

Screening chemical spaces for positive electrodes in beyond-Li-ion batteries using computational techniques

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Why beyond-Li-ion batteries?

Next generation of electric devices will benefit from higher energy density storage systems

- Multi-valent == More electrons (Ca²⁺, Mg²⁺, Al³⁺, etc.)
- Na-ion == earth abundance, cost-effective
- Li-ion technology approaching fundamental limits
 - Safety, supply-chain constraints; limits on achievable energy densities



Why Ca/Na?

- Superior volumetric capacity for Ca metal (~2077 Ah/l) than Li in graphite (~800 Ah/l)
- Ca/Na is safer than Li, less constrained geopolitically
- Na compatible with stainless-steel current collectors vs. Cu for Li



Ca: Find cathodes with reasonable voltage and capacity, and be thermodynamically stable Na: Find cathodes with robust structural stability (for cycle life) with reasonable energy density

Objectives



Discover new cathodes for Ca batteries



Screen for cathodes with robust structural stability for Na-ion batteries



Methods detour (and stability)

Methods: Edison vs. Iron Man



Trial and error of candidates in a lab





Density functional theory: (Approximately) predict material properties

 Exchange-correlation functional: Hubbard U corrected generalized gradient approximation (GGA+U)¹

1. Perdew et al., **Phys. Rev. Lett. 1996**, *77*, 3865

Calculating migration barrier for ionic diffusion: nudged elastic band (NEB)²



0 K thermodynamics: convex hull

E^{hull}: measure of stability of given structure+composition combination (at 0 K)



Intercalation vs. Conversion voltage



Intercalation vs. Conversion voltage



- A discharge (reduction) process will always favor the reaction that yields higher voltage
- Whether intercalation or conversion wins is polymorph (structure) dependent: particularly important in multivalent systems, such as Ca
- Ideally, we want intercalation >> conversion voltages

Conversion		Intercalation		
Discharged	$A + \lambda - M_2 X_4 \rightarrow AMX + M_2 X_3$	Discharged	$A + \lambda - M_2 X_4 \rightarrow \lambda - A M_2 X_4$	İ
Charged	$A + \alpha - M_2 X_4 \rightarrow AMX + M_2 X_3$	Charged	$A + \alpha - M_2 X_4 \rightarrow \alpha - A M_2 X_4$	

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Screen for cathodes with robust structural stability for Na-ion batteries

Let's look at ternary Ca-compounds



- Inorganic crystal structure database (ICSD¹): has > **210,000** compounds
- Only 365 are ternary compounds containing Ca
 - Compounds of composition $Ca_iM_jZ_k$; M, Z = elements other than Ca
- Let M = TM (i.e., transition metal) and Z = O, S, Se, or Te
 - Results in 181 unique compounds
- Charge-neutral charged compound (TM_iZ_k) available for $Ca_iTM_iZ_k$?
 - $CaMn_2O_4$ - Mn_2O_4 is ok, $CaVO_3$ - VO_3 not ok
 - 66 unique structures
- Either of $Ca_iTM_iZ_k$ or TM_iZ_k thermodynamically (meta)stable?
 - $E^{hull} \leq 30 \text{ meV/atom}$ (based on Materials Project²)
 - **10** unique compounds \rightarrow evaluate voltage, mobility

Final

candidates!

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https://materialsproject.org/ 2.

https://icsd.products.fiz-karlsruhe.de/

Voltages calculated with GGA+U

GGA+U chosen instead of SCAN(+U) to lower computational cost



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Ca diffusivity: nudged elastic band

 $D \approx \frac{x^2}{t}$ Electrode particle size Diffusion time (rate of operation)

$$D = \nu a^2 f g x_D \exp\left(-\frac{E_m}{k_B T}\right)$$

Diffusivity mainly governed by barrier Required diffusivity \rightarrow maximum of E_m Variables: size, time, and temperature



Nudged Elastic band: Sheppard et al., J. Chem. Phys. 2008, 128, 134106



2 candidates display reasonable E_m

- CaV₂O₄: 654 meV
- CaNb₂O₄: 785 meV

Migration pathways of candidates



 $CaV_2O_4: 8 \rightarrow 3 \rightarrow 8$ $CaNb_2O_4: 6 \rightarrow 4 \rightarrow 6 \rightarrow 4 \rightarrow 6$

Can we frame some design rules to discover more facile Ca diffusers?

