## **Supplementary Information**

## Investigating Ionic Diffusivity in Amorphous LiPON using Machine Learned Interatomic Potentials

Aqshat Seth, Rutvij Pankaj Kulkarni, Gopalakrishnan Sai Gautam\*

Department of Materials Engineering, Indian Institute of Science, Bengaluru 560012, India

\*Corresponding author: saigautamg@iisc.ac.in

## **Dataset Generation**

The unit cells for each of the 19 systems shown in Table 1. were relaxed and used to generate the rest of the dataset using the pymatgen<sup>1</sup> package. Further information on each category of structures and the number of structures in each category can be found below:

Chemical Systems
Li
Li <sub>2</sub> O
Li <sub>2</sub> O <sub>2</sub>
LiO <sub>8</sub>
LiP
Li <sub>3</sub> P
LiP <sub>7</sub>
Li <sub>3</sub> P <sub>7</sub>
Li <sub>3</sub> N
LiN <sub>3</sub>
LiPN <sub>2</sub>
Li <sub>3</sub> PO <sub>4</sub>
Li <sub>4</sub> P <sub>2</sub> O <sub>7</sub>
Li <sub>7</sub> PN <sub>4</sub>
Li <sub>2</sub> PNO <sub>2</sub>
$P_3N_5$
$P_2O_5$
PNO
LiPN <sub>2</sub>

Table S1. Initial set of structures used to generate the DFT-based dataset.

**Hydrostatic Strain:** 306 structures were generated by uniformly applying strains along the lattice directions with an increment of 1% volumetric change between the initial bulk structures and the hydrostatically strained structures starting from -10% and going upto +9%.

**Orthorhombic Strain:** 328 structures were generated by uniformly applying strains along one lattice direction, while keeping the other two lattice parameters fixed with an increment of 1% volumetric change between the initial bulk structures and the orthorhombically strained structures starting from - 10% and going upto +9%.

**Monoclinic Strain:** 252 structures were generated by uniformly changing one of the angles of the initial bulk structures while keeping the volume of the structure unchanged, with an incremental change of 10%.

**Slabs:** Slabs were constructed with a vacuum spacing of 20 Å and slab thicknesses between 10 Å to 30 Å. 72 slabs from elemental Li, 530 slabs from Li<sub>3</sub>P, 252 slabs from Li<sub>2</sub>O<sub>2</sub>, 179 slabs from Li<sub>2</sub>O and 186 slabs from Li<sub>3</sub>N were added to the dataset.

**Defective Structures:** The AIMD-based Li-rich and Li-poor supercell structures were generated at a composition of  $L_{i_{3.04}}PN_{0.046}O_{3.95}$  (39 structures) and  $L_{i_{2.95}}PN_{0.046}O_{3.91}$  (88 structures) respectively. The Li-rich and Li-poor unit cell structures were generated at compositions of  $L_{i_{3.5}}PN_{0.5}O_{3.5}$  (40 structures) and  $L_{i_{2.5}}PN_{0.5}O_3$  (89 structures) respectively. The structures generated by enumerating the removal of 1 P and the concomitant addition of 5 Li within  $L_{i_7}PN_4$  2x2x2 supercells were at a composition of  $L_{i_{8.71}}PN_{4.57}$  (74 structures)



**Figure S1.** Radial distribution function (RDF) of ab initio molecular dynamics (AIMD) generated melt-quench structures of a) Li<sub>3</sub>N, b) Li<sub>3</sub>P, and c) Li<sub>2</sub>O. The RDFs displayed are calculated in the quenched structure at 250 K, obtained after being heated to 1000 K in the case of Li<sub>3</sub>N and 2000 K for Li<sub>3</sub>P and Li<sub>2</sub>O.

**Table S2.** Hyperparameter values used for training the neural equivariant interatomic potential (NequIP) model. The training energy and force MAE were found to be of 5.5 meV/atom and 13.6 meV/Å, respectively with the validation energy and force MAE being 6.1 meV/atom and 13.2 meV/Å respectively.

Hyperparameter	Optimized Value
Cutoff radius	4.3 Å
Interaction blocks	6
Tensor product layers per interaction block	2
Neurons in each product layer	64
Basis function in trainable basis set	8
Polynomial cutoff for envelope function	6 Å
Learning rate	0.0044



**Figure S2.** Time-averaged RDFs upon quenching to 250 K for NequIP-generated amorphous LiPON structures. The structure is initially melted and quenched to a) 600 K, b) 900 K, c) 1200 K and d) 1500 K, before being finally quenched to 250 K.



**Figure S3.** Time-averaged RDFs at 250 K for NequIP-based amorphous LiPON structures. The structures are initially equilibrated at a) 600 K, b) 900 K, c) 1200 K and d) 1500 K before being quenched to 250 K.



**Figure S4.** Mean square displacement (MSD) versus time-step ( $\Delta t$ ) plots for the Li(110)||LiPON and Li(111)||LiPON interfaces in the directions of the lattice parameters and the total MSD. Plots represent calculations at 300 K, 600 K and 900 K, with a), b) and c) representing the Li(110)||LiPON interface and d), e) and f) representing the Li(111)||LiPON interface. Here, *c* refers to the perpendicular direction across the interface.



**Figure S5.** a) The Li-(111)||LiPON interface. b) Variation in the average Li-Li bond length and c) variation in the average number of Li neighbors along the *c*-axis.



**Figure S6**. Comparisons of the time-averaged RDFs of a) AIMD generated LiPON and b) Melt-quench based NequIP generated LiPON.



**Figure S7**. Mean square displacement (MSD) versus time-step ( $\Delta t$ ) plots for the melt-quench configuration generated via NequIP at 300 K.



Figure S8. Li(110)||LiPON interface consisting of 1316 atoms.

## References

 Ong, S. P.; Richards, W. D.; Jain, A.; Hautier, G.; Kocher, M.; Cholia, S.; Gunter, D.; Chevrier, V. L.; Persson, K. A.; Ceder, G. Python Materials Genomics (Pymatgen): A Robust, Open-Source Python Library for Materials Analysis. *Comput. Mater. Sci.* 2013, 68, 314–319. https://doi.org/10.1016/j.commatsci.2012.10.028.