Shear-Assisted Formation of Cation-Disordered Rocksalt $NaMO_2$ (M = Fe, Mn)

Supplementary Materials

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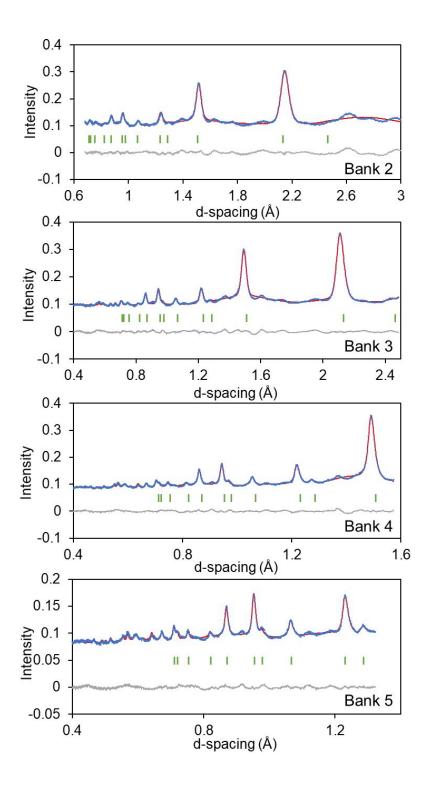


Fig. S1. Neutron diffraction patterns of the as-synthesized rocksalt NaFeO₂ fitted to the same structure used in synchrotron diffraction refinement ($Fm3\bar{m}$, a = 4.37 Å). The slight difference in the fitting mainly results from the excess carbon in the sample, which causes additional scattering with neutron.

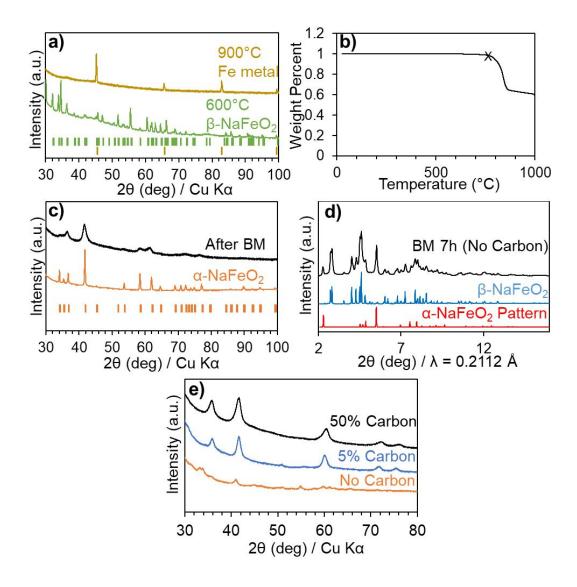


Fig. S2. Cation-disordered rocksalt NaFeO₂ cannot be synthesized without ball milling, carbon and β -NaFeO₂ as the precursor. a) XRD pattern of the β -NaFeO₂ and carbon mixture after heated to different temperatures in Ar atmosphere. After heated to 900°C, the β -NaFeO₂ were reduced to Fe metal. b) Thermal gravimetric analysis shows the reduction to Fe metal takes place at ~ 750°C. c) XRD patterns of the α -NaFeO₂ before and after the same ball milling condition as the β -NaFeO₂. The only observable change is the peak broadening due to the particle size reduction during the milling process. d) XRD pattern of β -NaFeO₂ ball milled for 7 hours without carbon shows a majority of the material remains β structure with small amount transformed into α phase. e) XRD patterns of the β -NaFeO₂ ball milled for 7 hours with different amount of carbon. Rocksalt phase can be observed with as little as 5 wt% carbon added.

