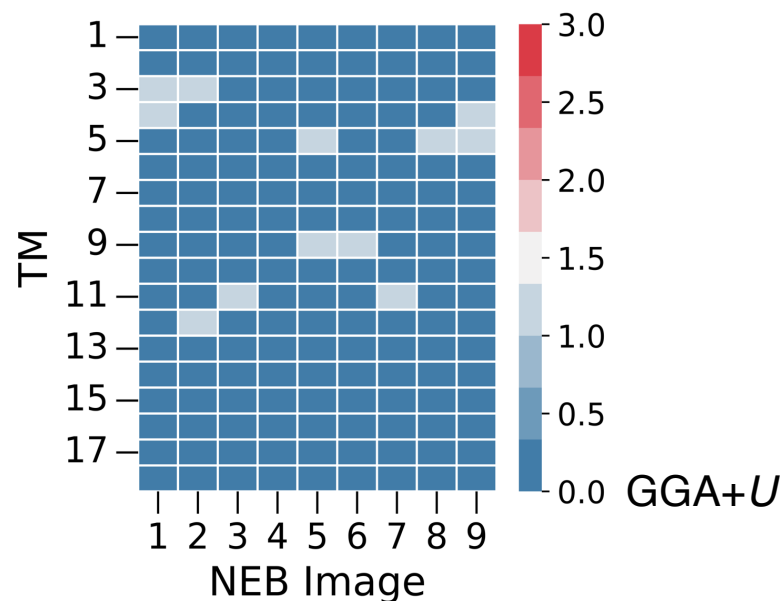
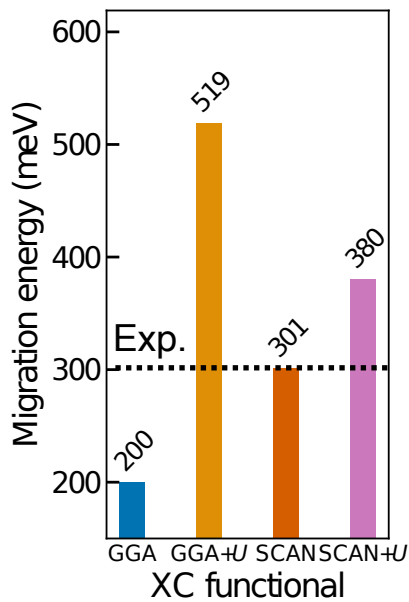
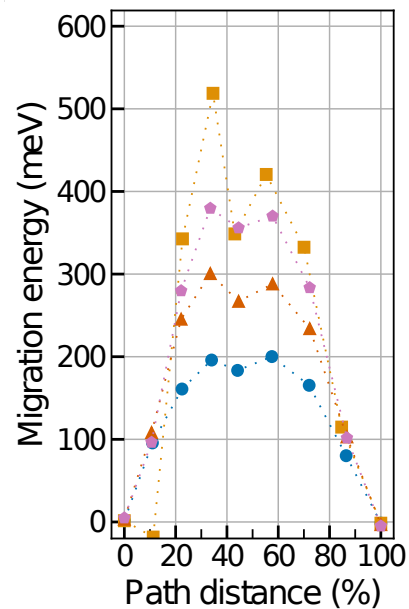
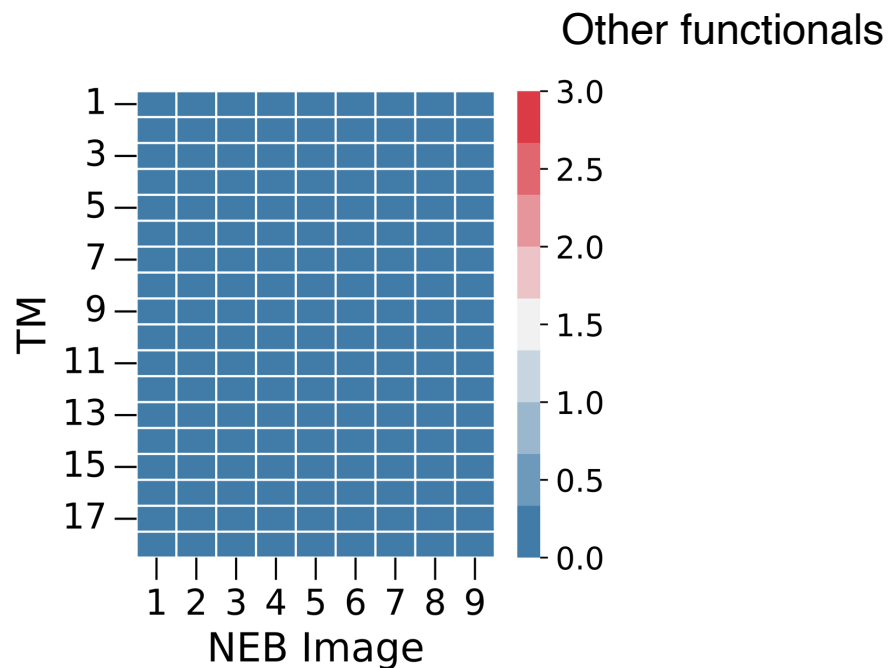
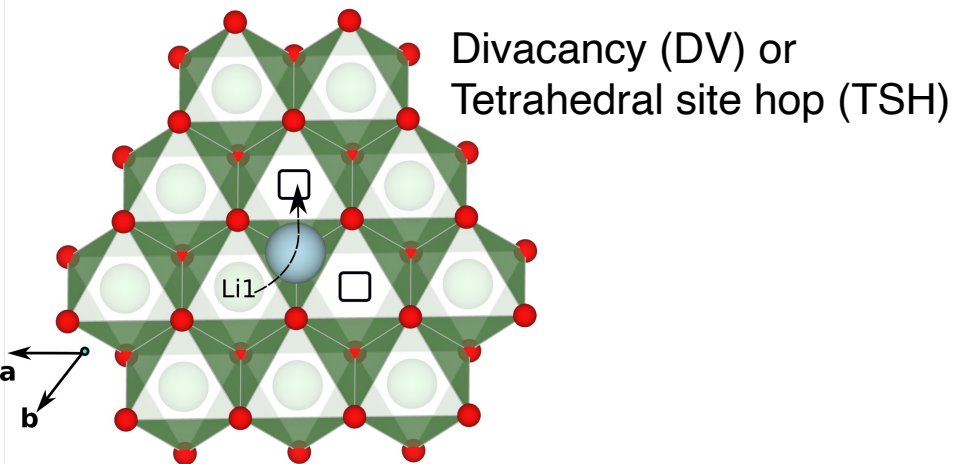