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Design rules: update

- Existing rules to identify facile ionic conductors¹ doesn't work for Ca:
- Avoid structures with Ca's "preferred" coordination of 8
 - CaV_2O_4 ($E_m = 654$ meV) and $CaMoO_3$ (2072 meV) have Ca in 8-coordination
- Reduce changes in coordination number during migration
 - CaV_2O_4 (coordination change of 5) and $CaNb_2O_4$ (change of 2) have low barriers
- Increase volume per anion (i.e., prefer S²⁻ instead of O²⁻) to reduce E_m
 - CaCu₂S₂ (E_m = 1622 meV) has higher barriers than several oxides

Summary:

- Identified two Ca-cathode candidates: CaV₂O₄ (post-spinel) and CaNb₂O₄ (layered)
- Updated design rules to identify other facile Ca conductors

Updated design rules for identifying facile Ca conductors:

- Structures should exhibit optimal area/diagonal/volume fraction of Ca at transition state
- Avoid face-sharing cations at transition state
- Minimize volume fraction change during migration

Objectives



Discover new cathodes for Ca batteries



Screen for cathodes with robust structural stability for Na-ion batteries

NaSICONs: Polyanionic hosts with robust structural stability

Experimentally:

- Single transition-metals $Na_xM_2(PO_4)_3$; M = Ti, V, Cr, and Fe studied
- *Mixed* transition-metal Na_xMM'(PO₄)₃; Ti+V/Mn/Fe/Cr, V+Mn, Cr+Mn explored Theoretically:

Systematic study missing on all 28 M, M' PO₄ combinations across Na concentrations

- Na superionic conductors: NaSICONs, polyanionic hosts
 - Original composition: Na_{1+x}Zr₂P_{3-x}Si_xO₁₂; General composition: Na_xM₂(ZO₄)₃



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Single transition metal NaSICONs $Na_xM_2(PO4)_3$; M = Ti, V, Cr, Mn, Fe, Co, Ni



Triclinic or Monoclinic at other x

Deepest minima for Cr; shallowest for Ni

 $M^{3+/4+}$ for 1 < x < 3; $M^{2+/3+}$ for 3 < x < 4

Minima at x = 3

- Predictions underestimate
- E.g., for V: 3.4 and 1.63 V (experimentally) vs. 2.96 and 1.54 V (theoretically)
- Qualitative trends are ok



Predicted

Ni and Co predicted to have the highest voltages, overall

Mn average voltage (across all Na) is higher than Cr and lower than Fe 19

Mixed transition metal NaSICONs Na_xTiM'(PO₄)₃; M'=Ti, V, Cr, Mn, Fe, Co, Ni



Screening all 28 NaSICON systems



Mn and Co (single) NaSICONs are stable



 $Na_3Mn_2(PO_4)_3$ and $Na_1Mn_2(PO_4)_3$ are both stable

- $E^{hull} = 0$ meV/atom for both compositions
- Na3 is monoclinic, Na1 is rhombohedral
- Needs more efforts to synthesise

 $Na_3Co_2(PO_4)_3$ is stable

- Na3 is rhombohedral
- Na₁Co₂(PO₄)₃ is unstable
- Co inclusion in NaSICON is ok

Ni compositions are unstable: not worth exploring experimentally

Phase diagrams based on 0 K DFT calculations of all available "ordered" structures in inorganic crystal structure database ²²

Conclusions and Acknowledgments

- Removing material bottlenecks is important for improving performance of energy devices
 - Need better, safer, and cheaper batteries (Ca/Na vs. Li)
- Ca-containing ternary compounds from ICSD screened
 - · Screening criteria: redox-activity, charge-neutrality, and thermodynamic stability
 - 2 possible candidates: CaV₂O₄ and CaNb₂O₄
- Screening of 28 unique NaSICON compositions as Na-ion cathodes
 - · Co and Mn based (theoretical) systems are promising: more efforts to synthesise them necessary
 - Cr^{3+/4+} redox useful to explore; Ni not useful

Ca-electrodes:

"Searching ternary oxides and chalcogenides as positive electrodes for calcium batteries", W. Lu, J. Wang, <u>G.S.</u> <u>Gautam</u>, and P. Canepa, **Chem. Mater. 2021**, *33*, 5809-5821

NaSICON screening:

"A chemical map of NaSICON electrode materials for sodium-ion batteries", B. Singh, Z. Wang, S. Park, <u>G.S.</u> <u>Gautam</u>, J-N. Chotard, L. Croguennec, D. Carlier, A.K. Cheetham, C. Masquelier, and P. Canepa, **J. Mater. Chem. A 2021**, *9*, 281-292



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