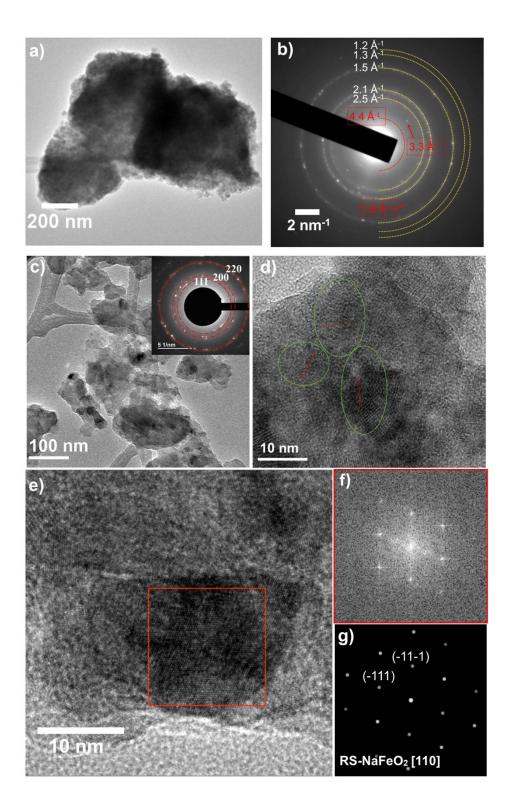


Fig. S3. TEM images and ED patterns of both 3-h-ball-milled (a and b) and 12-h-ball-milled samples (c through g). a) TEM image of a large particle (~ 500 nm) in the 3-h-ball-milled sample.
b) The electron diffraction of the whole particle shown in a showing the coexistence of rocksalt

phase and β -NaFeO₂ phase. The majority diffraction signal can be identified as disordered rocksalt (labelled yellow). However, β -NaFeO₂ diffraction spots are also visible (labelled red). **c)** TEM image and electron diffraction pattern (insert) of 12-h-ball-milled sample. The small dark spots are the nano-sized NaFeO₂ particles embedded in carbon matrix, which is shown as the light and cloudy area. The electron diffraction pattern also shows a rocksalt structure symmetry, consistent with the X-ray diffraction data. **d)** High magnification TEM image showing the grain sizes of 12-h-ball-milled sample has been reduced to < 10 nm. Several grains are highlighted by the green circles and their respective lattice fringes are indicated by the red dash lines. **e)** High magnification TEM image of a single grain in NaFeO₂ particle in the12-h-ball-milled sample. **f)** FFT of the area in the red square in **e**. **g)** Simulated diffraction spots of an ideal disordered rocksalt NaFeO₂ structure showing good agreement with **f**.

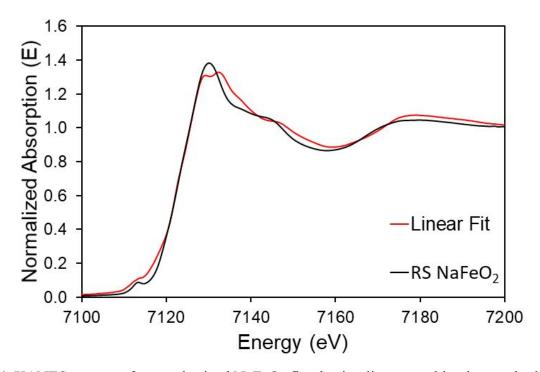


Fig. S4. XANES spectra of as synthesized NaFeO₂ fitted using linear combination method with 32.2% FeO and 67.8% Fe₂O₃.

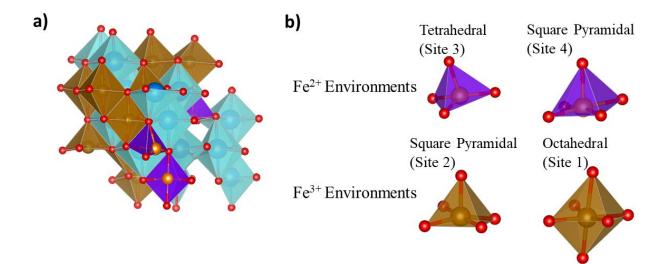


Fig. S5. a) DFT calculated structure for cation disordered rocksalt NaFeO₂ with 6% O vacancies. Maximum Fe-O bond was set to 4.5 Å. **b)** Four different Fe local environments isolated from DFT calculation structure from a). There are two different environments for Fe²⁺ (resembling tetrahedral and square pyramidal coordination) and two for Fe³⁺ (resembling octahedral and square pyramidal coordination). This coordination difference rises from their different valence states and their relative positions with respect to the O vacancies. It is expected that oxygen vacancies are more likely to be created near a reduced Fe ion (Fe²⁺), and therefore resulting in a lower averaged coordination number for Fe²⁺ ions compared to Fe³⁺ ions. All sites show large distortions; therefore, their chemical shifts (δ) and quadruple splittings (Δ EQ) cannot be directly compared with reference values. Instead, each fitting profile is assigned to the site that has closest reference values for both δ and Δ EQ.