Mg<sup>2+</sup> and holes should migrate in opposite directions



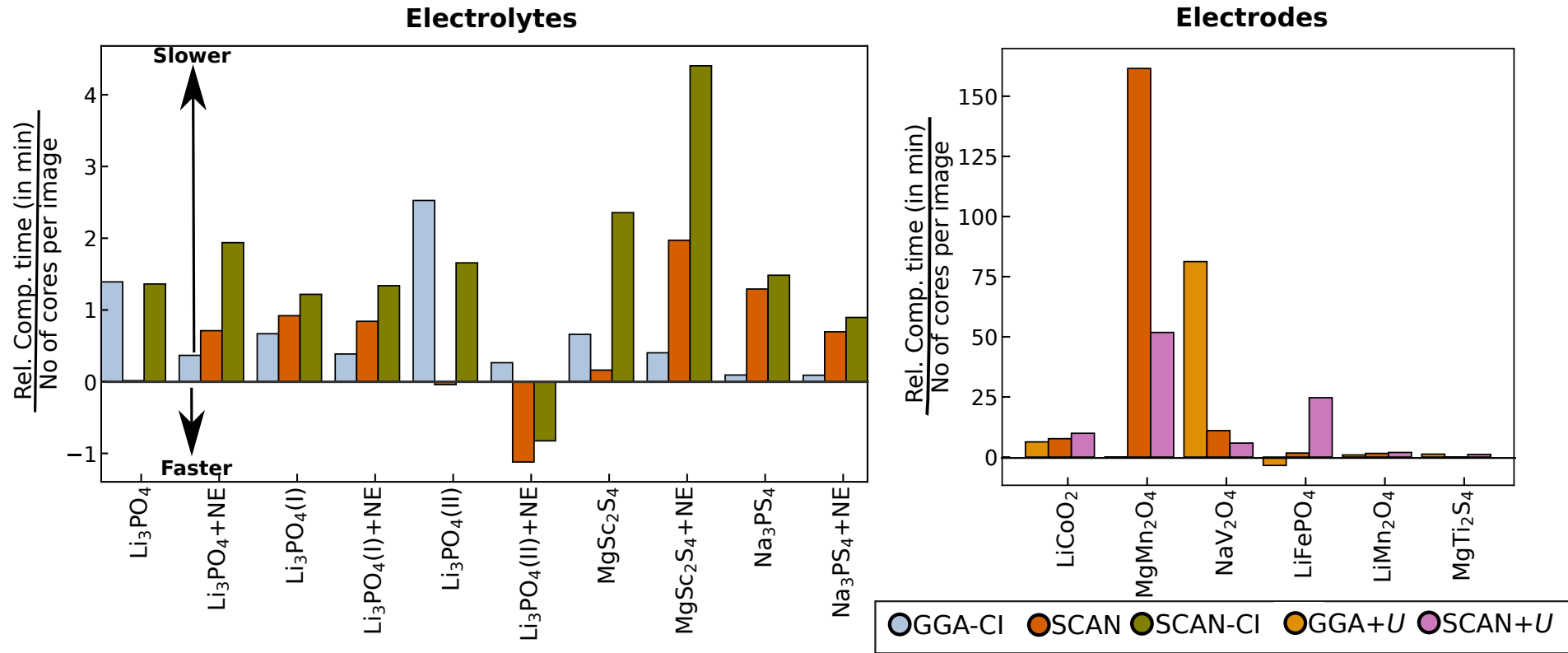
# LiCoO<sub>2</sub> (metallic)

