Unveiling Structural Integrity of Tunnel type - Na0.44MnO² Cathode for Sodium Ion Battery

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Experimental:

Synthesis of Na0.44MnO2:

The P2 type $\text{Na}_{0.44}\text{MnO}_2$ was synthesized through a facile and scalable solid-state approach using high purity precursors. All the chemicals were acquired from Sigma Aldrich and used without further purification. Stochiometric amount of Na_2CO_3 , and $MnCO_3$ is mixed well and pre calcined at 300 $^{\circ}$ C @ 2 $^{\circ}$ min⁻¹ for 8h. Upon cooled down to the room temperature the powder is well grinded to make pellets and undergo another sintering at 850° C ω 5 °min⁻¹ for 9h in Platinum boat. All the processes are performed in ambient conditions.

Physical characterization:

Structural characterization is employed using Bruker D8 Advanced X-ray diffractometer using Cu Kα radiation ($λ=1.5414Å$) within the 2θ range from 10° to 50°. The data were fitted with the Rietveld refinement method using MAUD software^{1,2}. The refinement was performed using

the orthorhombic structure with a $Pbam$ space group, considering the lattice constants, $a =$ 9.1Å, $b = 26.34$ Å, and $c = 2.82$ Å. The morphological analysis was studied by high resolution scanning electron microscope (HRSEM, FEI, Magellan 400 L) at the voltage of 5kV and current of 0.4 nA. The high-resolution transmission electron micrograph (HRTEM, JEM 32100, JEOL, accelerating voltage of 200kV, Gatan USC 4000 4x4k camera) of coated sample was recorded on lacey carbon-coated Cu gride. The elemental composition of the synthesized materials was investigated by Inductively coupled plasma atomic emission spectrometry (ICP-AES) technique with an Ultima-2 spectrometer (JobinYvon Horiba). The oxidation states of the various elements in each sample were measured using X-ray photoelectron spectroscopy (XPS, Thermo Scientific Nexsa soecro spectrometer).

Electrochemical study:

Electrode preparation: For making the slurry, the active material, Super P carbon black (Sigma Aldrich) and 10 % of poly-vinylidene fluoride (PVDF, Sigma Aldrich) binder in Nmethyl-2-2pyrrolidone (NMP, Sigma Aldrich) is taken in 80:10:10 ratio and mixed in a Thinky mixer. The slurry was coated on a clean and polished Al foil using Doctor blade at thickness of 150 µm and dried for overnight at 100°C. After rolling, 12 mm of the electrode was punched, measured the weight, and kept overnight under dynamic vacuum at 120°C to avoid moisture.

Electrochemical test: The electrochemical study is conducted in a two-electrode cells configuration using coin-type cells (CR2032) in which Na metal (14 mm) is used as both counter and reference electrode. 1M NaClO4/Ethylene carbonate (EC)/Diethylene carbonate (DEC) in 4:6 ratio is used as electrolyte solution. A total of 80 µl electrolyte solution is preserved with two Glass fiber separator (19mm) between working and Na electrode. The electrochemical measurements are performed in Bio-logic Science Instruments at 30°C in ambient condition. The galvanostatic discharge-charge cycling is recorded within the potential window of 1.7 V to 4.0 V at CC-CV mode and the rate currents are normalized by active mass of the working electrode for each material.

Computational Methods:

Candidate structures

We took the disordered-Na_{0.44}MnO₂ structure, as obtained from the experimental X-ray diffraction refinement and enumerated the possible symmetrically-distinct Na-vacancy configurations by using the pymatgen package³. In total, we generated 647 unique configurations and took the 20 structures with the lowest electrostatic energies (computed via Ewald summations)⁴ for further density functional theory (DFT)^{5,6} calculations to identify the ground state Na-vacancy ordering. We followed a similar procedure for finding the ground state structure of $\text{Na}_{0.22}\text{MnO}_2$, where we generated 476 and 70 unique Na-vacancy configurations by using the original, disordered- $Na_{0.44}MnO₂$ structure, and the DFT-identified ground state Na-vacancy configuration at $Na_{0.44}MnO₂$, respectively. Subsequently, we used a total of 20 lowest electrostatic energy configurations for performing DFT calculations, with 12 and 8 configurations taken from the 476 and 70 set of structures, respectively, to identify the ground state configuration. The DFT-calculated ground state Na-vacancy configurations at both the $\text{Na}_{0.44}\text{MnO}_2$ and $\text{Na}_{0.22}\text{MnO}_2$ compositions are displayed in Fig. S2.