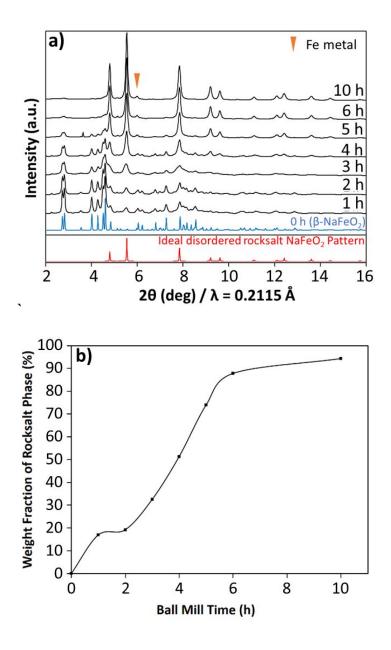


Fig. S6. a) Synchrotron XRD patterns of β -NaFeO₂ after different ball milling times. b) Estimated weight fraction of rocksalt phase during ball milling obtained from Rietveld refinement of synchrotron XRD data.

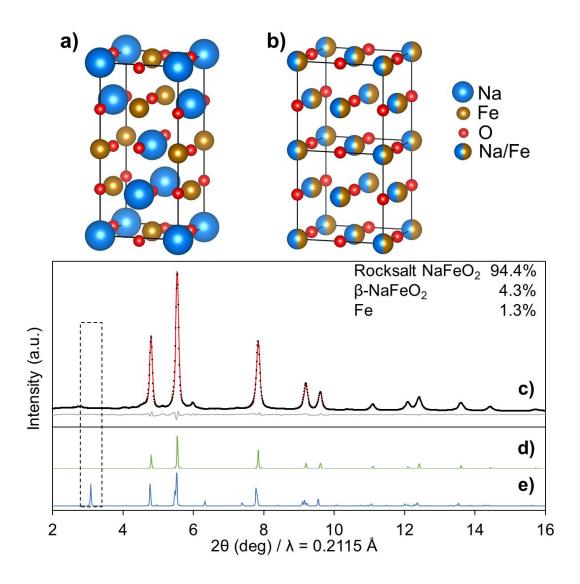


Fig. S7. a) Structure model of ordered rocksalt NaFeO₂ with zigzag cation ordering inherited from β -NaFeO₂ (similar to γ -LiFeO₂ ordering). **b)** Structure model of cation disordered rocksalt NaFeO₂ with Na and Fe atom sharing the same site. **c)** Rietveld-refined synchrotron X-ray diffraction pattern of as prepared cation-disordered rocksalt NaFeO₂ with small amount of Fe impurity and residual β -NaFeO₂. Note that the cation ordering peak at ~ 3 Å is absent. **d)** Simulated diffraction pattern of ideal disordered rocksalt NaFeO₂ shown in b). **e)** Simulated diffraction pattern of ordered rocksalt NaFeO₂ with zigzag cation ordering inherited from β -NaFeO₂ (shown in a)).

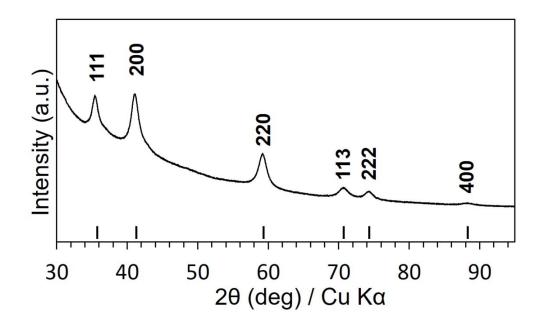


Fig. S8. XRD pattern of disordered rocksalt NaMnO₂ synthesized by ball milling β -NaMnO₂ with carbon. The sloping background is due to the Kapton tape used to prevent air exposure during the XRD scan.