## GGA+U overestimates



# Computational performance

## Is SCAN worth it?

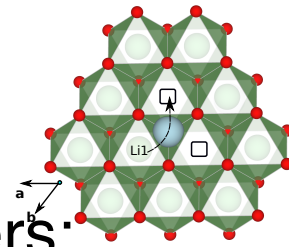


- Computational time with GGA/GGA+U  $\sim 75\%$  faster than SCAN
- SCAN faster than SCAN+U
- Significant convergence difficulties with SCAN and SCAN+U

GGA for "quicker" estimate

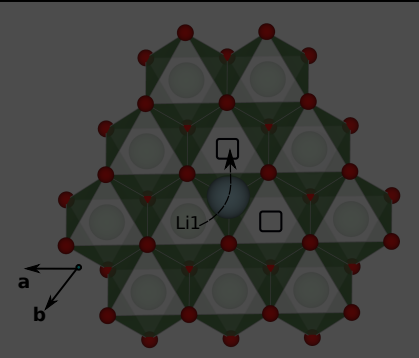
SCAN for "better" accuracy

# Summary

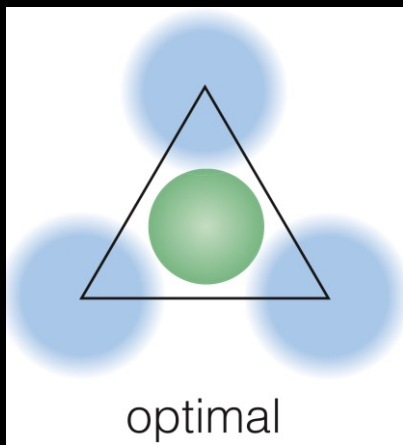


- Accurate computational predictions of migration barriers: important to improve rate performance of batteries
- NEB calculations – three computational handles relevant
  - XC framework
  - Uniform background charge
  - Climbing image
- Benchmark computational predictions with experimental barriers for 6 electrodes and 3 solid electrolytes
- SCAN: better accuracy on average
  - Describes better physics in both semiconducting and metallic hosts
  - CI/ne doesn't influence migration barriers significantly
- Computational hosts+ convergence difficulties: high for SCAN
  - SCAN for better accuracy, but GGA can provide good qualitative trends