We used a $1\times1\times3$ supercell for the NEB calculations to minimize interactions with periodic images and introduced a Na vacancy at the $Na_{0.22}MnO₂$ composition to model each hop. The minimum energy path (MEP) was initialized by a linear interpolation comprising of seven images between the initial and final configurations with a spring constant of $5 \text{ eV} \text{ Å}^{-1}$. For all NEB calculations, including relaxation of the initial and final images, we used the SCAN functional without any Hubbard *U* corrections, since SCAN yields more accurate E_{m} , on

average, across several systems⁷. We relaxed the initial and final images with the same convergence criteria as we had used for identifying ground state structures (see SI). We converged our NEBs till the force component perpendicular to the elastic band dropped below |0.05| eV Å−1 . For the NEB calculations, we used Γ-centered *k*-point meshes with density of 32 *k*-points per Å.

DFT Calculations

We performed all DFT calculations with the Vienna *ab initio* simulation package (VASP)^{8,9}, employing the projector-augmented-wave potentials¹⁰ identical to our previous work^{11,12}. We expanded the plane wave basis up to a kinetic energy cutoff of 520 eV and sampled the irreducible Brillouin zone with a Γ-centered *k*-point mesh, where the density of the mesh used was at least 32 *k*-points per Å. We relaxed the lattice vectors, cell shape, and cell volume of all structures until the total energies and atomic forces converged within 0.01 meV and |0.03| eV Å−1 , respectively. For all structure relaxation calculations, we used the Hubbard *U* corrected ^{13,14} strongly constrained and appropriately normed $(SCAN)^{15}$ functional (i.e., $SCAN+U$) to describe the electronic exchange and correlation, where we used a *U* value derived in previous works^{12,16,17}.

Results and discussions:

Figure S1: Elemental analysis by (a) EDS mapping and (b) corresponding spectra

Figure S2. Structure of Na_{0.44}MnO₂ (a) and Na_{0.22}MnO₂ (b). Yellow, purple, and red spheres indicate Na, Mn and O atoms respectively. Mn atoms labelled from 7 to 10 have an oxidation state of +3 and exhibit square pyramidal coordination, while the other Mn atoms have an oxidation state of +4 and exhibit octahedral coordination.

Figure S3. Calculated final minimum energy pathways traced by Na⁺ in a) hop 1, b) hop 2, and c) hop 3. Note that hop 3 occurs diagonally across the Na tunnels in the $Na_{0.22}MnO₂$ structure. Purple and brown polyhedra signify MnO_6 octahedra and MnO_5 square pyramids.

Figure S4. (a) Desodiation ordering of sodium ion within specific potential range

Figure S5. (a) Cyclic voltammetry at different sweep rate (b) plotting of peak current versus square root of scan rate

Figure S6. (a) Galvanostatic cycling stability and (b) voltage profile for different cycles at 0.5C rate

Figure S7. (a) HRSEM image of post-cycled cathode (b) magnified HRSEM image of selected area

Figure S8. Post-cycled XRD pattern of tt-NMO cathode

Table S1:

Lattice parameters

Table S2:

Structure parameters

Table S3:

Bond Length	Pbam	This study
Na1-O2	2.436	2.5344
Na1-O3	2.376	1.5837
Na1-O5	2.797	2.9429
Na1-O7	2.577	2.6234
Na1-O8	2.53	2.856
Na2-O4	2.36	2.0655
Na2-O6	2.381	3.1244
Na2-O1	2.452	2.524
Na3-01	2.48	2.7029
Na3-O5	2.59	2.7355
Na3-O6	2.572	2.4754
$\overline{Na4-O1}$	2.493	2.5332
Na4-O4	2.416	2.4675
Na4-O6	2.236	2.328
Na4-O9	2.314	1.2781
$Mn1-O2$	1.866	2.0623
$Mn1-O3$	1.902	2.0341
$Mn1-O8$	3.403	3.5375
$Mn2-O4$	1.919	2.1385
$Mn2-06$	1.901	1.943
$Mn3-O2$	2.146	2.2606
$Mn3-O7$	1.892	1.7823
$Mn4-O4$	1.928	2.0516
$Mn4-O5$	1.887	1.425
$Mn4-O7$	3.538	2.4463
$Mn5-O1$	1.869	2.1032
$Mn5-O3$	1.943	2.4279
$Mn5-O9$	2.074	2.4670

Bond length comparison between standard structure and this study:

Table S4. Calculated on-site magnetic moments and oxidation states of Mn atoms within the unit cell of $Na_{0.22}MnO_2$. The Mn sites with $+3$ oxidation states are labelled in **Figure S3** as well (i.e., Mn sites in square-pyramidal coordination).

Table S5: comparison between our material and reported literature.

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