Phase name	Cation-Disordered Rocksalt NaFeO ₂								
Atom	Site	Site occupancy x y z U							
Na	4a	0.5	0	0	0	0.005982			
Fe	4a	0.5	0	0	0	0.005986			
0	4b	0.95	0	0	0.5	0.03337			

Space group: $Fm3\overline{m}$; a = 4.37 Å; R_w = 2.82; goodness of fit = 2.74.

Table S1. Synchrotron	XRD data	Rietveld	refinement	results	for	cation-disordered	rocksalt
NaFeO ₂ .							

NaFeO ₂ Structure	O Vacancy (eV)	2 O vacancies (eV)	
β	3.4	6.7	
Rocksalt	2.8	5.6	
α (ground state)	2.9	6.8	

Table S2. DFT calculated defect formation energies for different NaFeO₂ structures. Rocksalt structure shows the lowest formation energy for oxygen vacancies among the three, and the energy differences become greater as the vacancy concentration increases. Oxygen vacancy formation energies are determined by removing Oxygen from a supercell with 32 oxygen atoms. Different vacancy positions are tested to find the lowest energy configuration.

O vacancy conc.	0	3.1	6.3
E_{RS} - E_{β} (meV/atom)	50.25	37.6	22.9

Table S3. DFT calculations showing decreasing energy differences between rocksalt structure and β structure NaFeO₂ with increasing oxygen vacancy concentration.

Bond	ΔE (eV)	Ν	R (Å)	DWF (10 ⁻³ Å ²)	R-factor (%)
		2	1.0() 0.00	7.0(+ 4.01	
Fe-O1		3	1.96 ± 0.02	7.06 ± 4.01	
Fe-O2		2	2.10 ± 0.04	7.06 ± 4.01	
Fe-O3		1	2.63 ± 0.11	7.06 ± 4.01	
	-0.41 ± 2.01				0.94
Fe-Na1		3	2.88 ± 0.05	9.05 ± 10.01	
Fe-Fe1		6	3.09 ± 0.04	10.24 ± 1.77	
Fe-Na2		3	3.32 ± 0.07	9.05 ± 10.01	

 ΔE : inner shell potential shift; N: coordination numbers; R: bond distance; DWF: Debye–Waller factor (σ^2 , disorder).

Table S4. Curve fitting results for the Fe K-edge EXAFS spectrum of the as-prepared cationdisordered rocksalt NaFeO₂ shown in Fig. 2b.

Empirical Pair Potential

The empirical pair potential is of the following form with parameters listed in Table S4:

$$V_{ij} = \frac{q_i q_j}{r_{ij}} + 4\varepsilon_{ij} \{ (\frac{\sigma_{ij}}{r_{ij}})^{12} - (\frac{\sigma_{ij}}{r_{ij}})^6 \}$$

	<i>q</i> (e)	ε(eV)	σ(Å)
Na	0.525671	8.65971E-04	1.18606
Fe	1.455048	2.27328E-05	1.58667
0	-0.990359	1.78159E-06	6.31161

Table S5. Parameters for the NaFeO₂ pair potential.

For the Lennard-Jones part, combination rules are used when calculating interactions between different species. The pair potential well reproduces relative stabilities between the α -, β - and ordered rocksalt NaFeO₂ from DFT. Fig. S9 shows the minimum energy path calculated by the pair potential comparing with the DFT one (Fig. 6a, black curve).

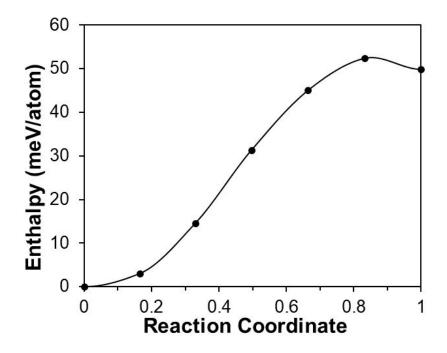


Fig. S9. The minimum energy path for β to rocksalt NaFeO₂ phase transition calculated with the empirical pair potential.