# Objectives

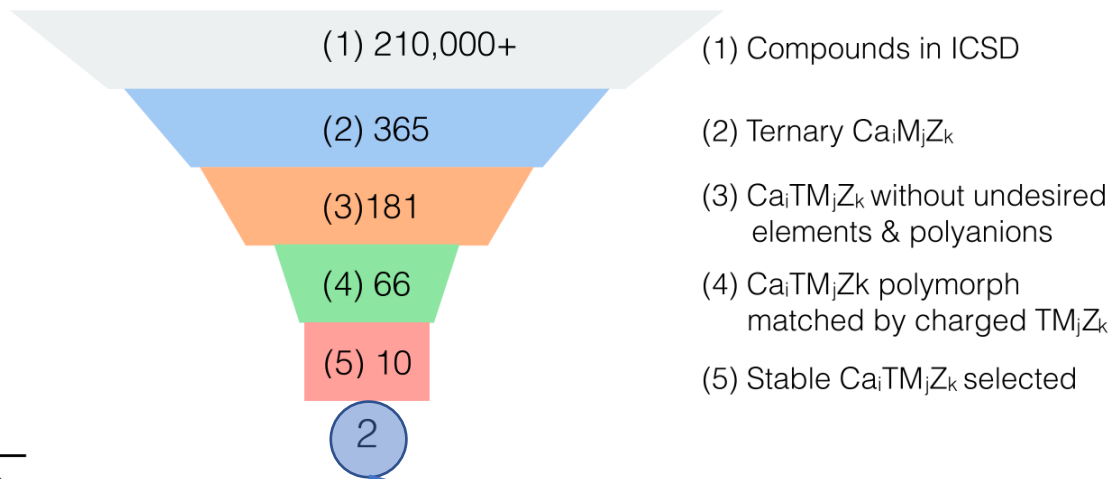
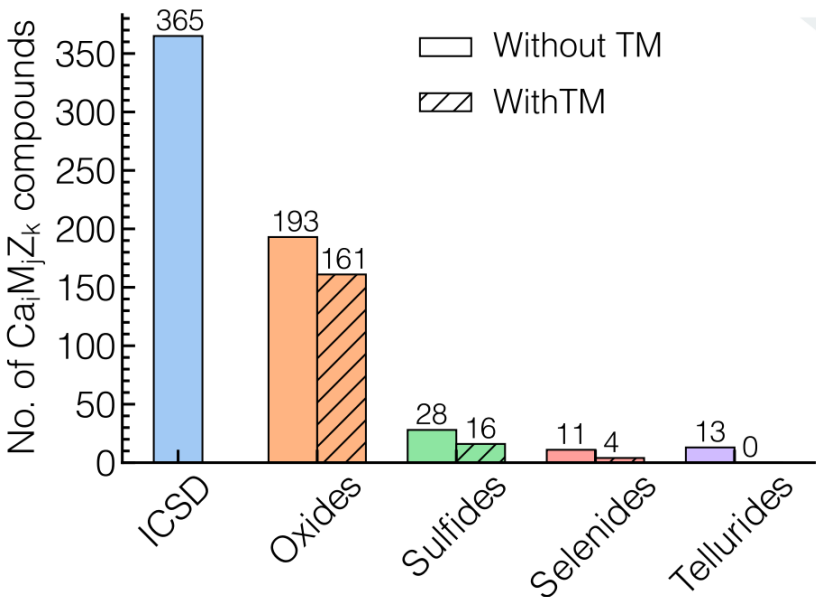


Assessing accuracy of computational predictions of ionic mobility



Examples of cathode discovery in Ca-batteries  
(Ternary chemical space, NaSICONs)

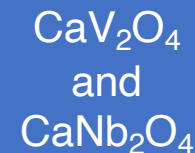
# Ternary Ca-compounds



Inorganic crystal structure database (ICSD<sup>1</sup>): has > **210,000** compounds

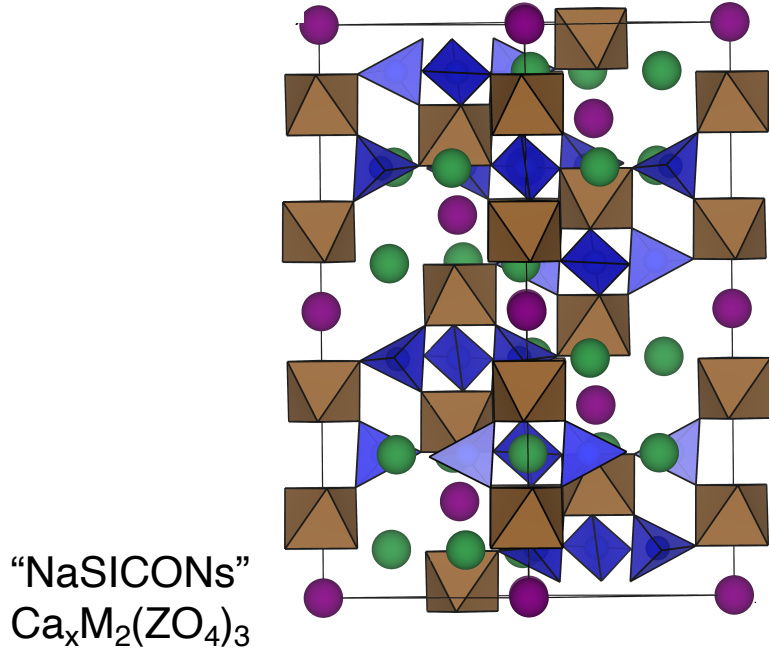
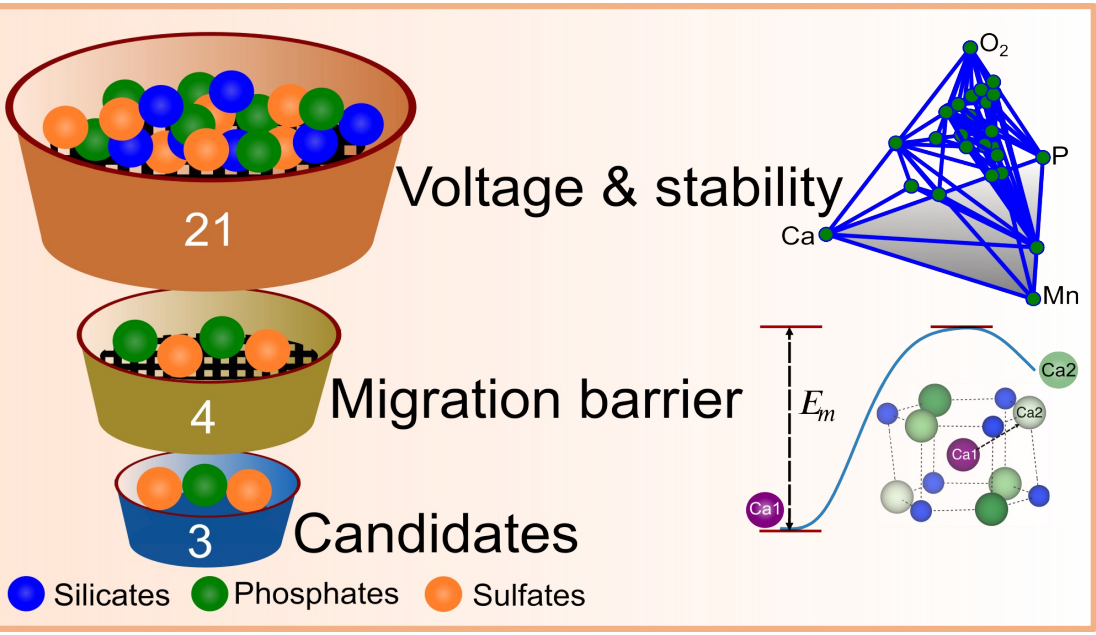
- Only **365** are ternary compounds containing Ca
  - Compounds of composition  $\text{Ca}_i\text{M}_j\text{Z}_k$ ; M, Z = elements other than Ca
- Let M = TM (i.e., transition metal) and Z = O, S, Se, or Te: **181** compounds
- Charge-neutral charged compound ( $\text{TM}_j\text{Z}_k$ ) available for  $\text{Ca}_i\text{TM}_j\text{Z}_k$ ?
  - $\text{CaMn}_2\text{O}_4$ - $\text{Mn}_2\text{O}_4$  is ok,  $\text{CaVO}_3$ - $\text{VO}_3$  not ok
  - **66** unique structures
- Either of  $\text{Ca}_i\text{TM}_j\text{Z}_k$  or  $\text{TM}_j\text{Z}_k$  thermodynamically (meta)stable?
  - $E^{\text{hull}} \leq 30$  meV/atom (based on Materials Project<sup>2</sup>)
  - **10 unique compounds** → evaluate (voltage and) mobility

Final candidates!

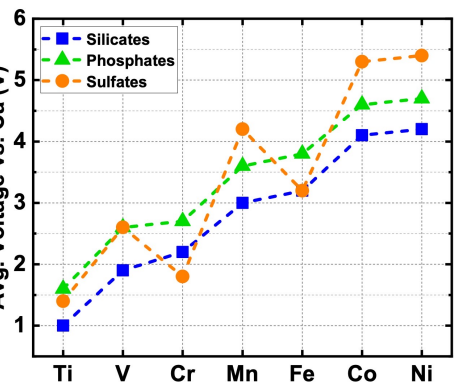


1. <https://icsd.products.fiz-karlsruhe.de/>  
 2. <https://materialsproject.org/>

# Sodium superionic conductors (NaSICONs)

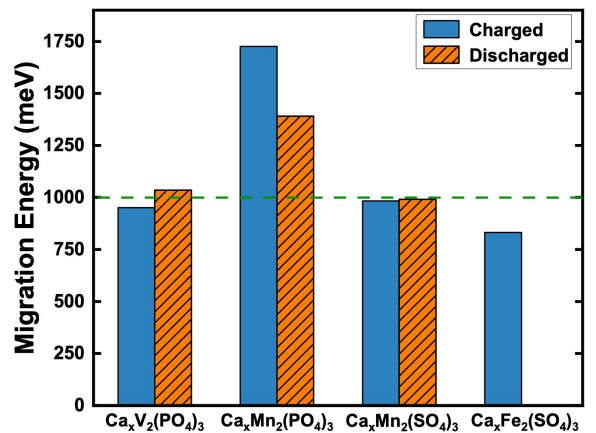


## High-throughput DFT calculations: 3 candidates



$Ca_2M_2(SiO_4)_3$	71	93	706	111	192	237	269
$Ca_4M_2(SiO_4)_3$	93	100	450	83	93	84	110
$Ca_{0.5}M_2(PO_4)_3$	-45	-8	12	-23	92	194	1173
$Ca_{2.5}M_2(PO_4)_3$	129	54	108	-11	35	50	693
$M_2(SO_4)_3$	-159	-107	-224	-74	-182	64	71
$CaM_2(SO_4)_3$	174	63	172	21	29	27	27
	Ti	V	Cr	Mn	Fe	Co	Ni

$E^{Hull}$  (meV/atom)



$Ca_xV_2(PO_4)_3$ ,  $Ca_xMn_2(SO_4)_3$ , and  $Ca_xFe_2(SO_4)_3$

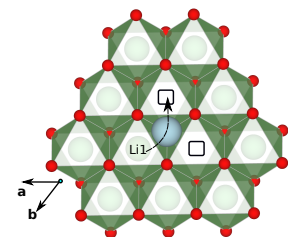


# Conclusions

- Removing material bottlenecks is important for improving performance of energy devices
  - Need cathodes and solid electrolytes with good power performance
  - Need robust computational predictions of ionic mobility
- SCAN provides better accuracy on  $E_m$  predictions, on average
  - Computational costs + convergence difficulties: high
- Ca-containing ternary compounds and NaSICONs screened
  - 2 possible candidates:  $\text{CaV}_2\text{O}_4$  and  $\text{CaNb}_2\text{O}_4$
  - $\text{Ca}_x\text{V}_2(\text{PO}_4)_3$ ,  $\text{Ca}_x\text{Mn}_2(\text{SO}_4)_3$ , and  $\text{Ca}_x\text{Fe}_2(\text{SO}_4)_3$  are promising



भारतीय विज्ञान संस्थान



## NEB Benchmarking:

“Effect of exchange-correlation functionals on the estimation of migration barriers in battery materials”, R. Devi, B. Singh, P. Canepa, and G. Sai Gautam, **npj Comput. Mater.** **2022**, *8*, 160

## Ternary Ca-cathode screening:

“Searching ternary oxides and chalcogenides as positive electrodes for calcium batteries”, W. Lu, J. Wang, G.Sai Gautam, and P. Canepa, **Chem. Mater.** **2021**, *33*, 5809-5821

## Ca-NaSICON screening:

“Exploration of NaSICON frameworks as calcium-ion battery cathodes”, D.B. Tekliye, A. Kumar, X. Weihang, T.D. Mercy, P. Canepa, and G.Sai Gautam, **Chem. Mater.** **2022**, *34*, 10133-10143